



**Universidade de São Paulo**  
**Instituto de Física**

# **“Produção e Caracterização de Filmes Finos”**

**José Fernando Diniz Chubaci**

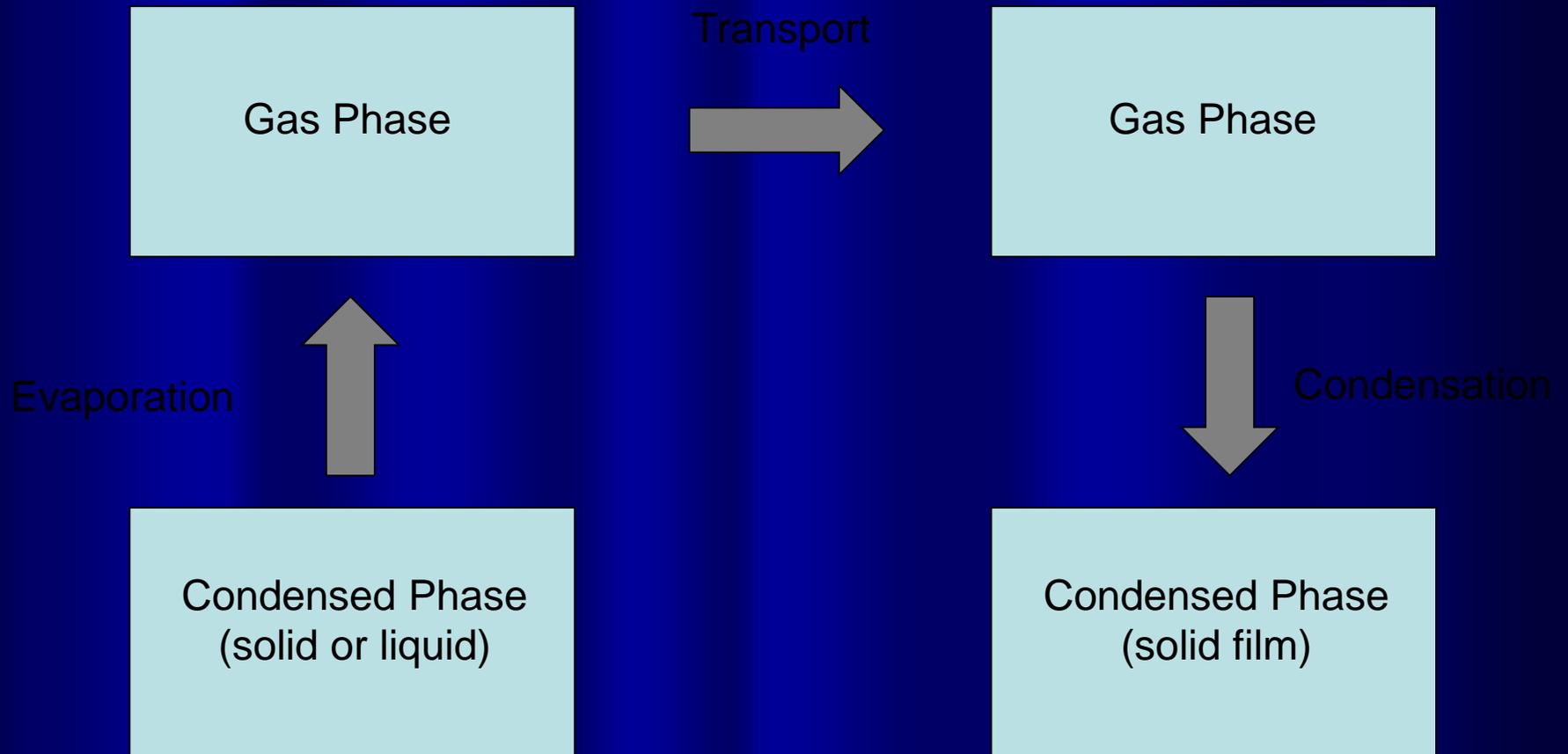
**LACIFID – Laboratório de Cristais Iônicos, Filmes Finos e Datação**

**DFN – Departamento de Física Nuclear**

**IFUSP – Instituto de Física, Universidade de São Paulo, São Paulo**

# Summário

- ☞ Filmes Finos - processos;
- Processos de formação de filmes finos;
- Óxidos e Nitretos;
- Perspectivas.



## Physical Vapor Deposition (PVD).

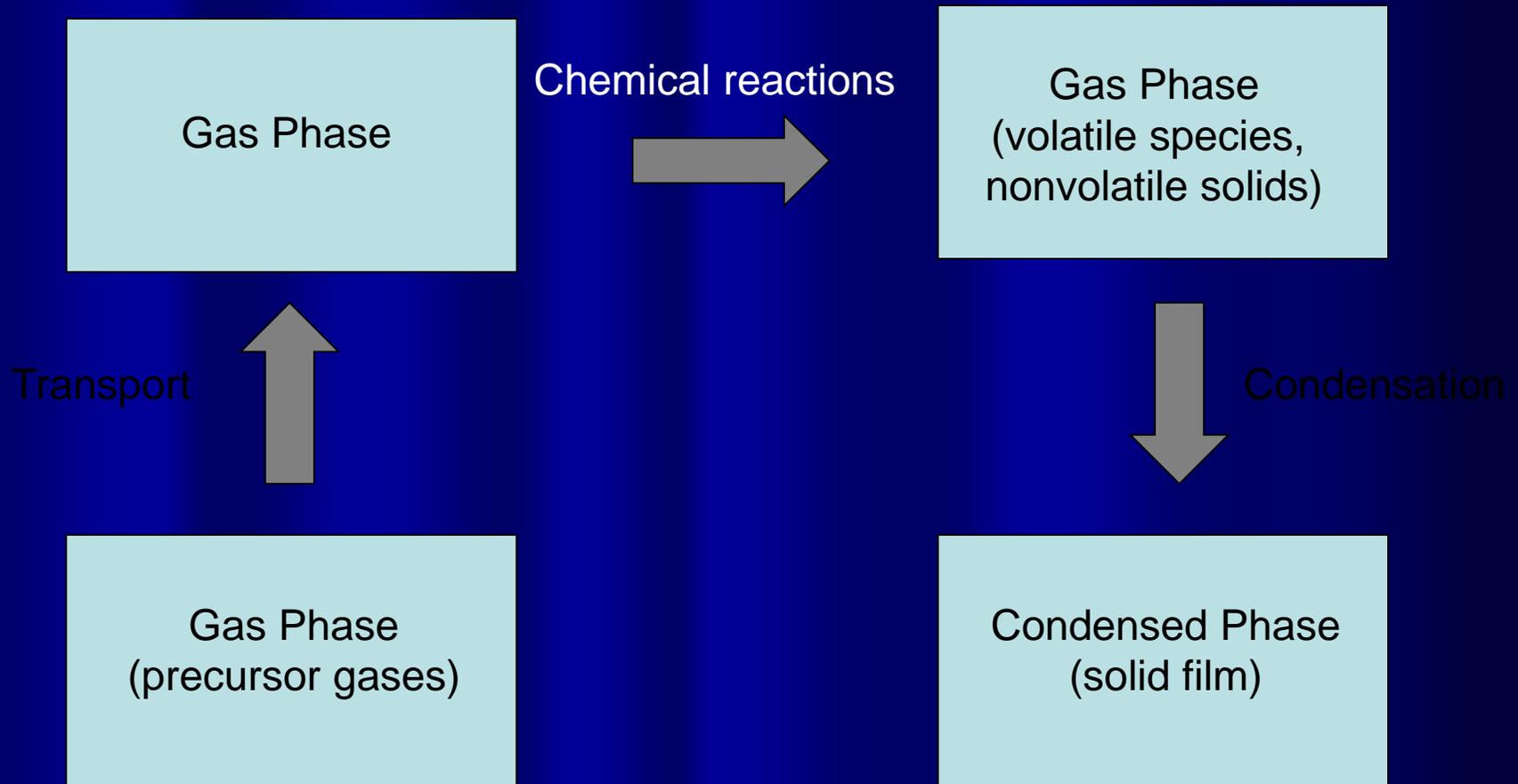
Pyrolysis:  $\text{SiH}_4 (\text{g}) \rightarrow \text{Si} (\text{s}) + 2\text{H}_2 (\text{g})$ , 650°C

Reduction:  $\text{SiCl}_2 (\text{g}) + 2\text{H}_2 (\text{g}) \rightarrow \text{Si} (\text{s}) + 4\text{HCl} (\text{g})$ , 1200°C

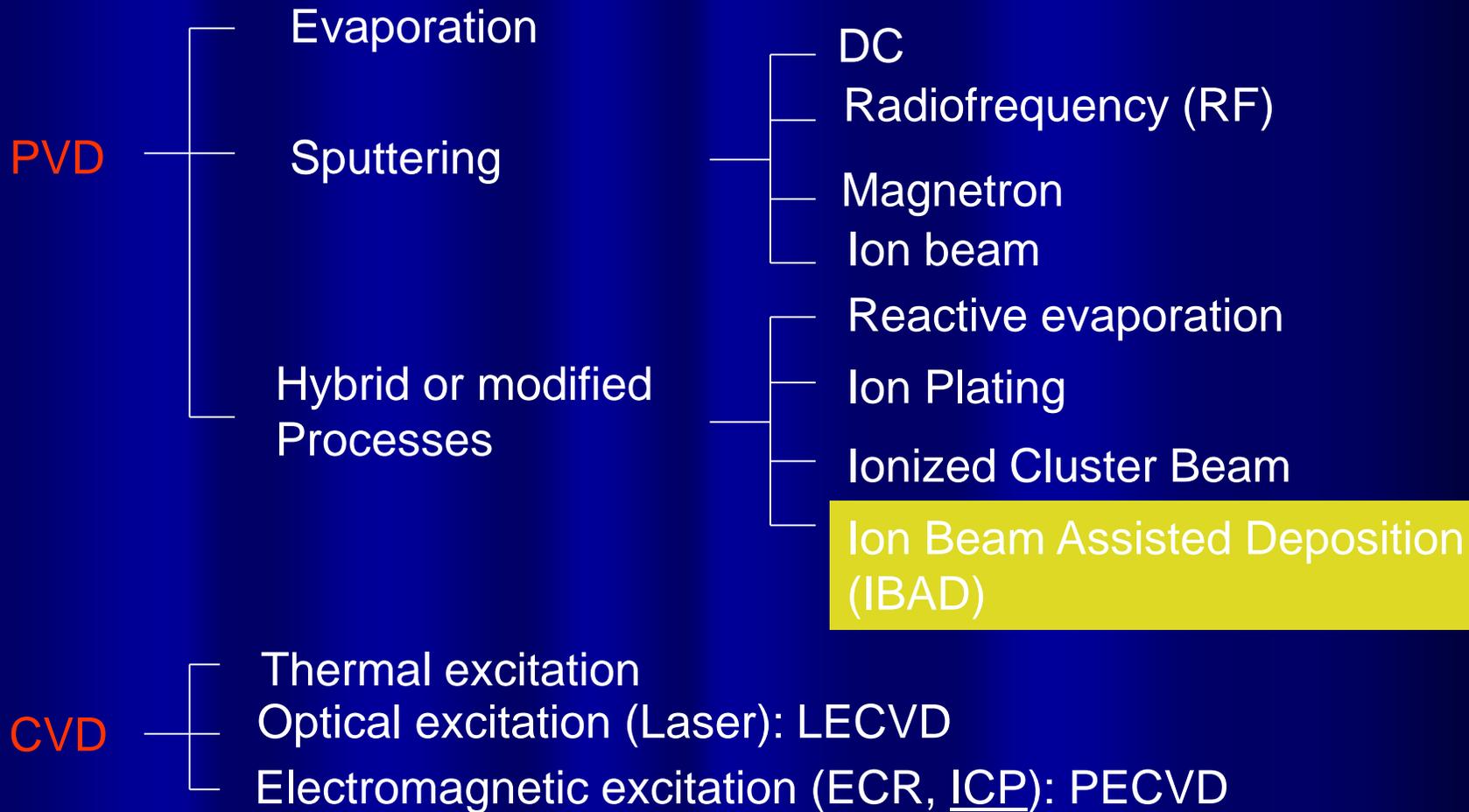
Oxidation:  $\text{SiH}_4 (\text{g}) + \text{O}_2 (\text{g}) \rightarrow \text{SiO}_2 (\text{s}) + 2\text{H}_2 (\text{g})$ , 450°C

Carbide:  $\text{SiCl}_4 (\text{g}) + \text{CH}_4 (\text{g}) \rightarrow \text{SiC} (\text{s}) + 4\text{HCl} (\text{g})$ , 1400°C

Nitride:  $\text{BF}_3 (\text{g}) + \text{NH}_3 (\text{g}) \rightarrow \text{BN} (\text{s}) + 3\text{HF} (\text{g})$ , 1100°C



Chemical Vapor Deposition (CVD).



## Thin-Film Formation Processes.

Ultra-High Vacuum

High Vacuum

Rough (Low) Vacuum

Surface Analysis

e-beam evaporation

DC, RF, magnetron sputtering

Molecular Beam  
Epitaxy (MBE)

Ion beam sputtering

Ion plating

Ion Implantation

Reactive evaporation

Low-Pressure Chemical  
Vapor Deposition (LPCVD)

Deposition processes at three vacuum ranges.

# Chemical Vapour Deposition (CVD)

- Gaseous compounds react to form a dense layer on a heated substrate. The most widely deposited wear-resistant coatings are TiC, TiN, chromium carbide and alumina. Deposition temperatures are generally in the range 800-1000°C which restricts the range of materials which can be coated and can lead to component distortion. Thicknesses are limited to about 10µm due to the thermal expansion mismatch stresses which develop on cooling which also restrict the coating of sharp edged components.
- **Advantages**
  - High coating hardness
  - Good adhesion (if the coating is not too thick)
  - Good throwing power (i.e. uniformity of coating)
- **Disadvantages**
  - High temperature process (distortion)
  - Sharp edge coating is difficult (thermal expansion mismatch stresses)
  - Limited range of materials can be coated
  - Environmental concerns about process gases

# Physical Vapour Deposition (PVD)

- **Advantages**

- Excellent process control
- Low deposition temperature
- Dense, adherent coatings
- Elemental, alloy and compound coatings possible

- **Disadvantages**

- Vacuum processes with high capital cost
- Limited component size treatable
- Relatively low coating rates
- Poor throwing power without manipulation of components

- low pressure coating processes in which the coating flux is produced by a physical process. There are two main types:-

- Evaporation

- Sputtering

- In both cases the source material is a solid (metal or ceramic). A reactive gas may be used in the deposition chamber to deposit compound coatings from an elemental source or maintain the stoichiometry of coatings from compound sources. Typical coating thicknesses range from 1-10 $\mu$ m for wear-resistant coatings, though thinner layers are used in microelectronics and thicker layers are used for high temperature corrosion protection of gas turbine components.

# Evaporation Processes

The vapour pressure of most materials increases with temperature and if it exceeds the ambient pressure the material will rapidly evaporate into the environment. In a coating chamber the pressure is reduced and the source material heated until a desired vapour flux is maintained which is controlled by the source material, the source temperature and the system pressure.

Heating can be performed in several ways:-

- Resistive heating (e.g. aluminium evaporation from  $\text{TiB}_2$  boat)
- Electron beam evaporation (e.g. metals such as tungsten)
- Cathodic arc evaporation (e.g. titanium evaporation for TiN coatings)

The vapour pressures of different metals vary over several orders of magnitude so it is difficult to evaporate alloys and control composition.

As-deposited evaporated coatings are porous due to the limited mobility of coating atoms on component surfaces. This can be controlled by heating or ion plating (see later)

Spatter from localised boiling can lead to droplet formation which affects coating performance

# Sputtering Processes

- Main sputtering processes:-

- DC diode sputtering  
(for conducting targets)

- RF sputtering  
(for insulating targets)

- When energetic ions strike a surface, material is ejected by the transfer of momentum from the ion to the target atoms (akin to billiard ball collisions at the atomic scale). This can be conveniently achieved in a low pressure glow discharge of an inert gas such as argon.
- In such a process the target material is made the cathode and is raised to a potential of several hundred volts. Electrons leaving the cathode stream out into the gas phase where they can impact with argon atoms, ionising them. The positively charged argon is then accelerated to the cathode where it impacts and sputters away material.
- The sputtering yields of different elements for given impact conditions do not vary very much so target alloy compositions can be maintained in the coating except in cases where there are large differences in the atomic weights of alloy constituents.

The coating rate scales with the electrical power used to sustain the discharge. The coating rate also depends on the plasma density, so techniques to increase this (e.g. by confining the electrons close to the target using magnets) will increase the coating rate. However, as much as 95% of the power is dissipated as heat in the target so good cooling is essential.

# Ion Implantation

- **Advantages**
- Low temperature process
- Very versatile - every stable element in the periodic table can be implanted into any vacuum compatible target
- Highly controlled
- No distortion - can be applied to finished components
- Not a coating process
- **Disadvantages**
- Line of sight process
- Expensive vacuum equipment needed
- Very thin treated layer
- A vacuum process in which a beam of ions is directed at the surface and injected into it. The ions lose energy in collisions with the target atoms and come to rest in the surface layer of the material with an approximately Gaussian distribution. The ion penetration depth depends on the ion species, ion energy and target material, but is generally less than  $1\mu\text{m}$ . For steels the main ion used is nitrogen, which hardens the surface by forming nitride precipitates and solid solutions. The damage introduced by the implantation process also introduced a compressive residual stress which improves fatigue performance.

Ion implantation is routinely used for semiconductor doping and treatment of expensive plastics injection moulding tools where any wear is detrimental.

# Welding Processes

- **Advantages**

- Cheap
- Applicable to large components
- Localised coating possible
- Excellent adhesion

- **Disadvantages**

- Limited range of coating materials
- Minimum thickness limits

- The same methods which can be used for joining materials can be used to deposit wear resistant coatings (hardfacings). Coating materials range from low alloy steels to tungsten carbide composites.
- High deposition rates are possible and very thick coatings can be produced. It is impractical to produce layers less than 2-3mm thick.
- There can be problems with cracks in weld deposits

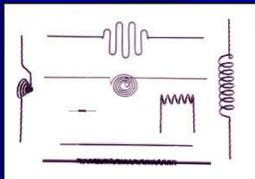
# Thermal Evaporation in Vacuum: From Source to Substrate

we will first discuss the thermal evaporation parameters, such as vapor pressure, evaporation rates and directionality. This will be followed by a technical description of various types of evaporation sources, both resistively heated and electron beam heated.

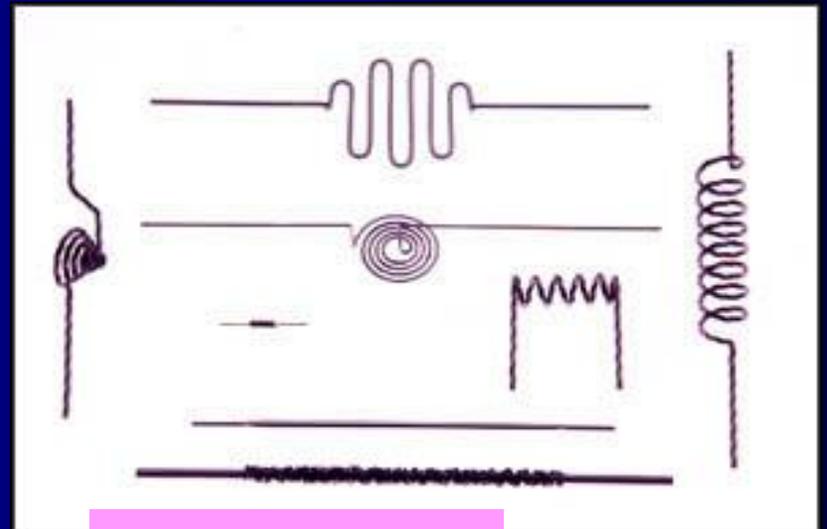
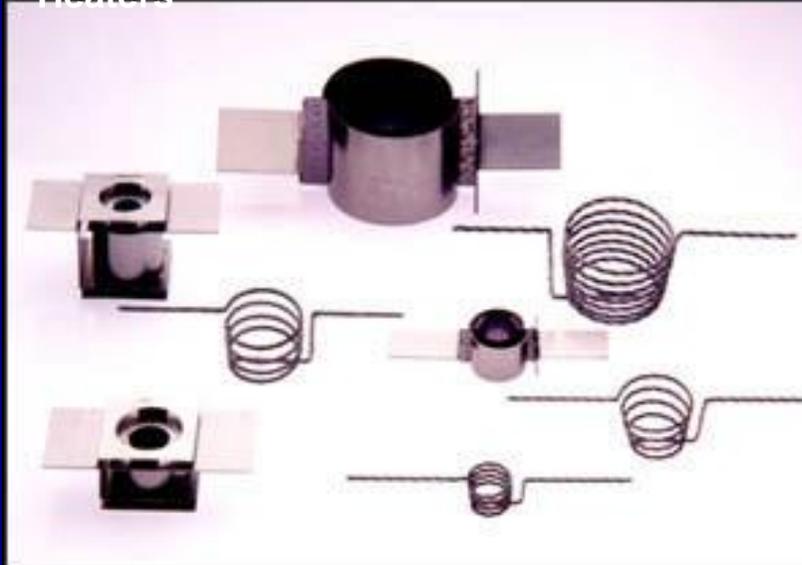
We will then discuss how to monitor and control the evaporation processes, with emphasis especially on in-process techniques.



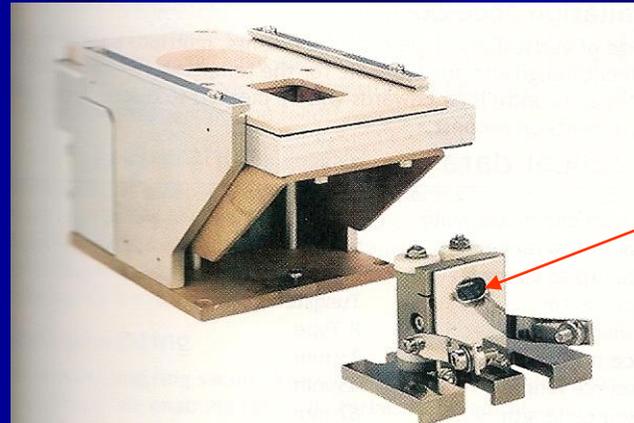
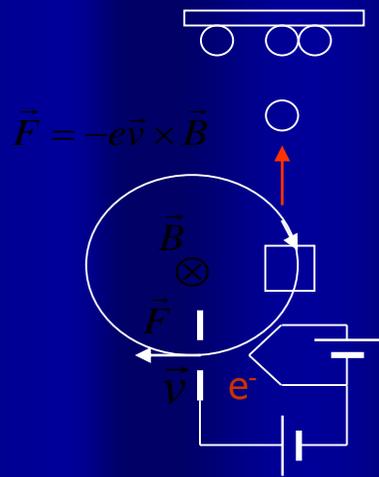
Filaments Boats



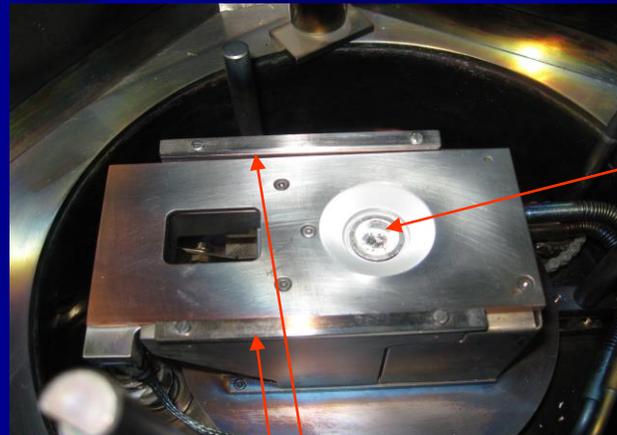
# Crucible Heaters



Filaments Boats



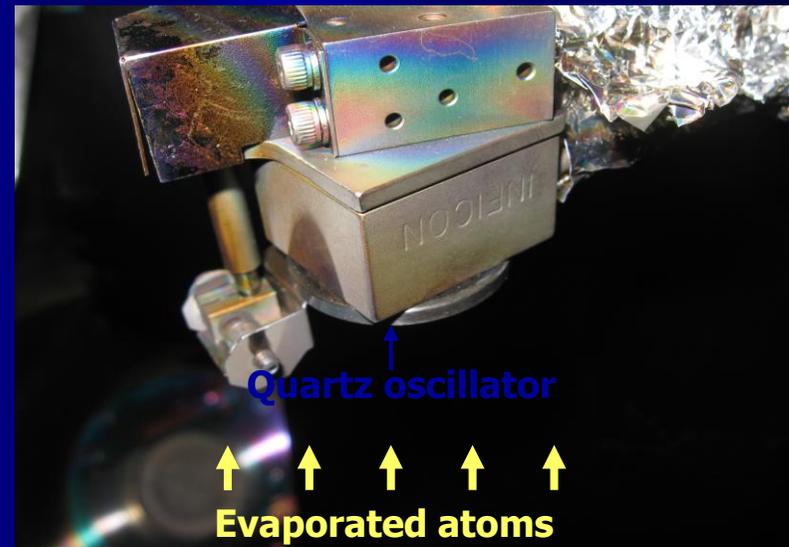
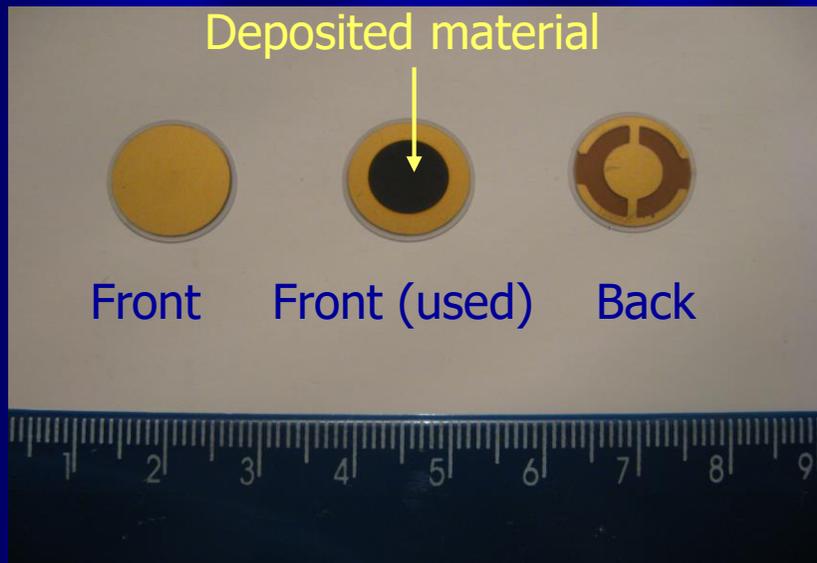
Filament (cathode)



Crucible

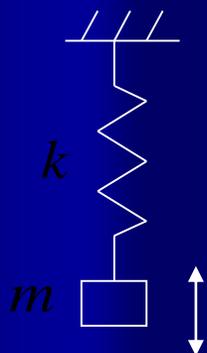
Magnet

## Electron-beam evaporator.



Quartz oscillators

Monitor head



$$f = \frac{1}{2\pi} \sqrt{\frac{k}{m}}$$

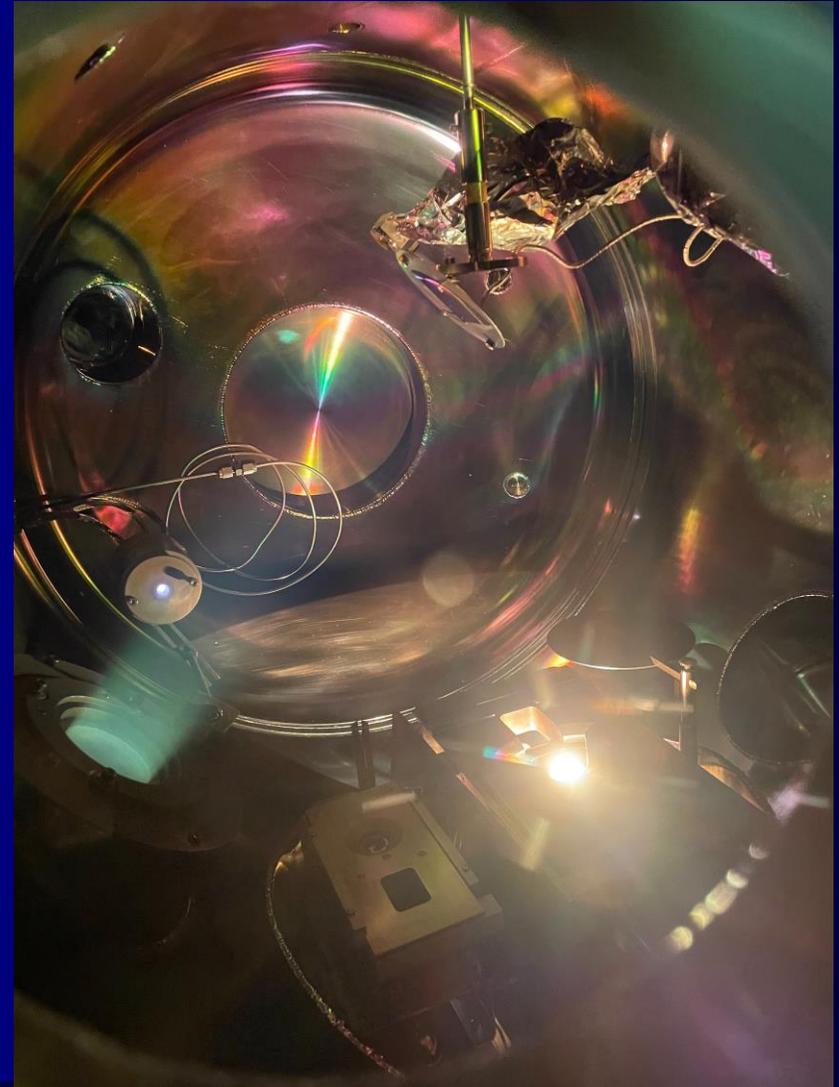
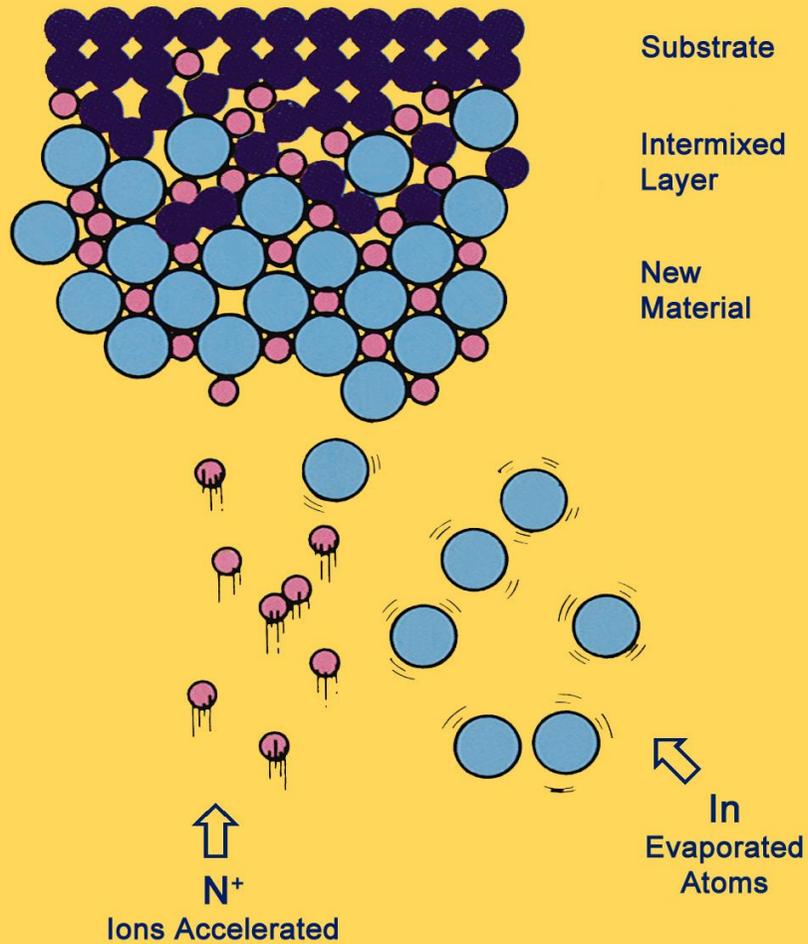
Quartz deposition rate monitor.

# What physics is in all this ?

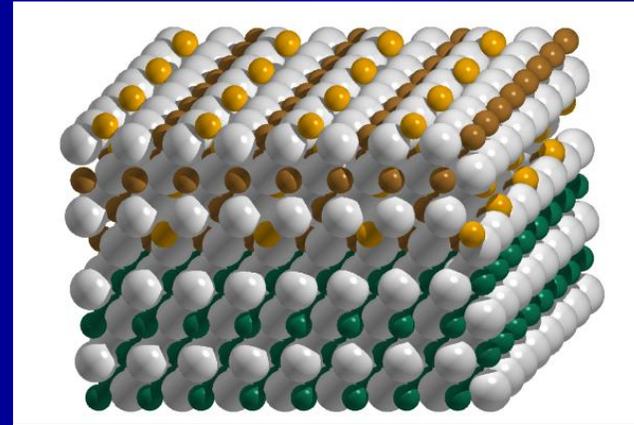
- thermodynamics and kinetics
  - phase transition - gas condenses to solid
  - nucleation
  - growth kinetics
  - activated processes
    - desorption
    - diffusion
  - allowed processes and allowed phases
- solid state physics
  - crystallography
  - defects
  - bonding
- electricity and magnetism
  - optics
  - conductivity - resistivity
  - magnetic properties
- mechanics
  - stresses in films
  - friction and wear

# IBAD Method

## Ion Beam Assisted Deposition



# Atomic Layer-by-Layer Growth

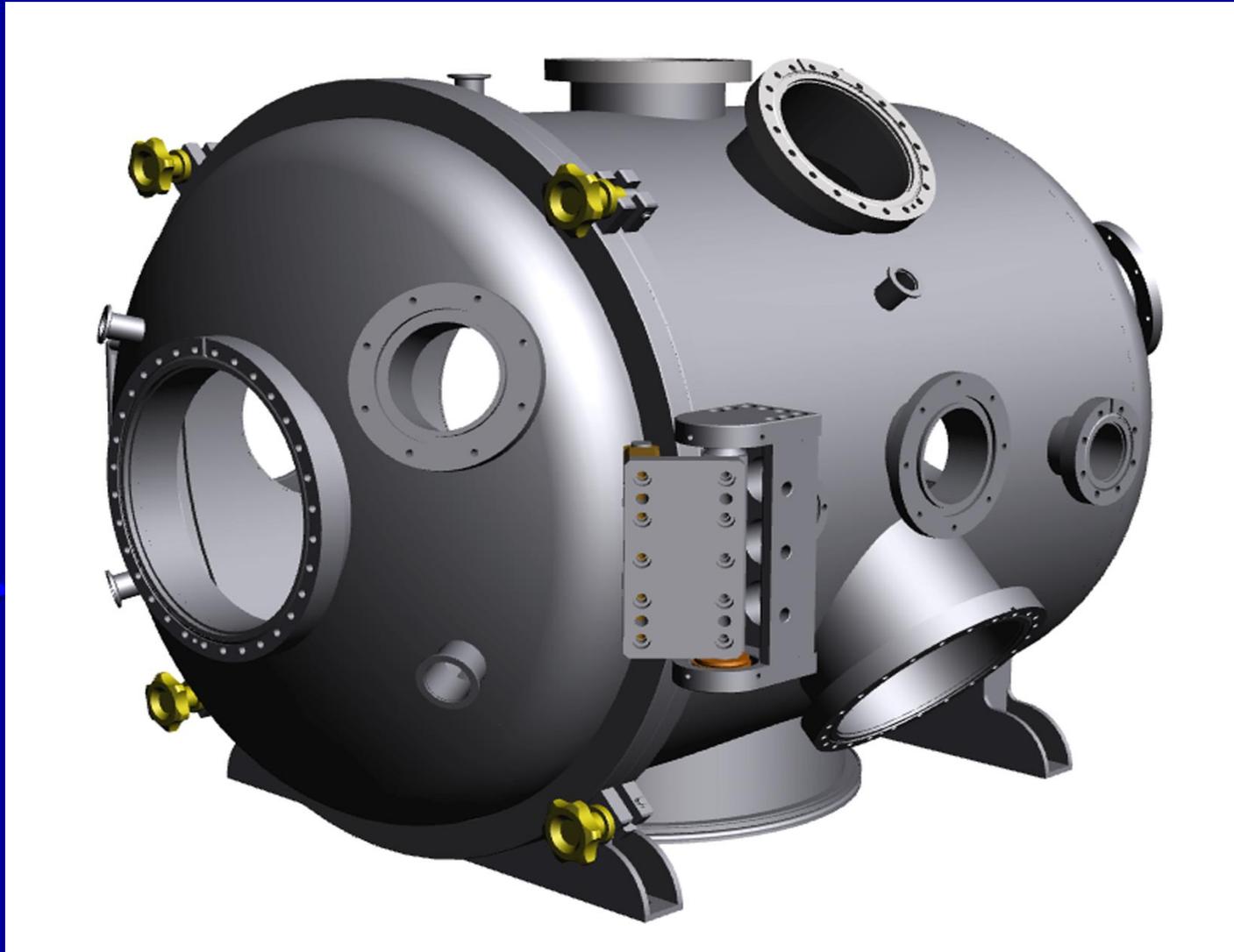


*Ultimate goal:*

*Epitaxial growth of perfect thin layers with atomic precision onto (selected parts of) a single crystalline substrate, in order to manipulate materials properties (or to design ultrathin devices).*

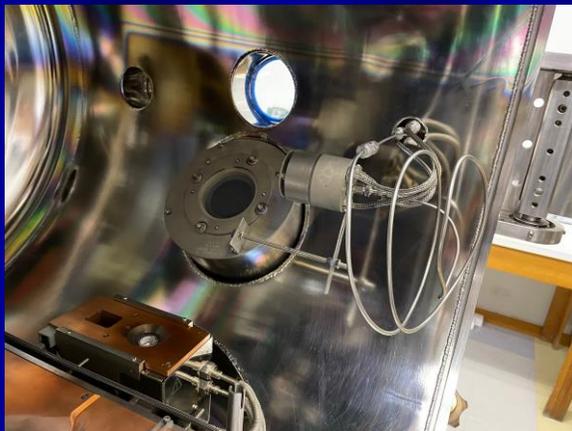
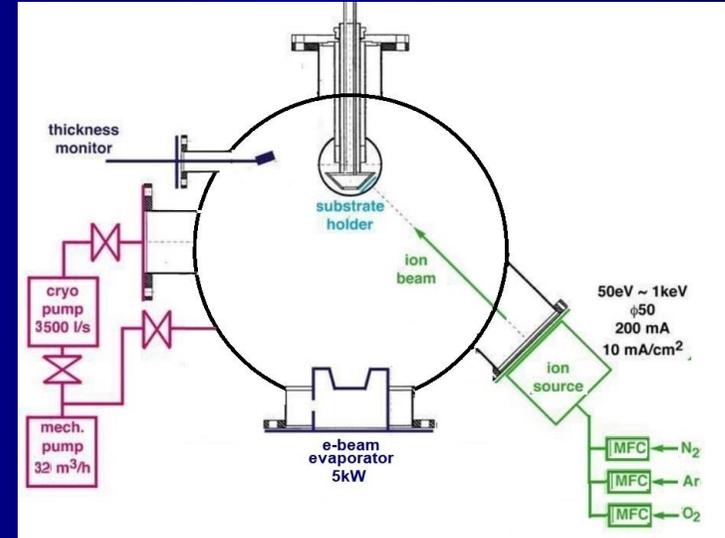
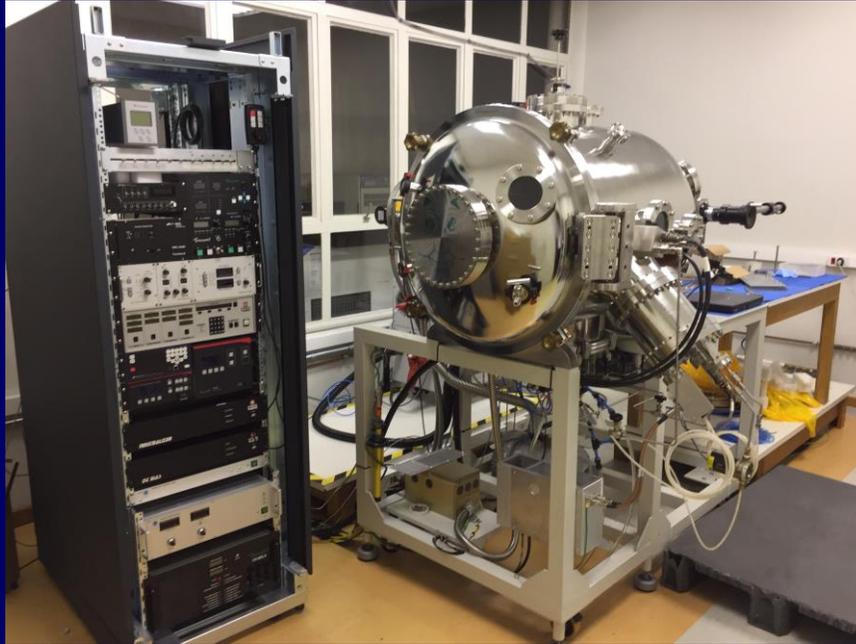
**New Vacuum Chamber – under final construction:  
diameter: @700 mm  
length: @1000 mm**

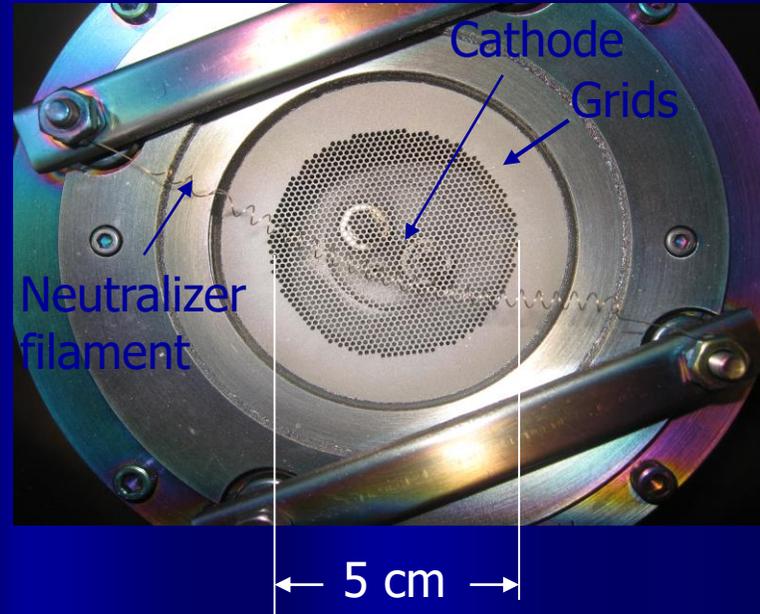
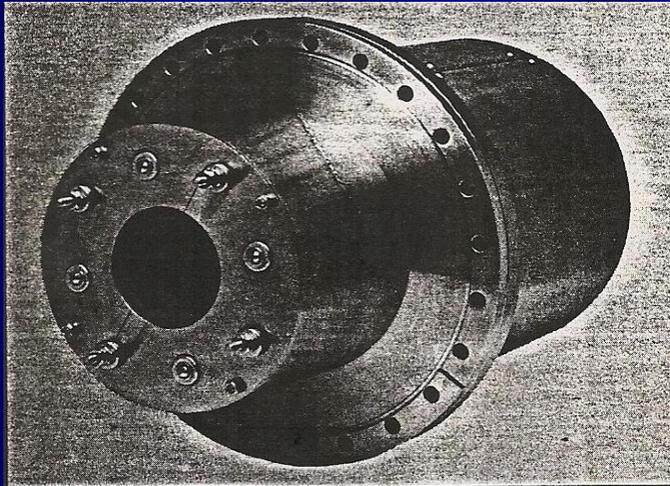
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# IBAD system: Details

Base Pressure =  $7.0 \times 10^{-5}$  Pa  
Working Pressure =  $2.0 \times 10^{-2}$  Pa

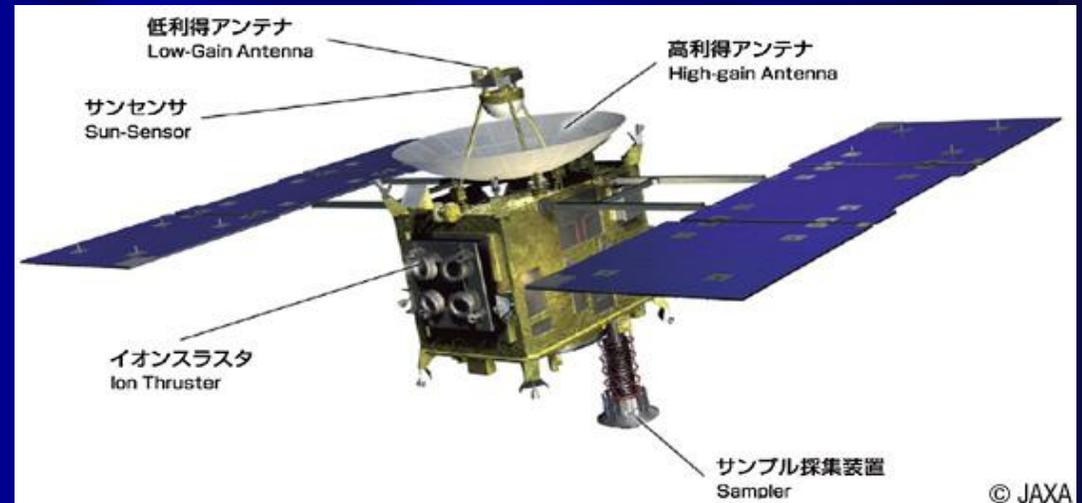
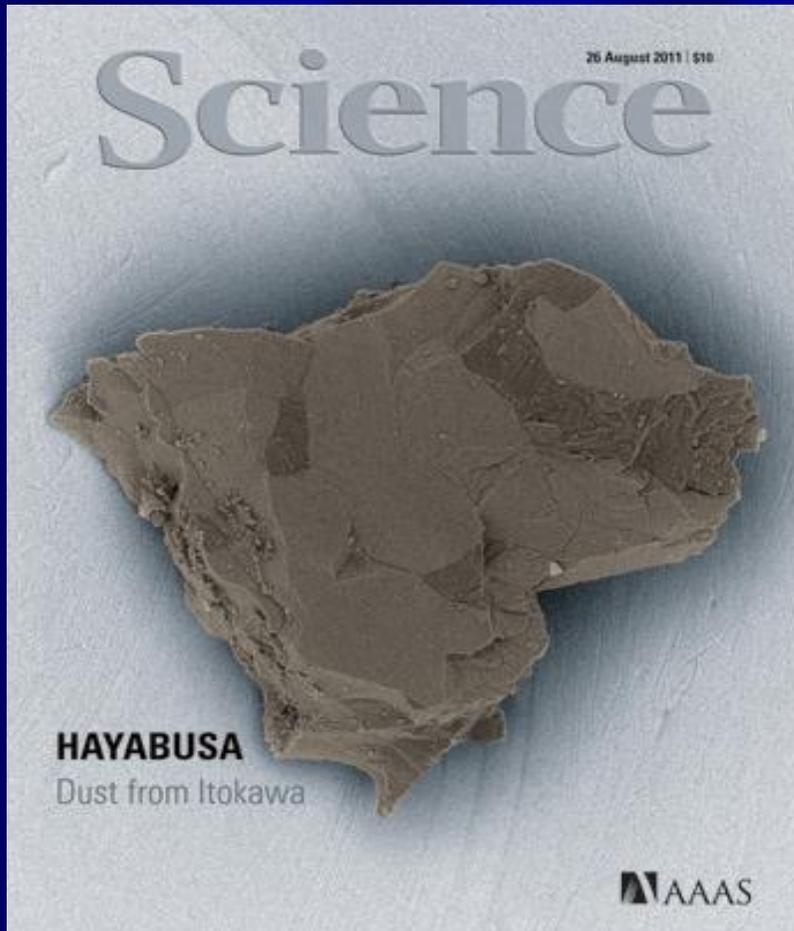




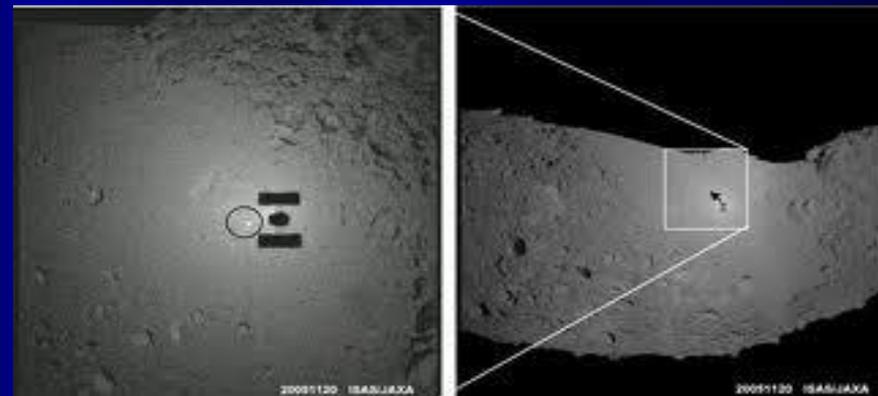
Front view

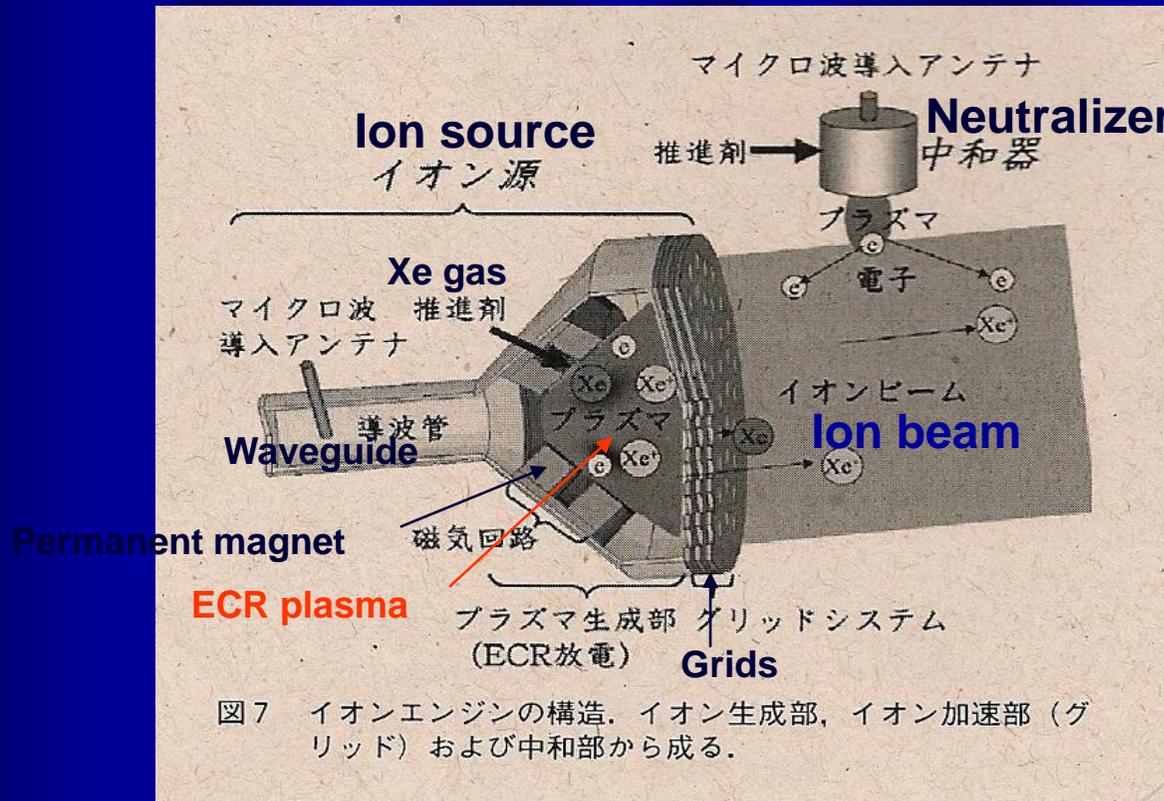
Fig. 3.5. Ion beam source.

# HAYABUSA



## Itokawa





Total weight = 510 kg  
 Weight of LXe = 66 kg  
 $E = 1.5 \text{ keV}$   
 $F = 4.2 \text{ GHz}$   
 $B = 0.15 \text{ T}$

$$F = \Delta p / \Delta t$$

$$E = p^2 / m$$

Rare gases:  
 He, Ne, Ar, Kr, Xe, Rn, Ku

Ion engine of the asteroid explorer HAYABUSA.  
 From: J. Plasma Fusion Res. **86**, 282 (2010).

# Analysis Systems



**Spectrophotometer  
Cary 7000**



**Spectrofluorimeter  
Horiba Fluorolog 3**



**EPR Spectrometer MiniScope  
MS5000 – Magnostec -  
Freiberg**

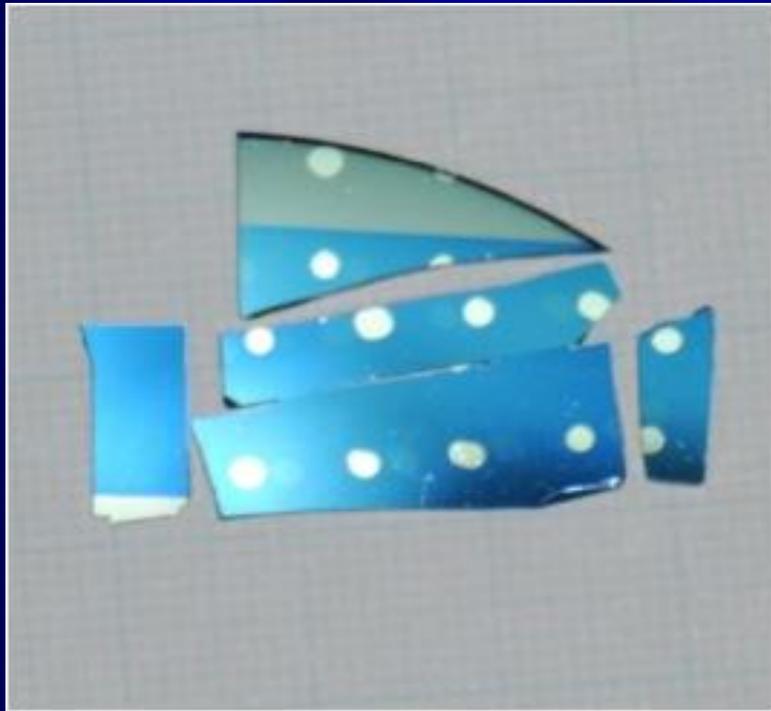
# OBJETIVOS

- **Deposição e caracterização de óxidos com alta constante dielétrica, tais como  $\text{HfO}_2$ ,  $\text{Al}_2\text{O}_3$ ,  $\text{Ga}_2\text{O}_3$ ,  $\text{GaGdO}$  sobre diferentes substratos por deposição assistida por feixe de íons (IBAD);**
- **Fabricar filmes até espessura de 500 nm de ferroelétricos como  $\text{Ba}_x\text{Sr}_{1-x}\text{TiO}_3$ ;**
- **Otimizar a deposição de filmes finos para deposição de camadas com alta qualidade para a fabricação de dispositivos.**

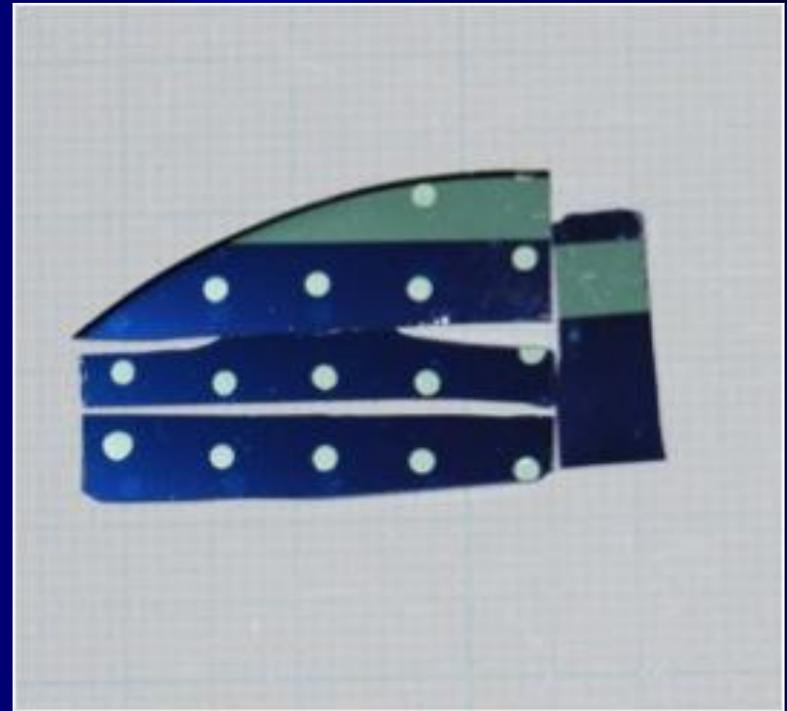
# Materials, Deposition Conditions & Analysis Methods

- Al 99.99% ingots
- O<sub>2</sub> 99.999%
- Substrates:
  - Si(100)
  - A-plane sapphire
  - C-plane sapphire
  - R-plane sapphire
  - GaN/A-plane sapphire
- Ion beam energy:
  - 100 eV
- Film thickness (Al):
  - 500 Angstrom
- Arrival rate ratio ARR(O/Al):
  - 0.8 and 5.5
- Substrate temperature:
  - Room Temp. (R.T.), 100, 200, 300, and 450°C
- SEM (JEOL,2010 )
- Surface Profilometer (Slon, Dektak 3030)
- XRD (Rigaku, Rint 2100)
- XPS (Kratos, XSAM HS,  $h\nu = 1486.6$  eV)
- Raman (Renishaw, Raman 2000)

# $\text{Al}_2\text{O}_3$ films



$\text{Al}_2\text{O}_3$  film produced with 300 eV ion beam energy and 100 nm of thickness on Si(100).



$\text{Al}_2\text{O}_3$  film produced with 800 eV ion beam energy and 100 nm of thickness on Si(100).

# Optical Absorption %R

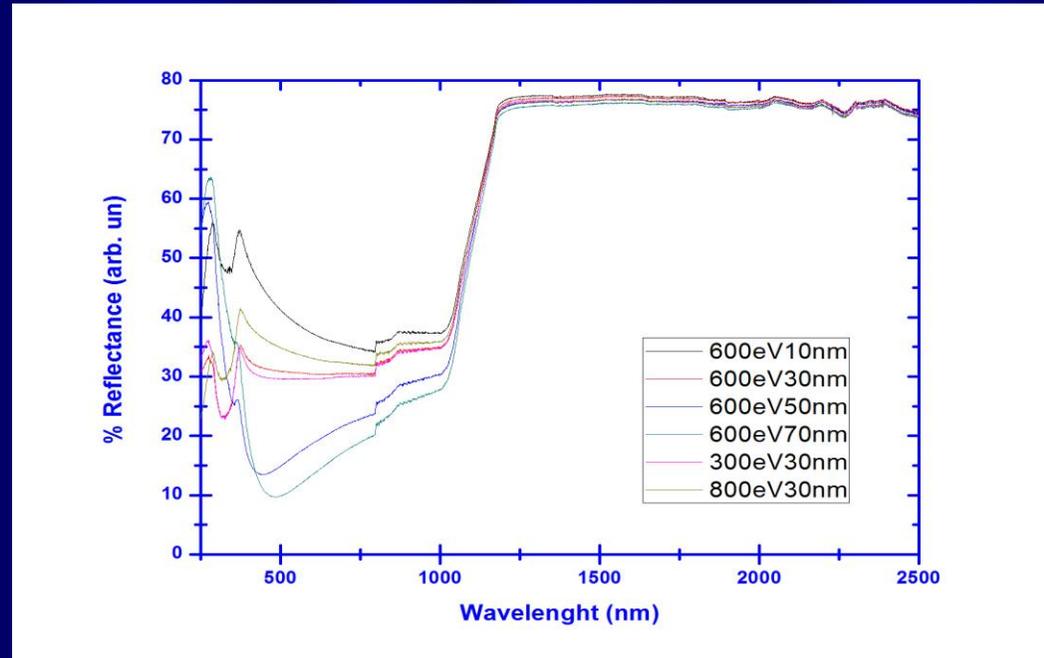


Figure 3: %R spectra from the aluminum oxide film prepared on silicon substrate at an ion beam energy of 800 eV.

# %Reflectance

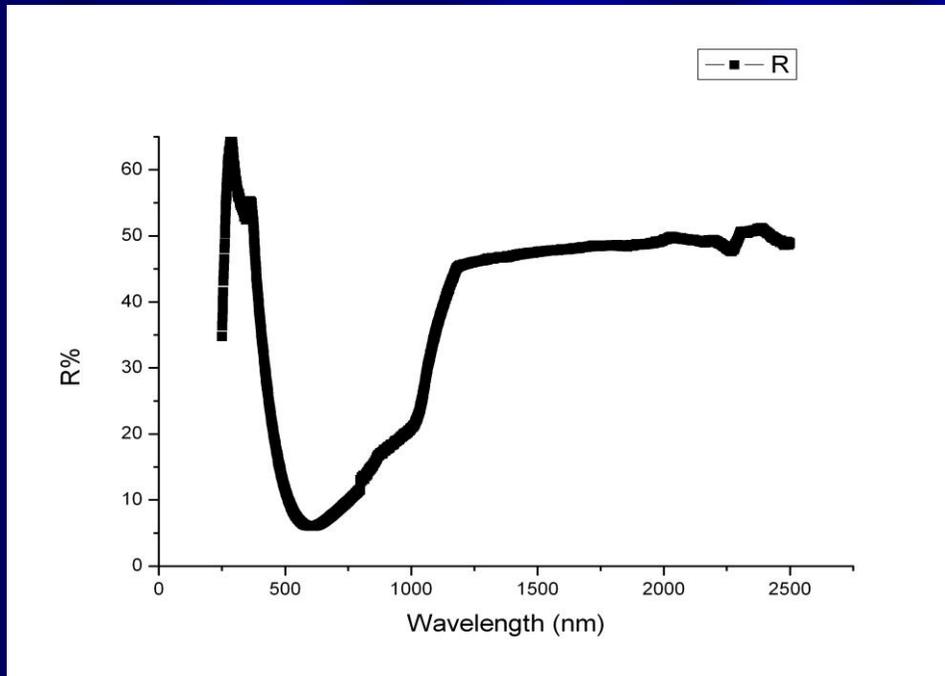


Figure 1: %R spectra from the aluminum oxide film prepared on silicon substrate at an ion beam energy of 300 eV.

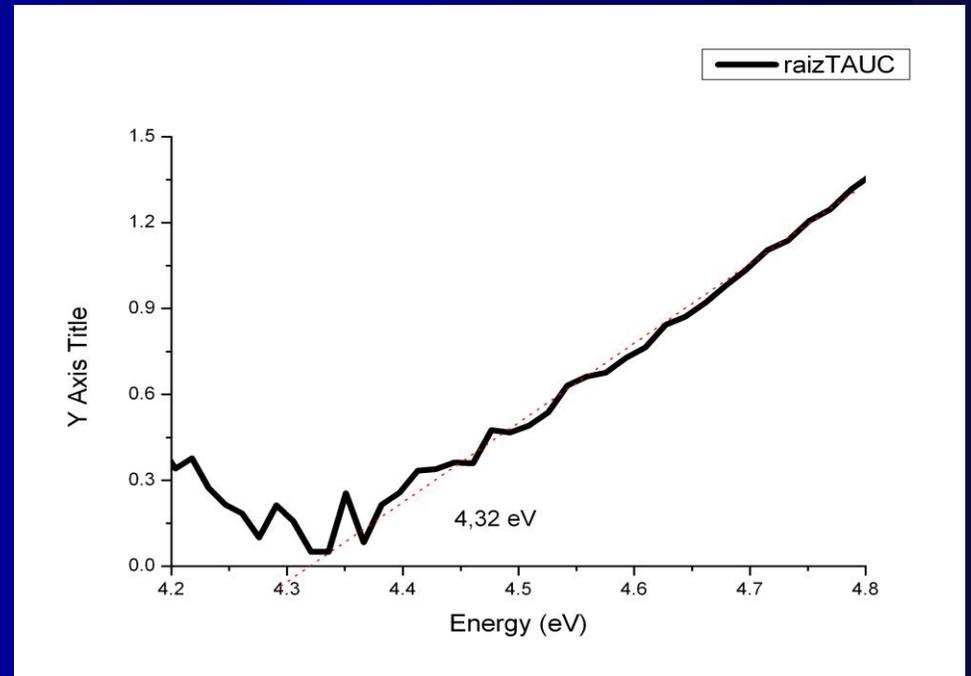


Figure 2: %R spectra from the aluminum oxide film prepared on silicon substrate at an ion beam energy of 300 eV. The X and Y data was converted into energy and Tauc equation to estimate the optical band gap.

# %Reflectance

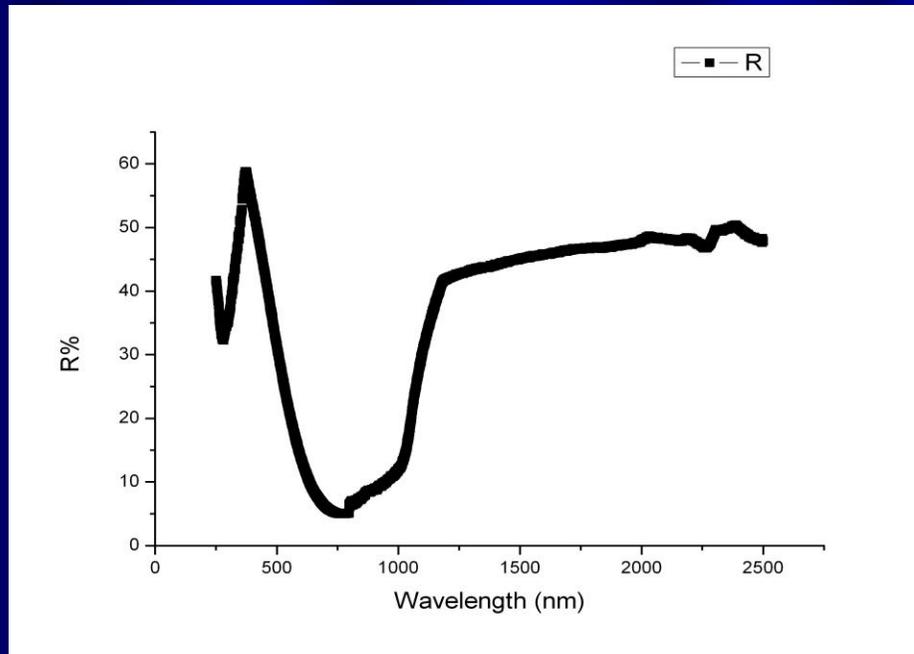


Figure 3: %R spectra from the aluminum oxide film prepared on silicon substrate at an ion beam energy of 800 eV.

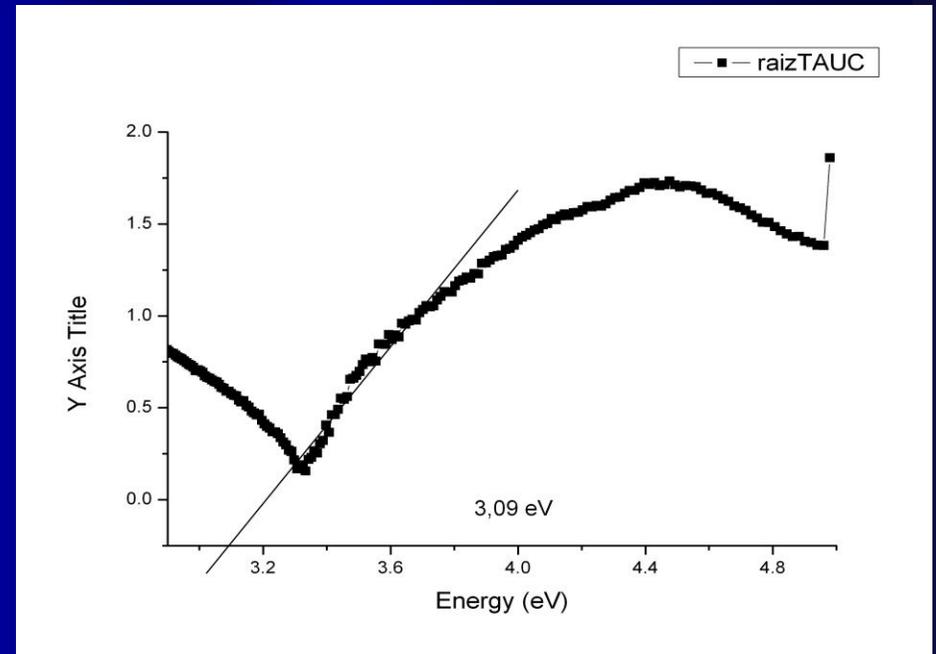
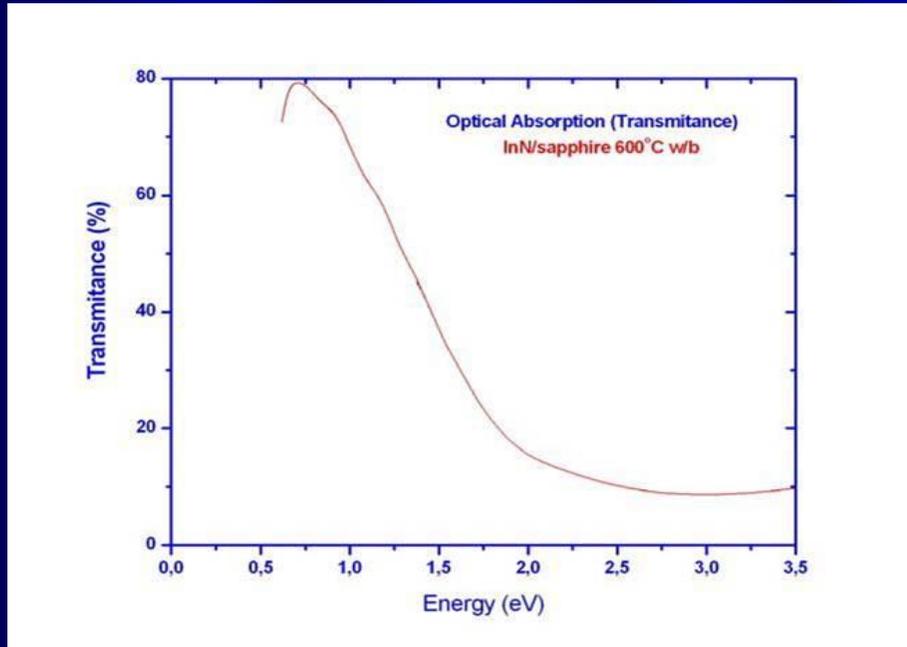
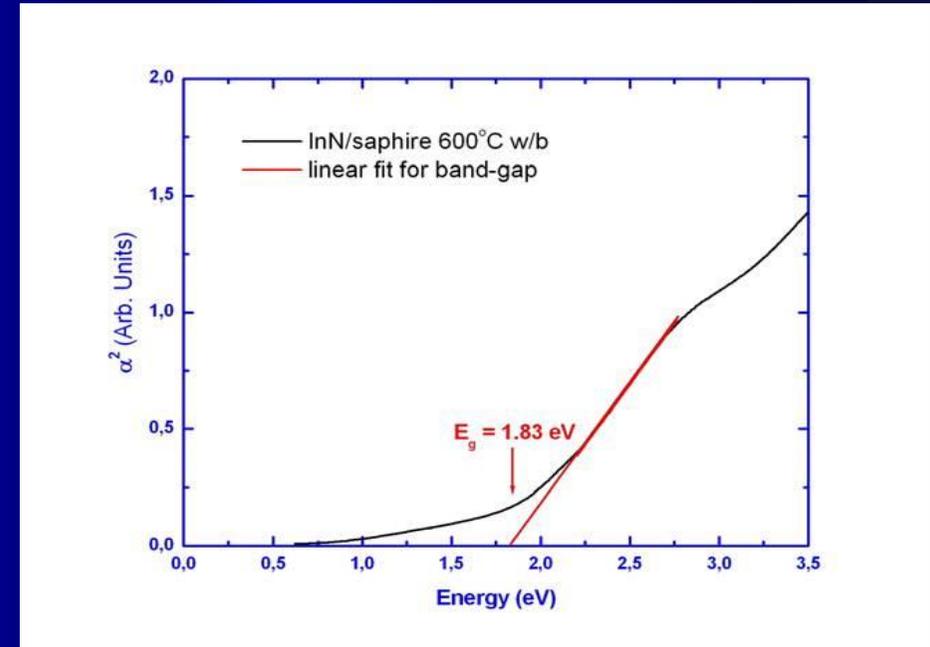


Figure 4: %R spectra from the aluminum oxide film prepared on silicon substrate at an ion beam energy of 800 eV. The X and Y data was converted into energy and Tauc equation to estimate the optical band gap.

# Optical Absorption



**Transmission spectrum of InN/sapphire film produced at 600°C, E = 0.8 keV and ARR(N/In) = 5.3**



**Band-gap determination of InN/sapphire film produced at 600°C, E = 0.8 keV and ARR(N/In) = 5.3. Assuming that InN has a direct transition,  $\alpha^2 = b(E - E_g)$ .**

# CxV

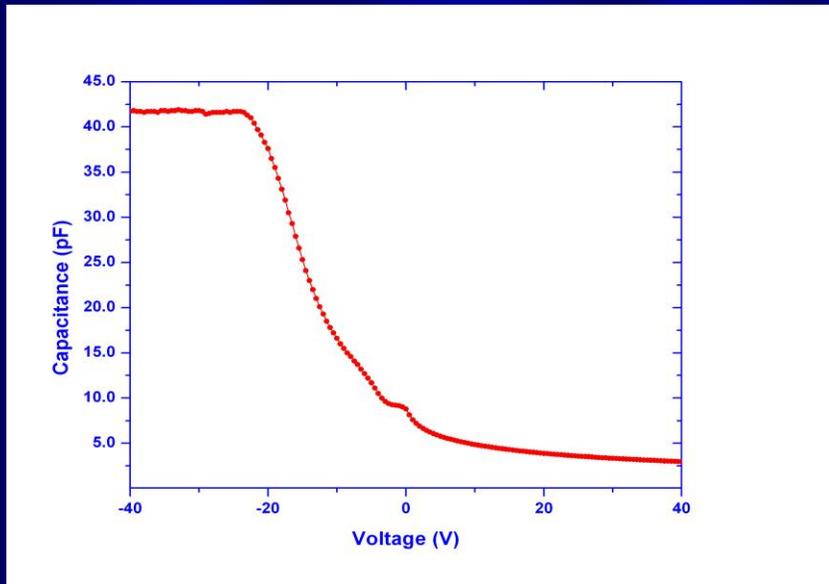


Figure 5: CxV measurement of the aluminum oxide film prepared on silicon substrate at an ion beam energy of 300 eV.

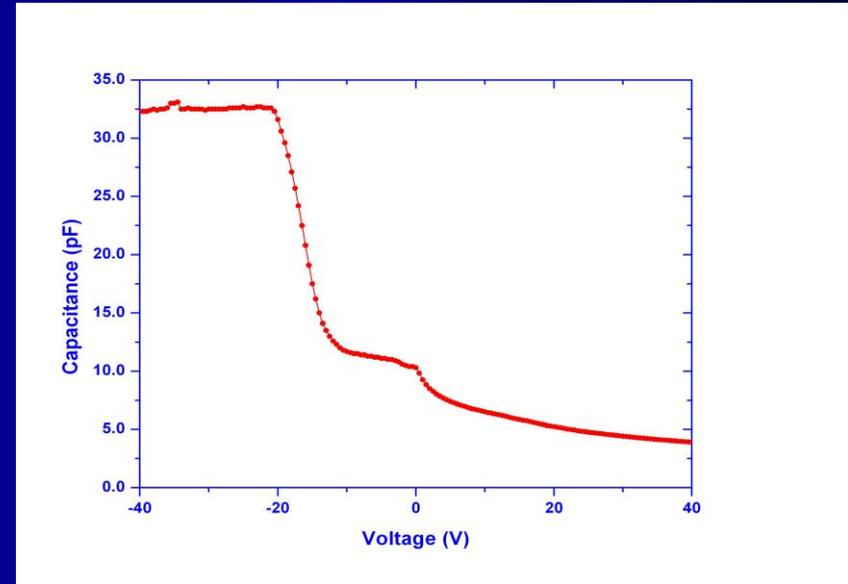
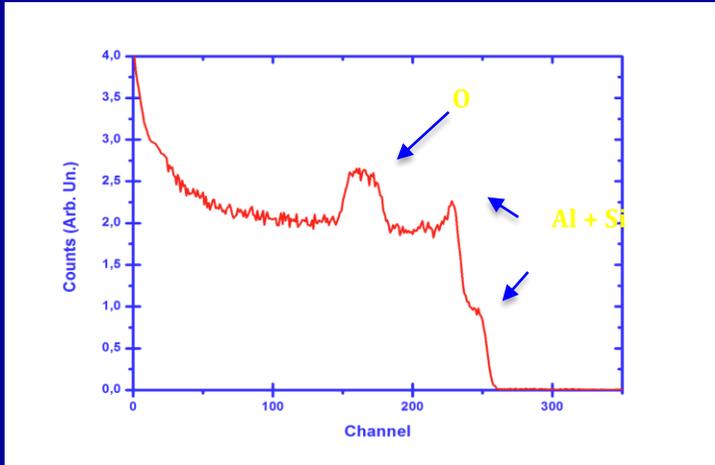


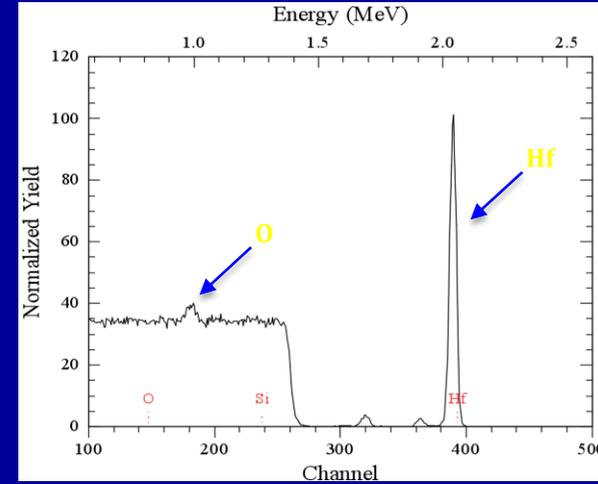
Figure 6: CxV measurement of the aluminum oxide film prepared on silicon substrate at an ion beam energy of 800 eV.

$\text{Al}_2\text{O}_3$  - XPS Composition Ratio O/Al = 1.87  
 $\text{Al}_2\text{O}_3$  - RBS Composition Ratio O/Al = 1.85

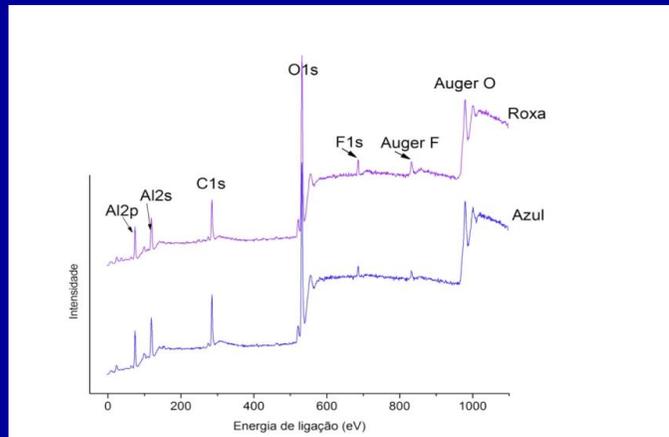
$\text{HfO}_2$  – RBS data under analysis



RBS Spectra of  $\text{Al}_2\text{O}_3$

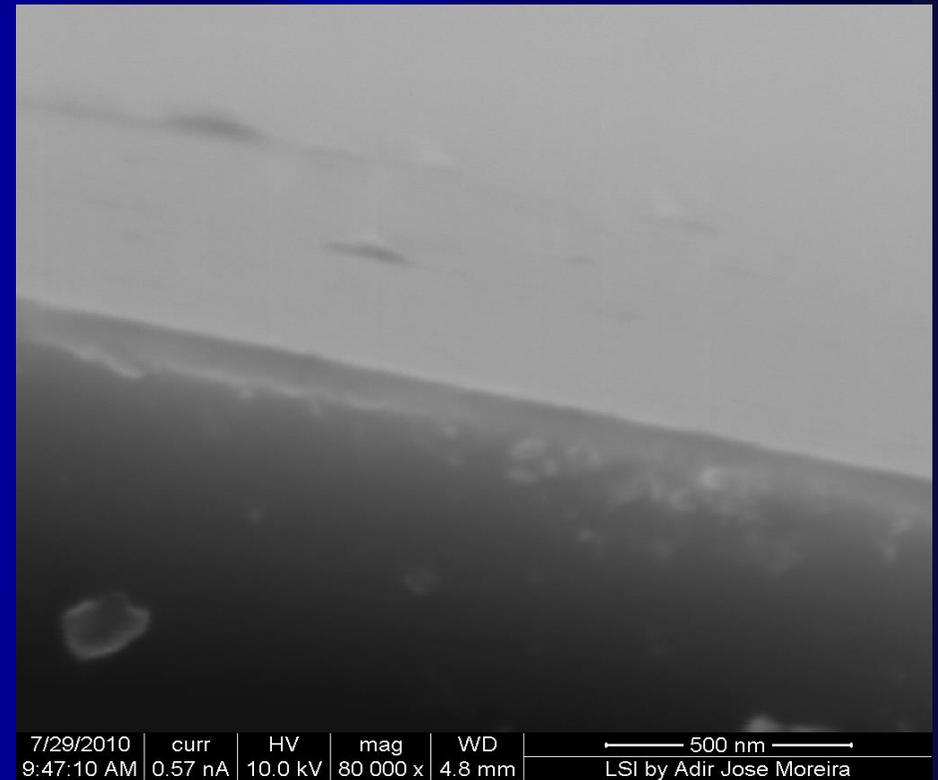
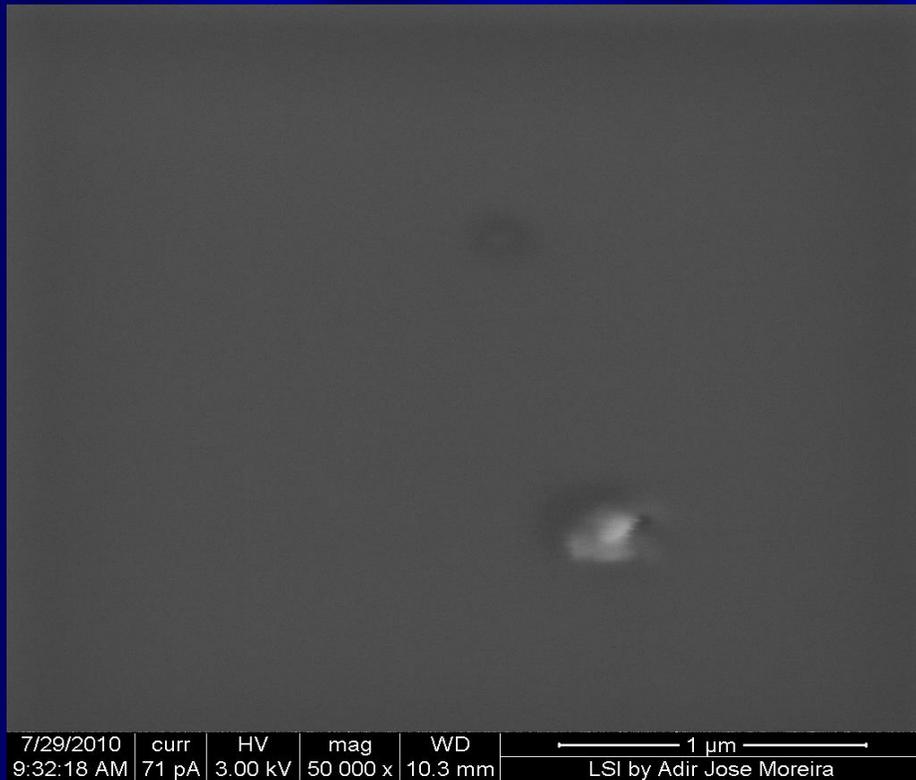


RBS Spectra of  $\text{HfO}_2$

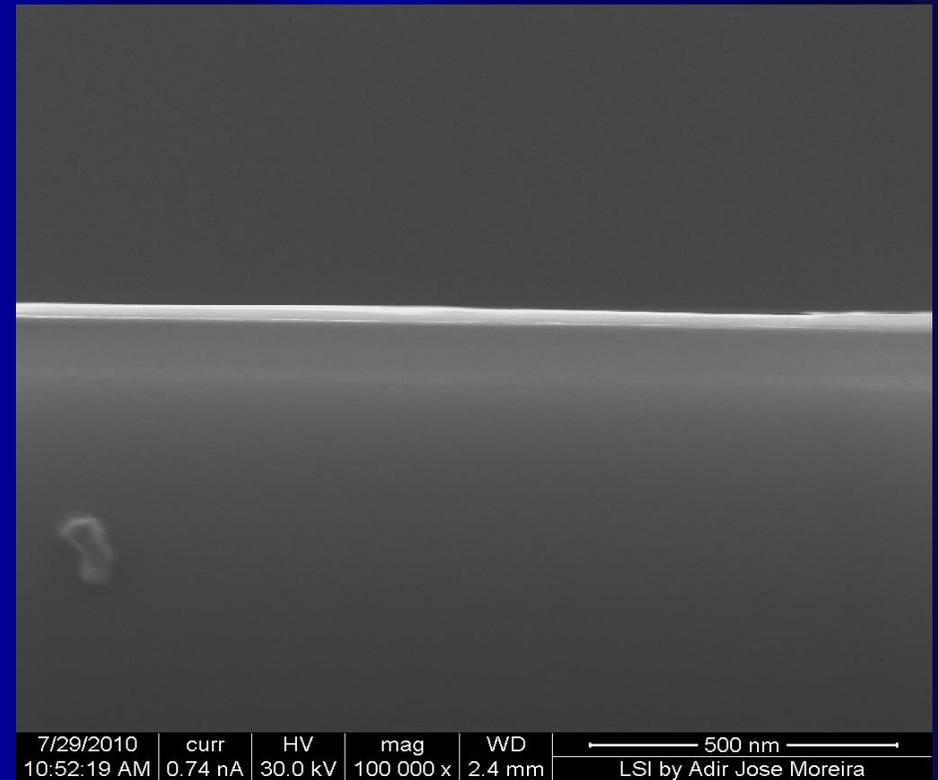
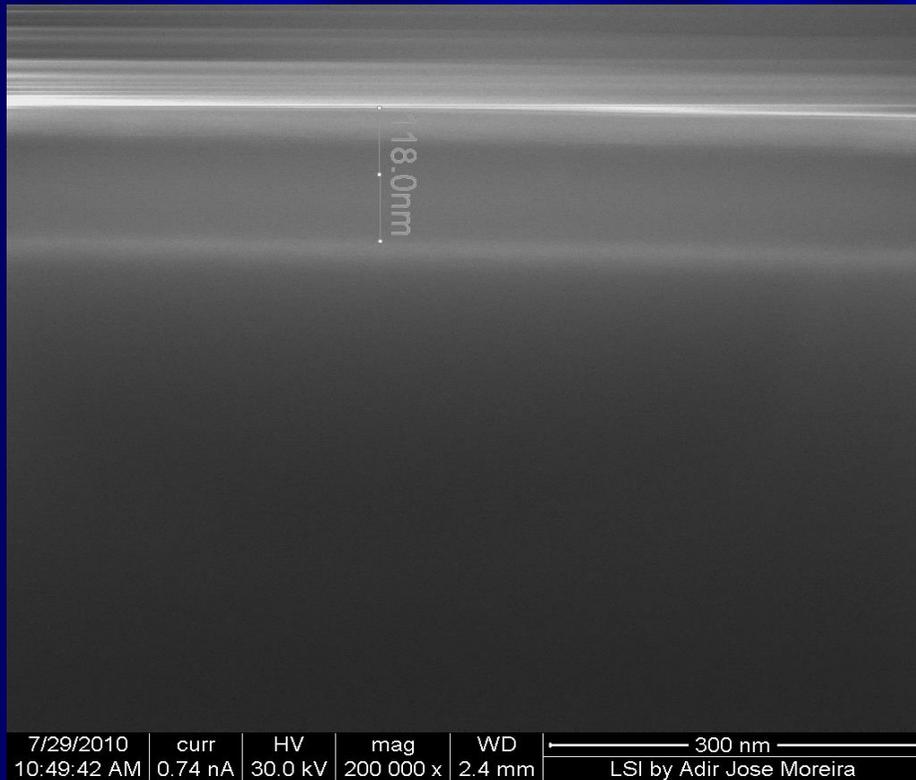


XPS Spectra of  $\text{Al}_2\text{O}_3$

# SEM Images – sample prepared at 300 eV

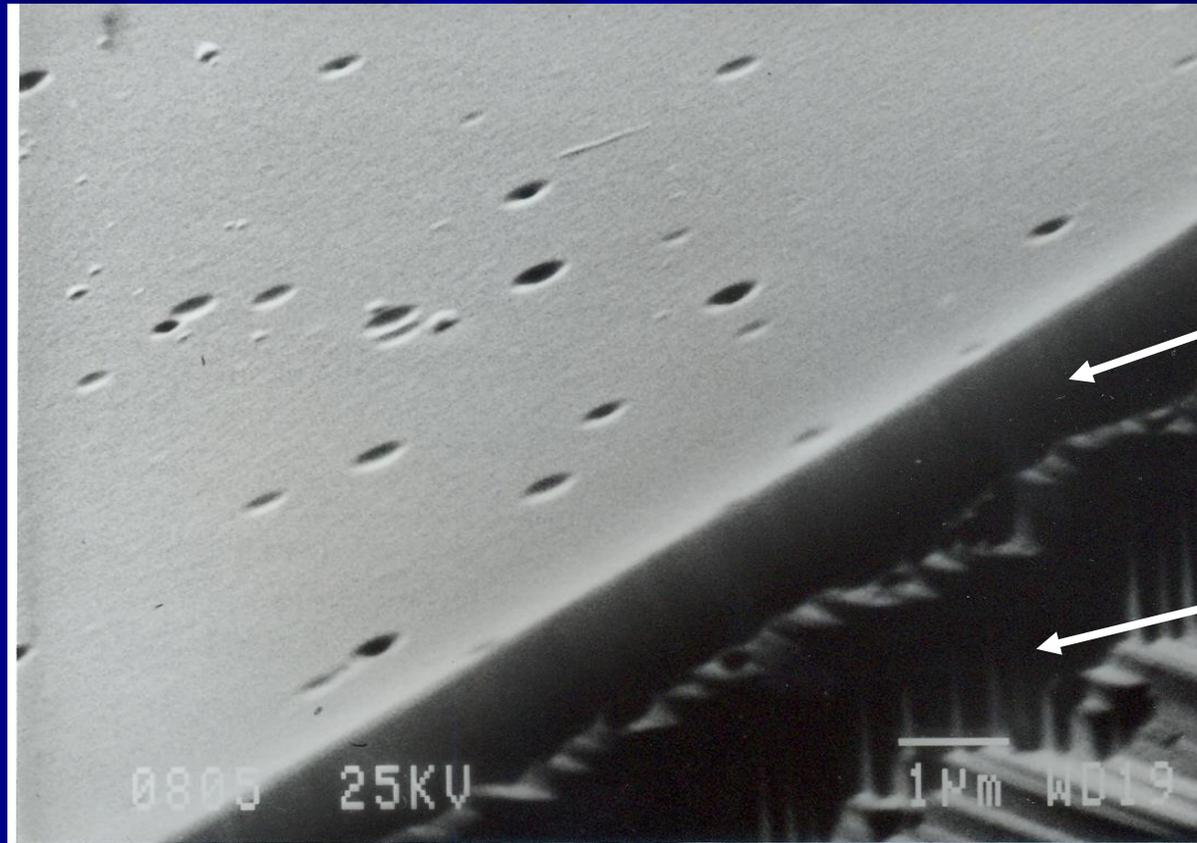


# SEM –sample prepared at 800 V



# Cross-sectional SEM Microphotograph

Etch pits



GaN

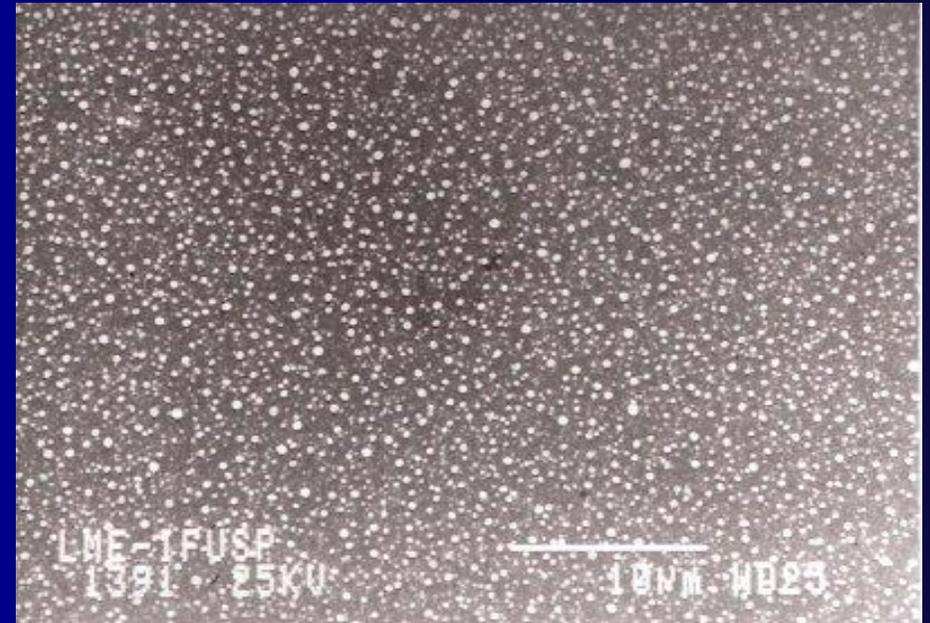
Sapphire

**GaN/A-plane sapphire - 500 eV**

# SEM Microphotographs



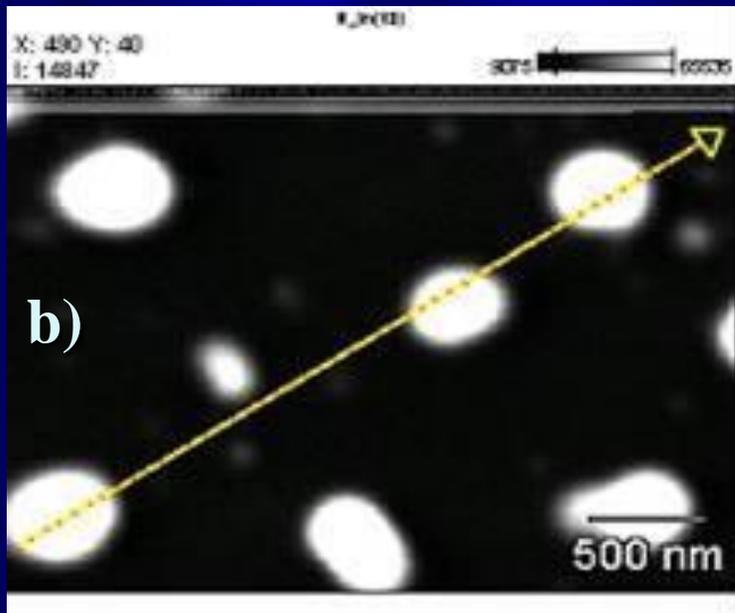
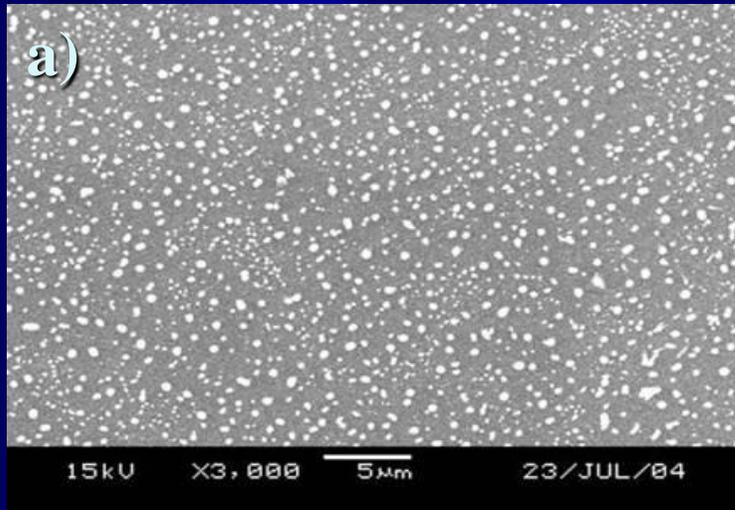
**Glass - R.T.**



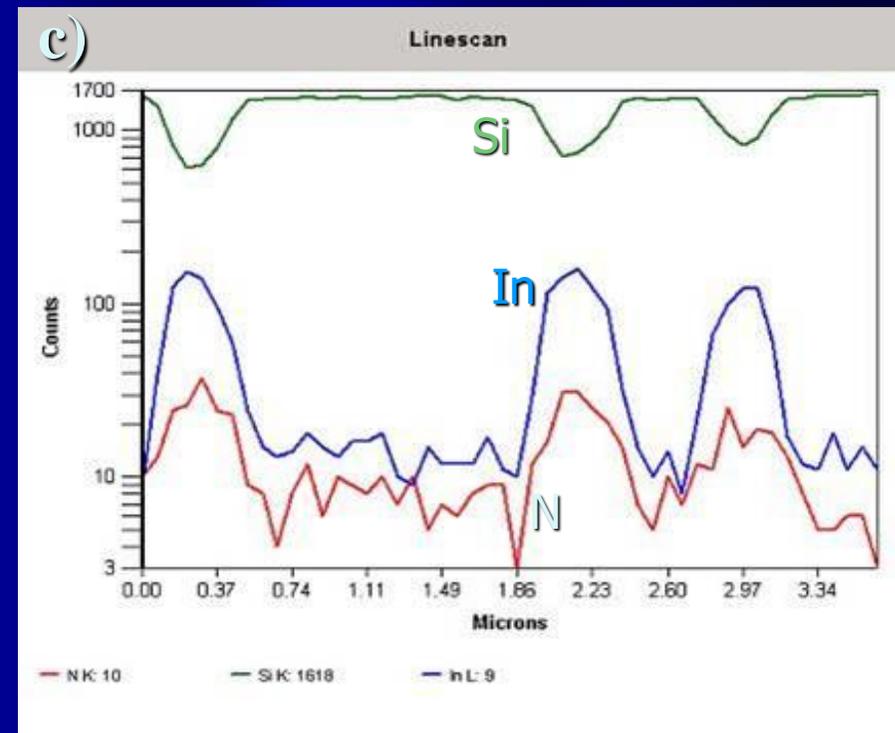
**In droplets**

**Glass - 200°C**

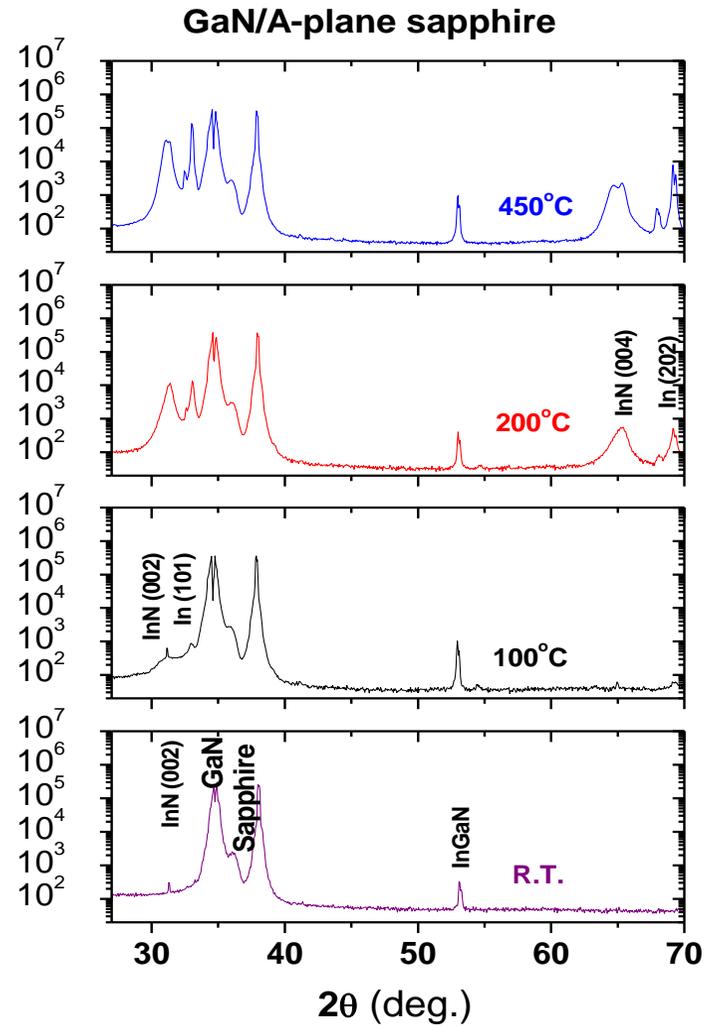
# SEM images



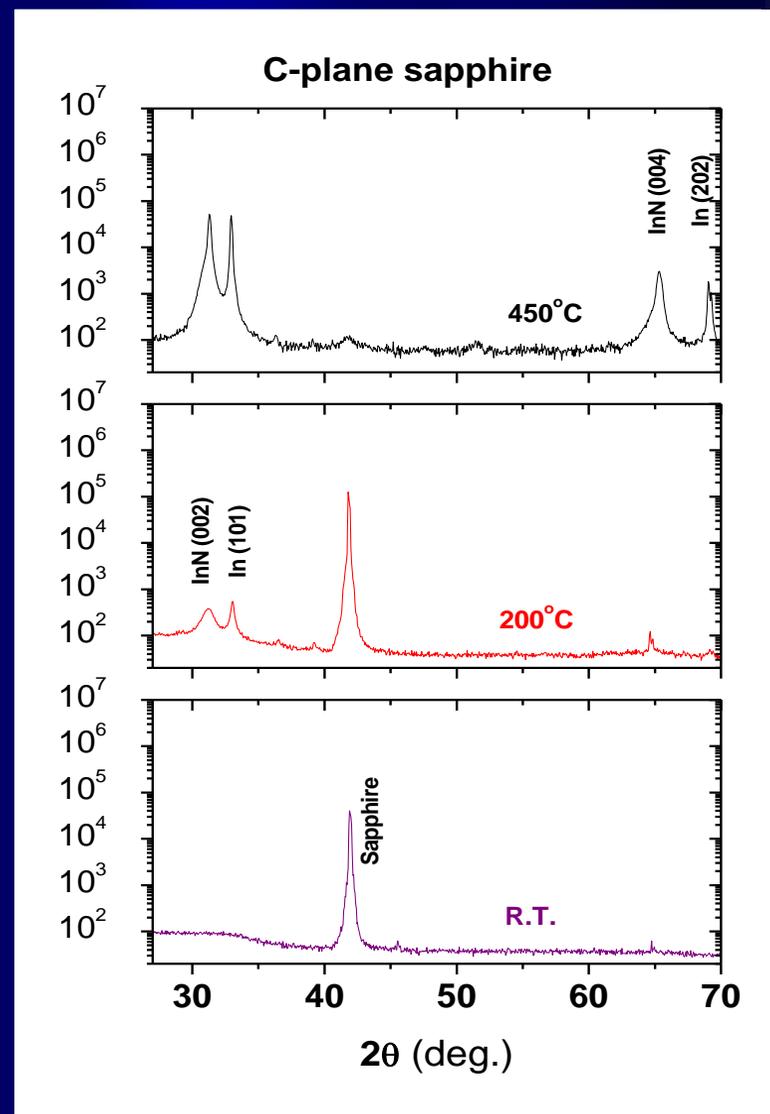
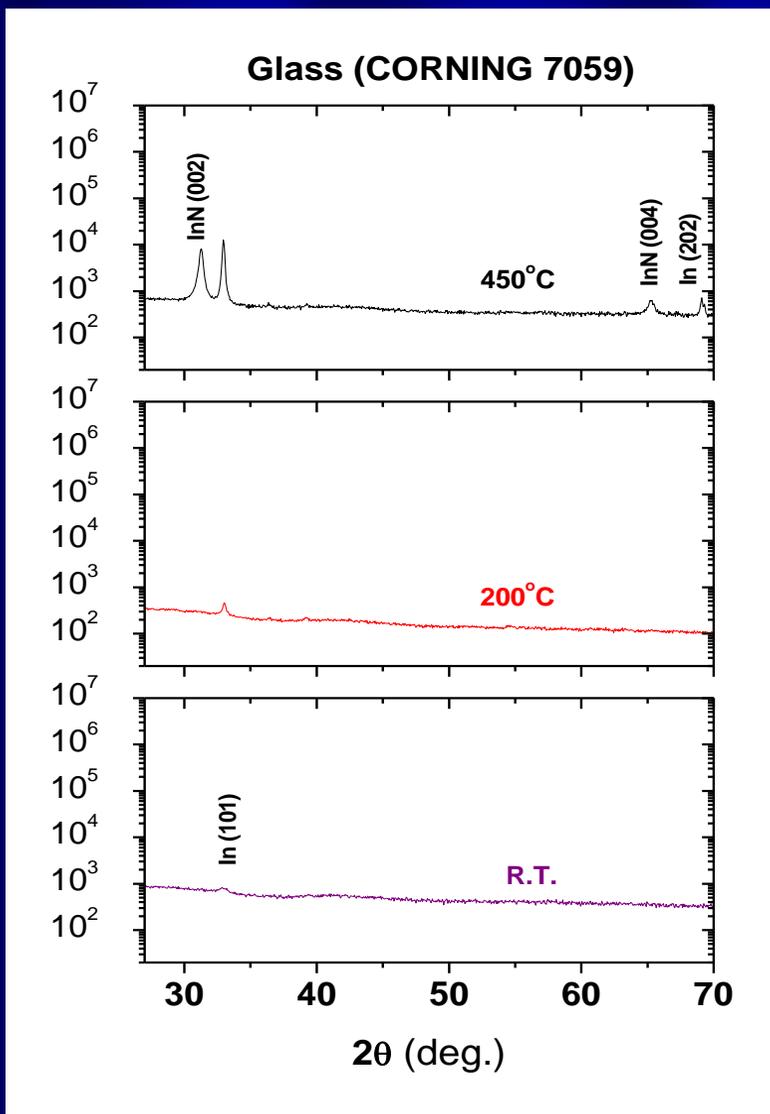
- a) SEM micrograph of an InN/Si(111) film produced at  $E = 0.8$  keV,  $ARR(N/In) = 5.3$ ,  $T = 520^\circ\text{C}$  with a magnification of 3000x.
- b) SEM micrograph of the same film with a magnification of 35000x.
- c) X-ray elemental profile through the line marked on figure b).



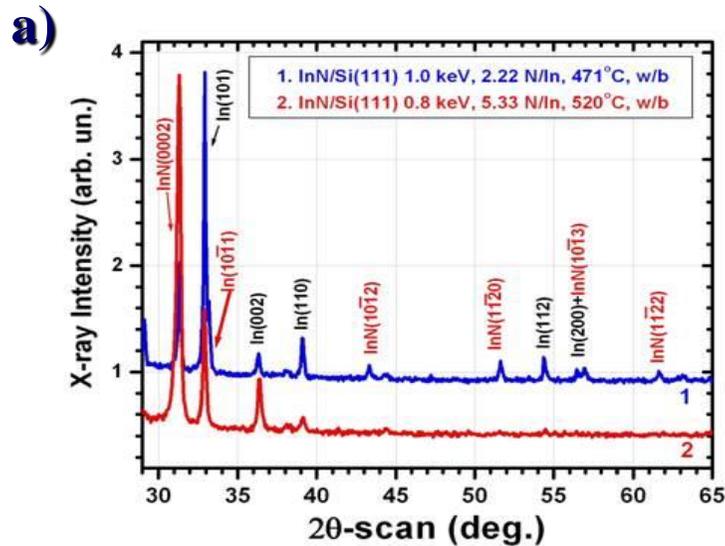
# XRD



# XRD



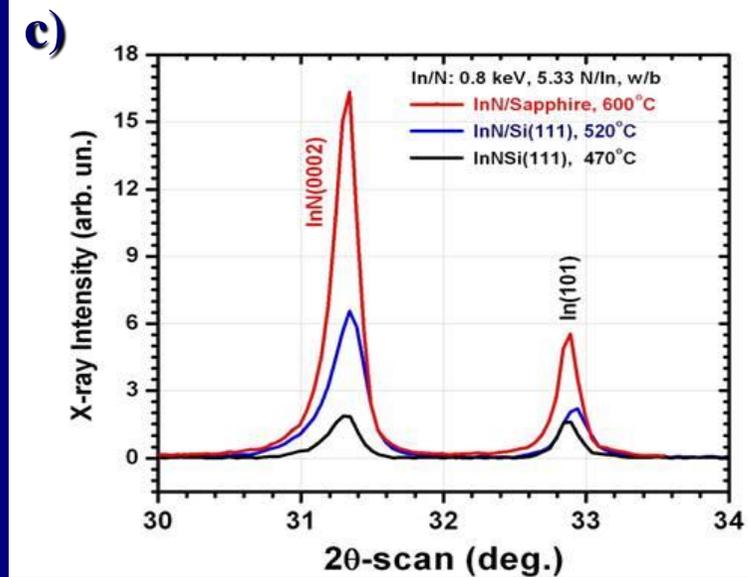
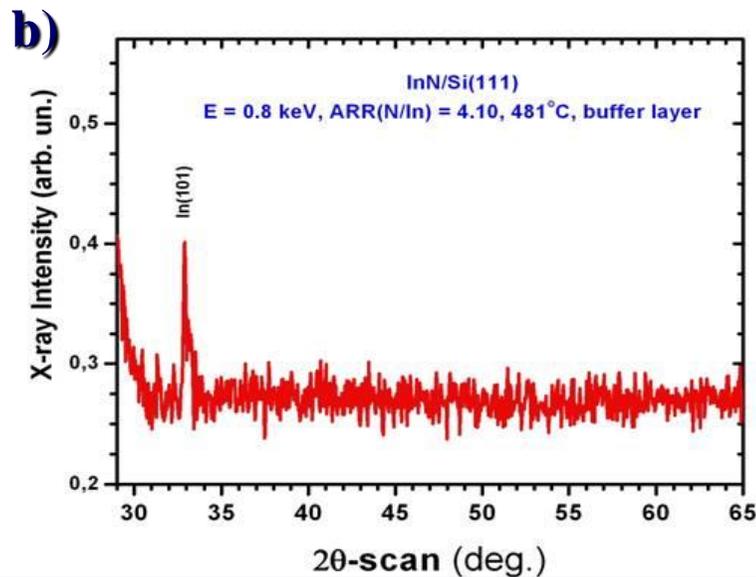
# XRD



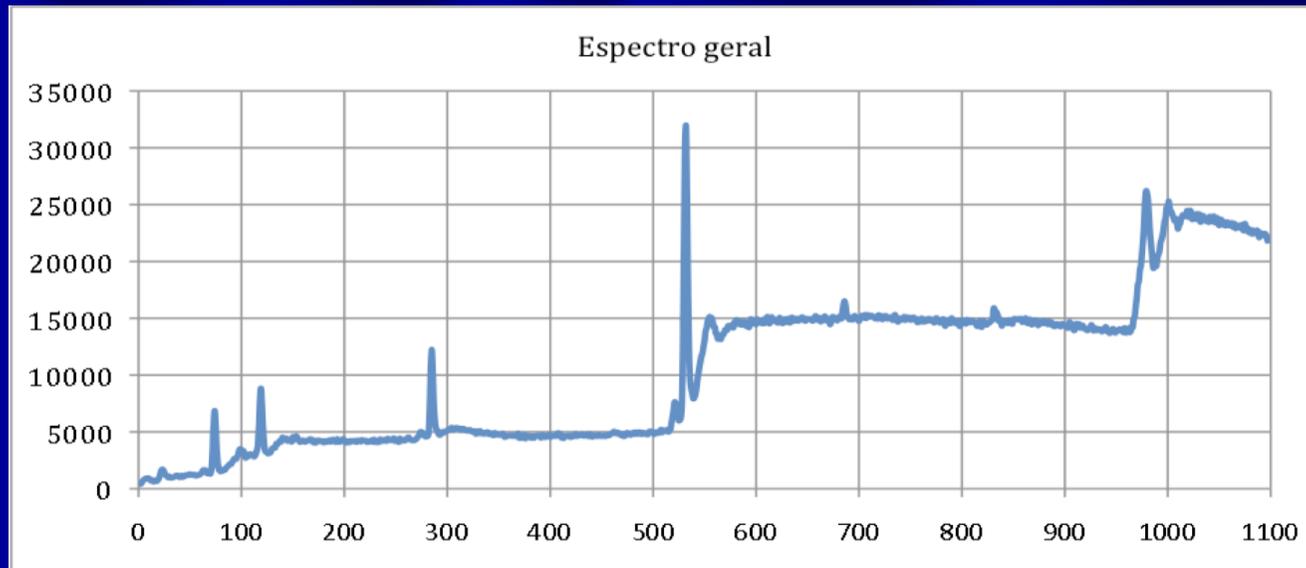
a) XRD patterns of InN/Si(111) without any buffer layer.

b) XRD pattern of InN/Si(111) with a buffer layer.

c) Dependence of the XRD pattern on temperature and substrate.

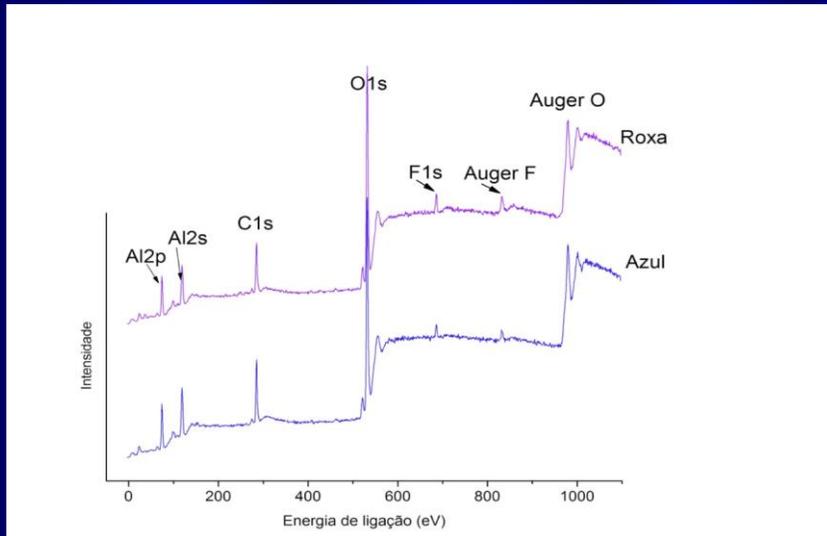


# XPS spectra

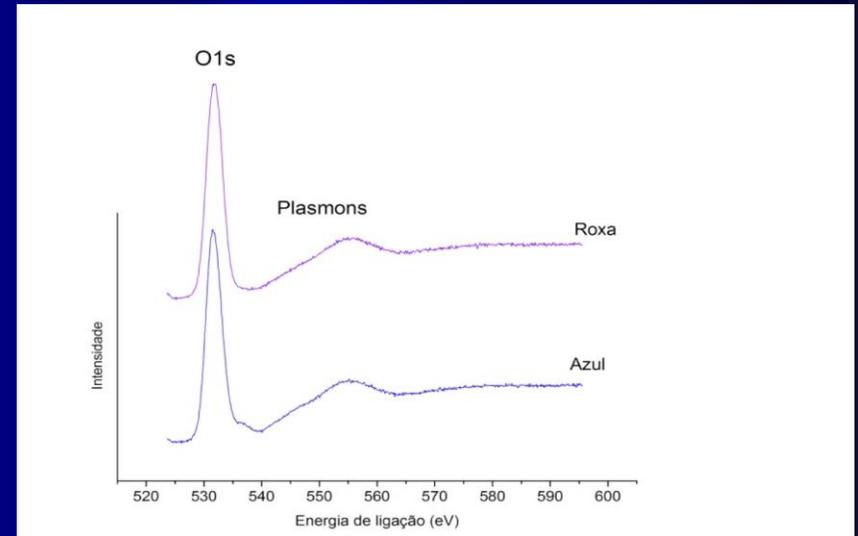


**Figure 3: %R spectra from the aluminum oxide film prepared on silicon substrate at an ion beam energy of 800 eV.**

# XPS



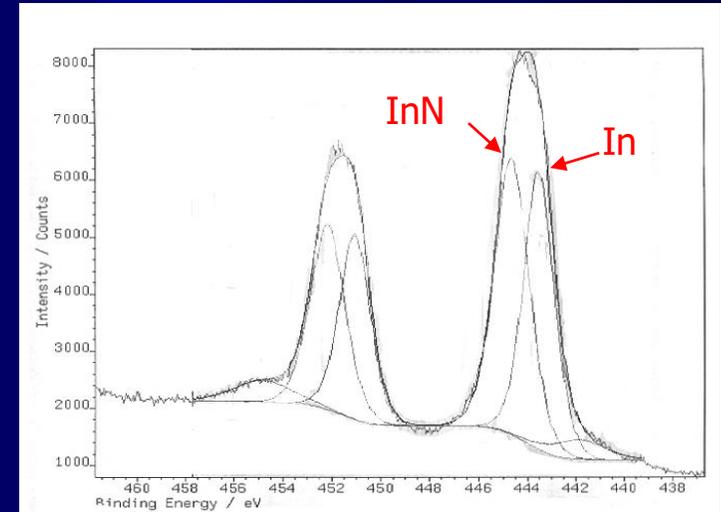
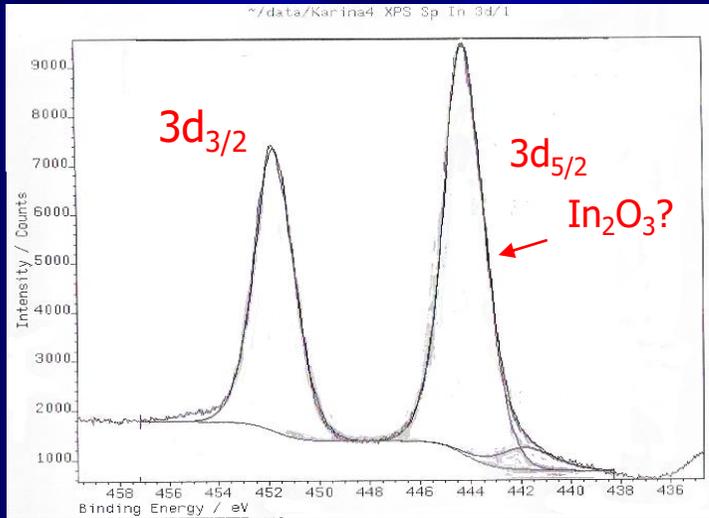
**Fig. 3:** Blue and Purple samples XPS Wide spectra (0-1100 eV)



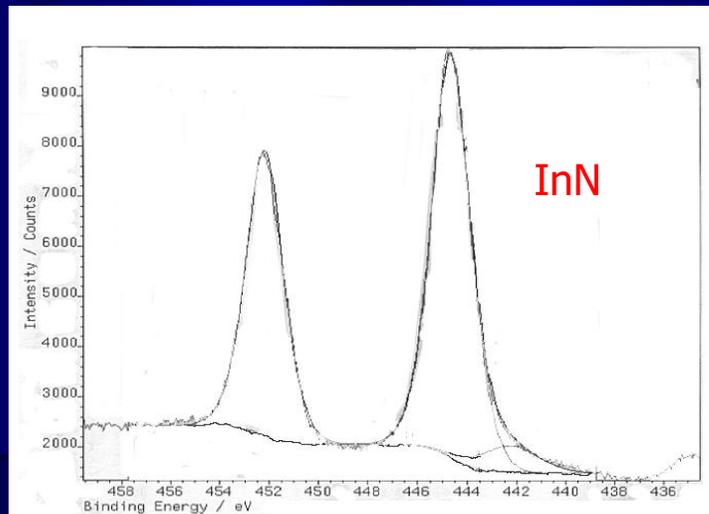
O1s spectra with plasmons.

# XPS - In 3d with surface cleaning by Ar ion etching

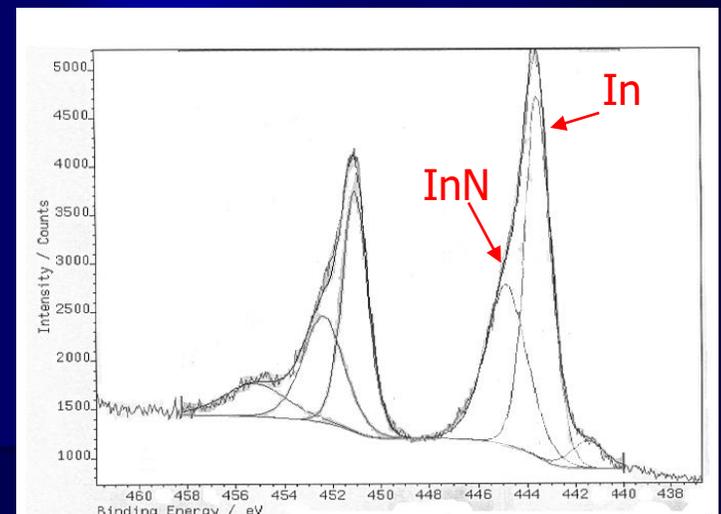
C-plane sapphire - R.T.      Glass - R.T.



## GaN/A-plane sapphire - R.T.

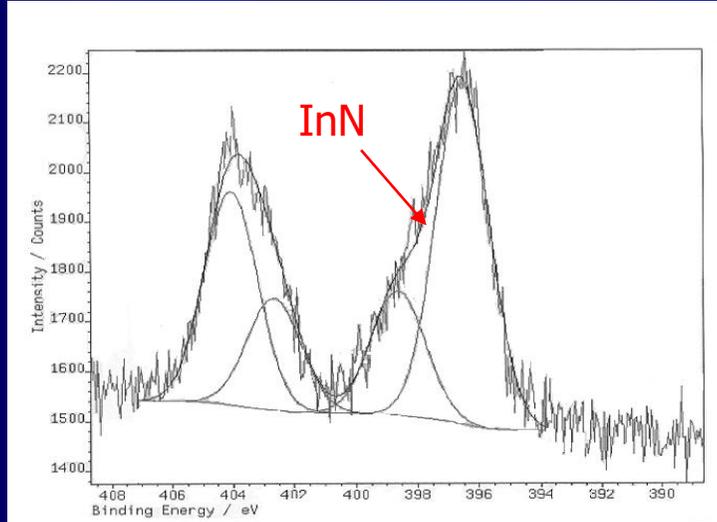


## Glass - 450°C



# XPS - N 1s with surface cleaning by Ar ion etching

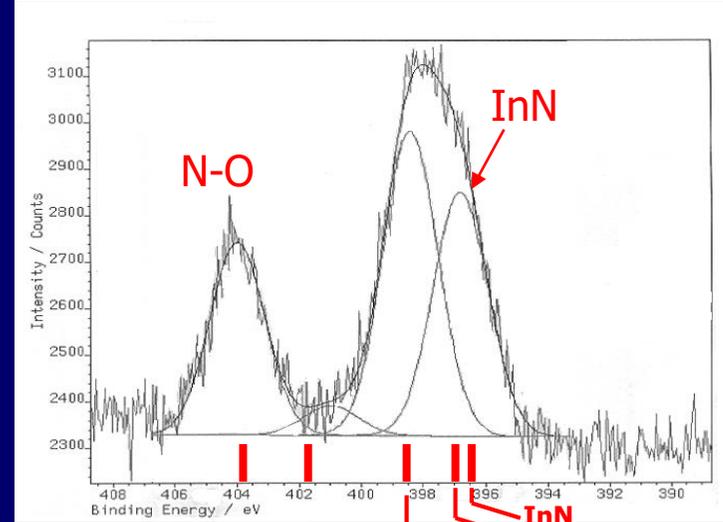
C-plane sapphire - R.T.



N 1s  
In 3d

0.50

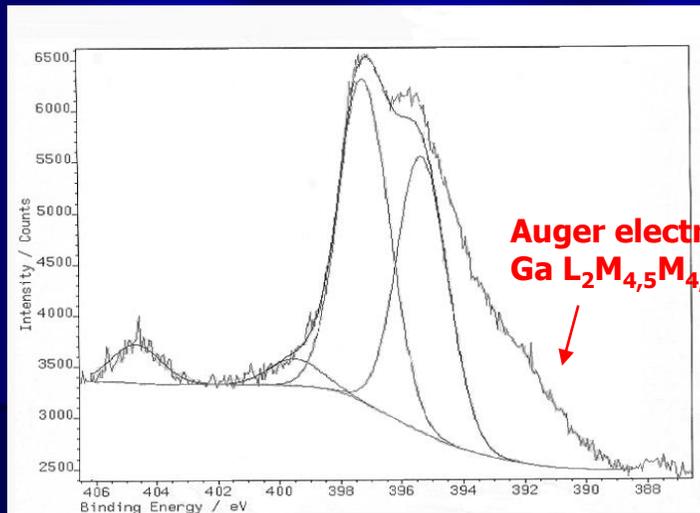
Glass - R.T.



N 1s  
In 3d

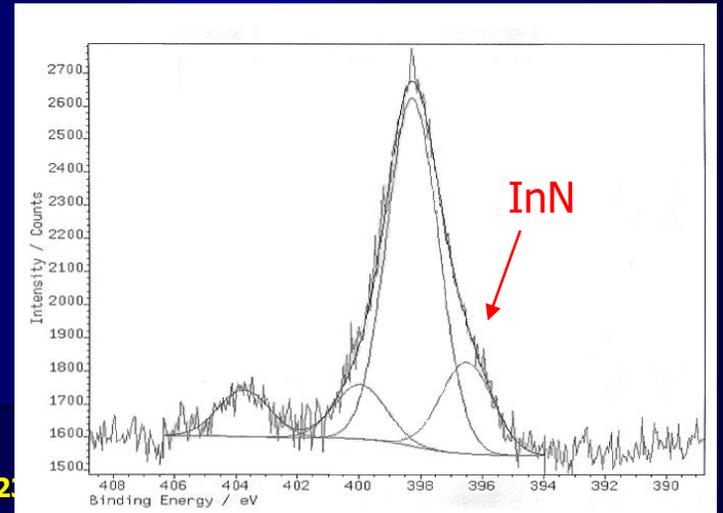
0.50

GaN/A-plane sapphire - R.T.



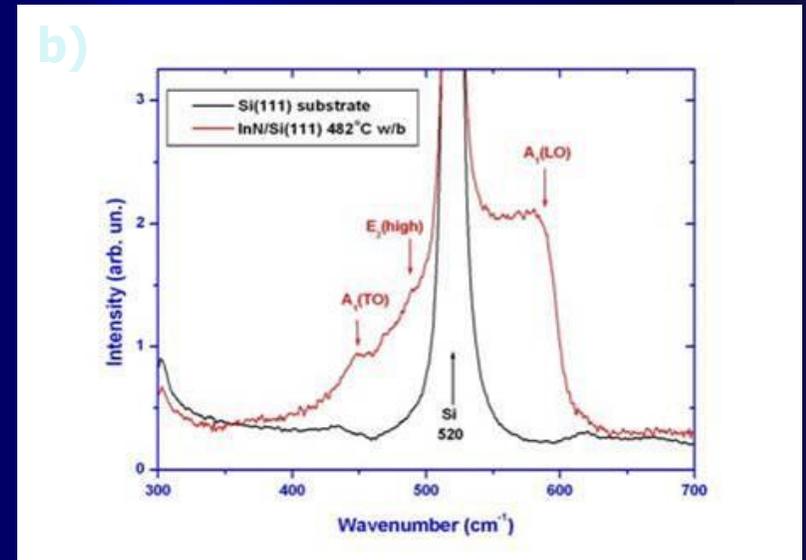
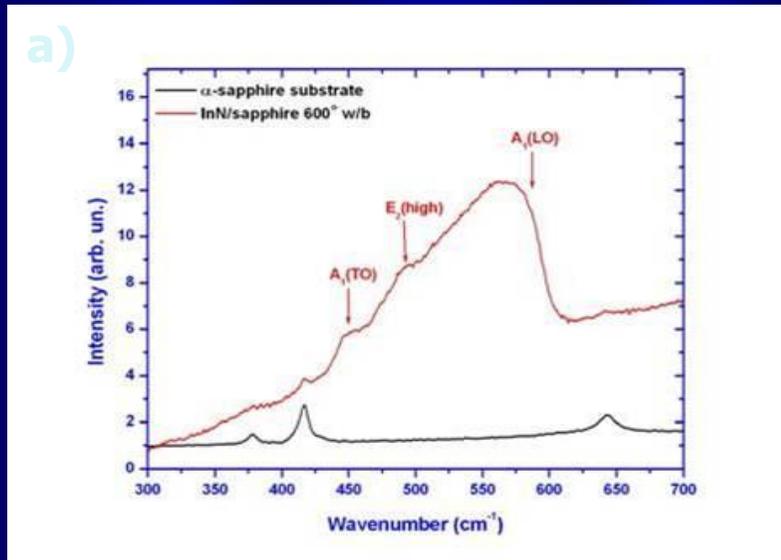
1.66

Glass - 450°C



1.30

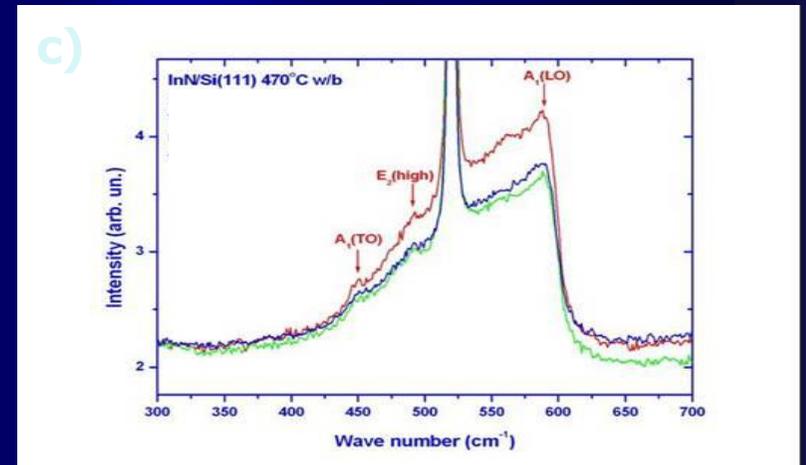
# Raman Spectroscopy



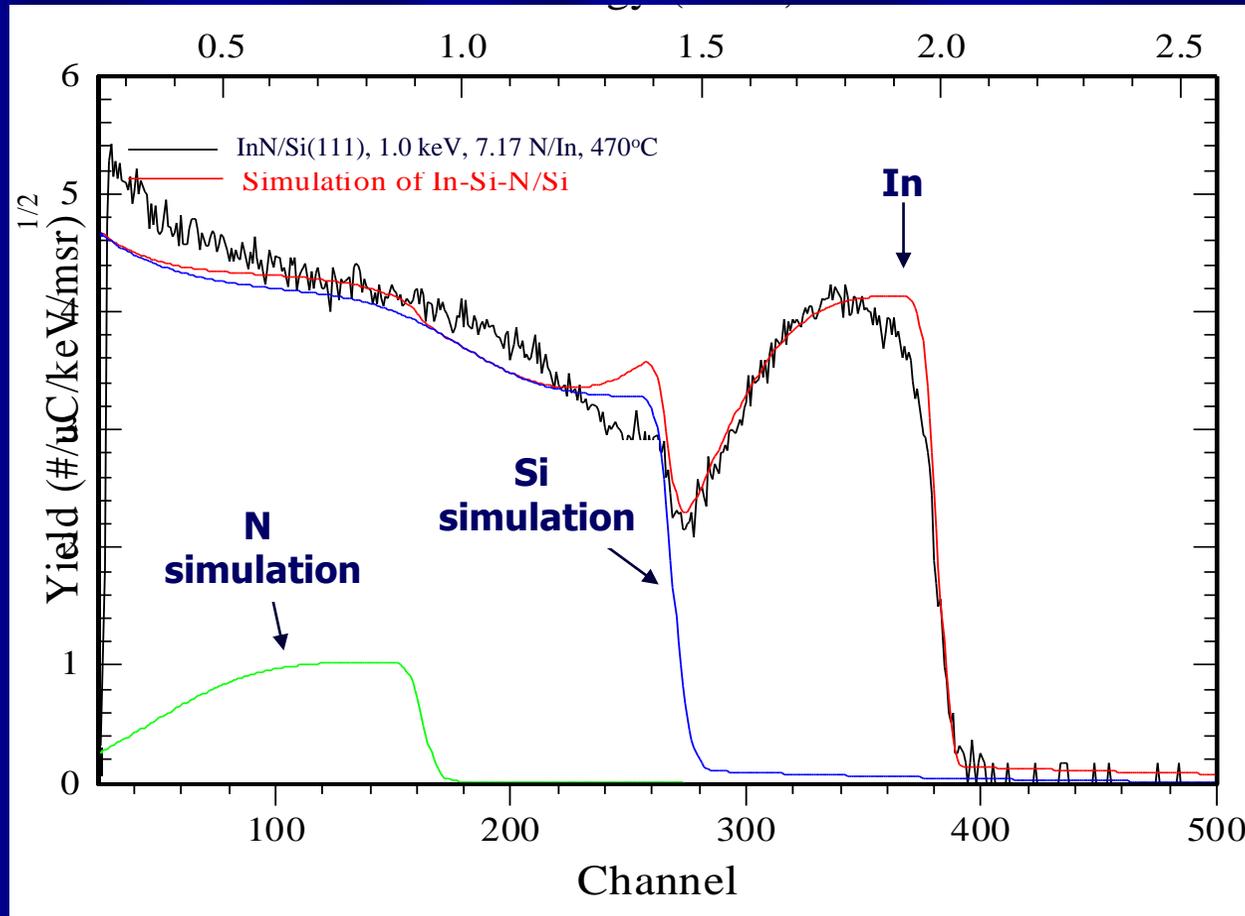
a) Raman spectrum for the InN/sapphire film. Three phonon modes for wurtzite-type InN are observed. The spectrum of the sapphire substrate is shown.

b) Raman spectrum for the InN/Si(111) film. Three phonon modes for wurtzite-type InN are observed. The peak of Si(111) substrate can be also observed.

c) Raman spectra for the InN/Si(111) film showing an anisotropy of the InN  $A_1(\text{LO})$  at  $590 \text{ cm}^{-1}$  after three sequential readings turning the sample by  $30^\circ$ .



# RBS

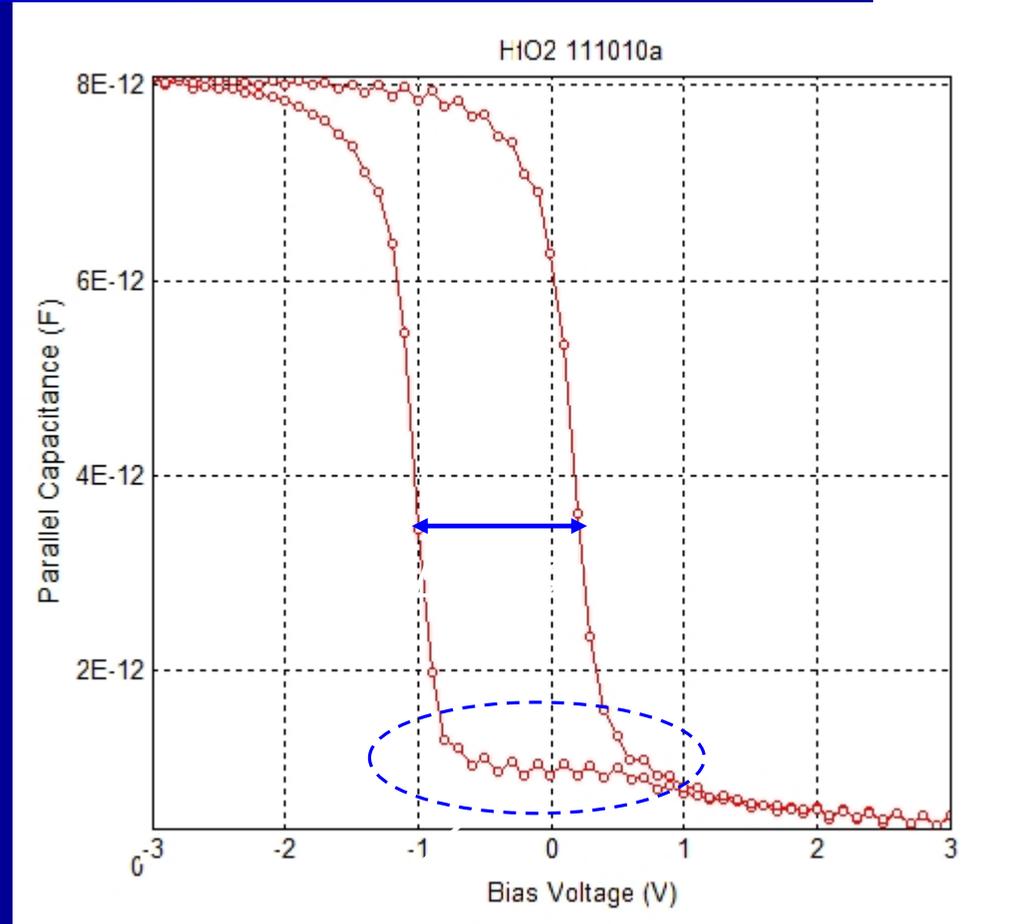


**RBS spectrum of an InN/Si(111) showing a predominant presence of In over N on the surface of the film. This sample showed a poor crystallization characteristic.**

# Capacitance-Voltage Measurements on IBAD HfO<sub>2</sub> Films

Capacitor thickness = 1057Å

- HfO<sub>2</sub> film deposited by IBAD
- 100 μm diameter dot  
k = 12.3
- Hysteresis indicates charge trapped within the oxide



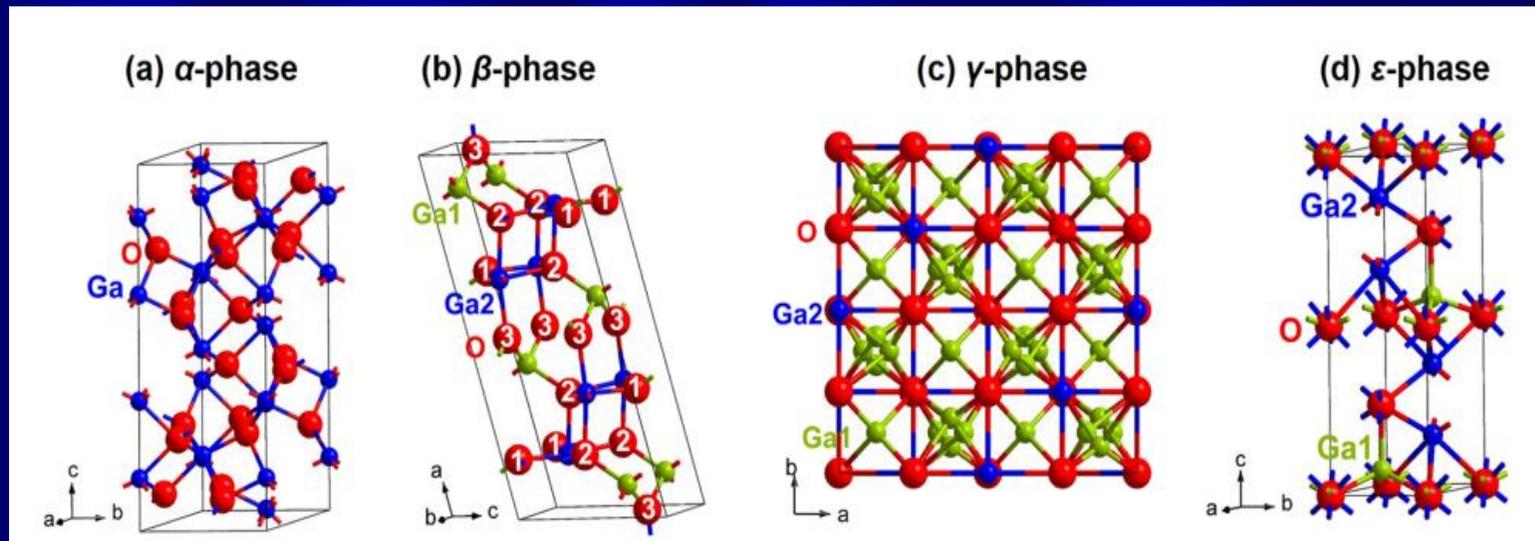
Interface trapped charges

# Gallium Oxide - $\text{Ga}_2\text{O}_3$

- *It is a wide band gap semiconductor, between 4.5 eV - 4.9 eV*
- *$\text{Ga}_2\text{O}_3$  single crystals have  $\alpha$ ,  $\beta$ ,  $\delta$ ,  $\varepsilon$  and  $\gamma$ , polymorphs with free energy of formation in the order of  $\beta < \varepsilon < \alpha < \delta < \gamma$*
- *Among the different  $\text{Ga}_2\text{O}_3$  polymorphs, the  $\beta$ -phase is the most stable crystalline phase with a monoclinic structure*

# Gallium Oxide - $\text{Ga}_2\text{O}_3$

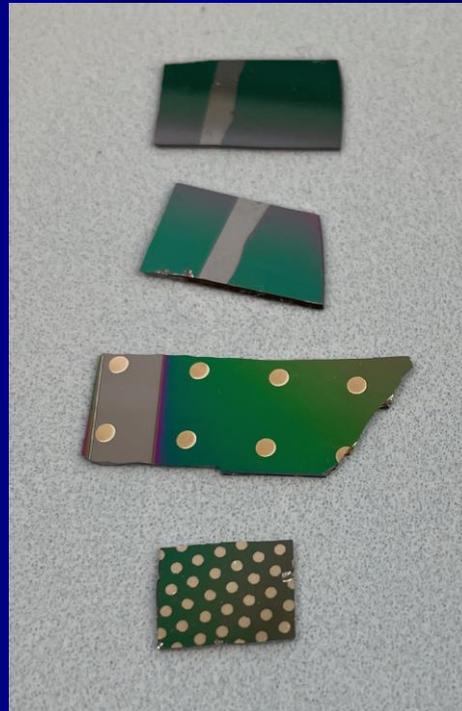
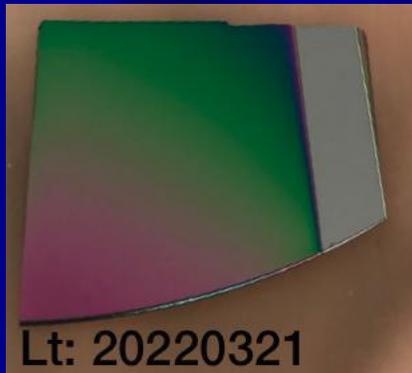
## *Crystallographic structure of gallium oxide phases*



# Deposition on Si(100) substrates

Sample	Substrate	Temperature	d.d.p.	Beam
GaO1	Si p	20 °C	500 V	50 mA/500 V
GaO2 blue	Si p	20 °C	500 V	50 mA/500 V
GaO2 red	Si p	20 °C	500 V	50 mA/500 V
GaO4	Si p	400 °C	500 V	50 mA/500 V
GaO5	Si p	800 °C	500 V	50 mA/500 V

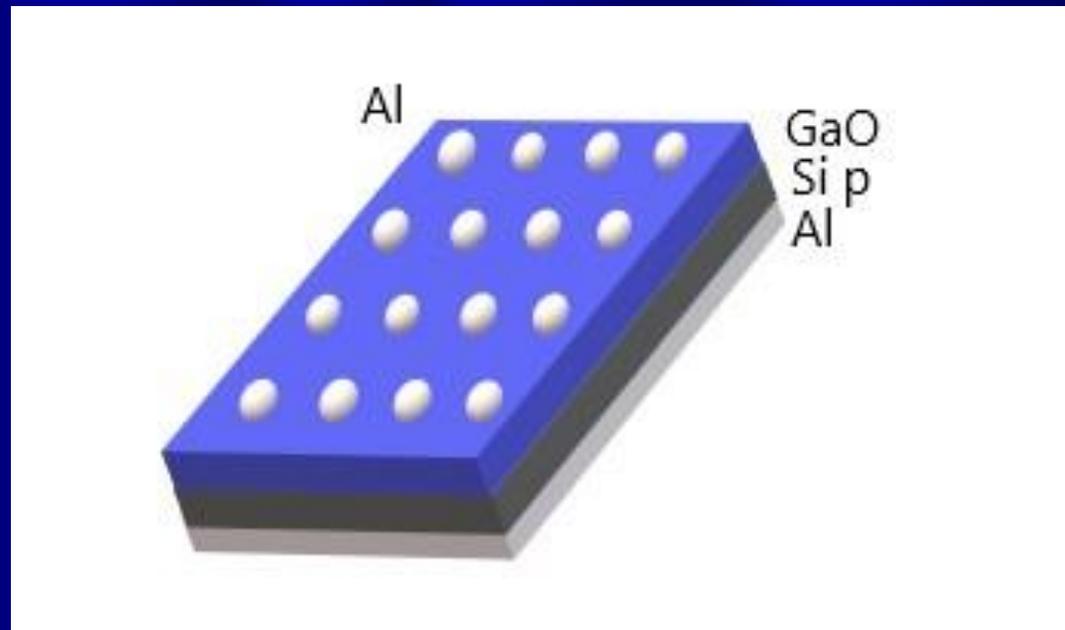
# Ga<sub>2</sub>O<sub>3</sub> films



Ga<sub>2</sub>O<sub>3</sub> film deposited on Si(100), with 500 eV ion beam energy

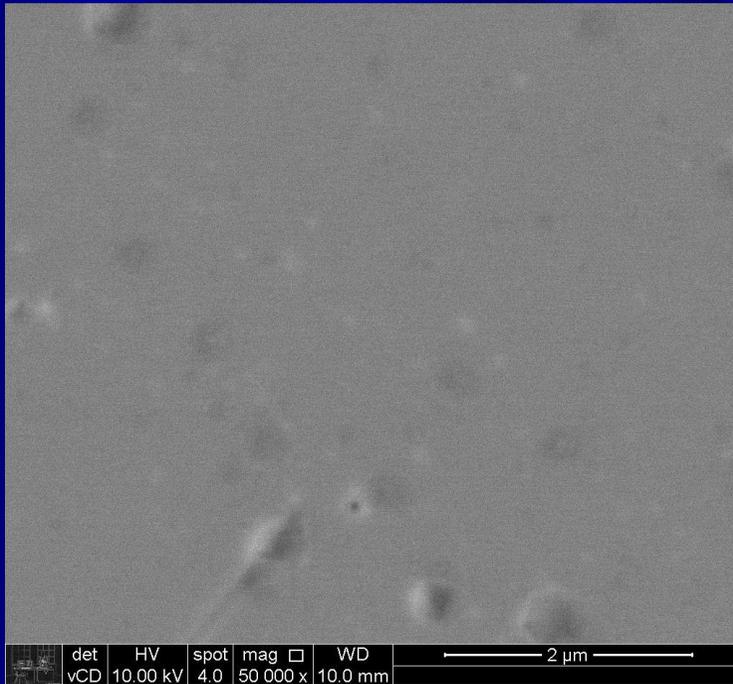
# Testing Device Assembly

*For electrical measurements, testing devices were manufactured on thin  $\text{Ga}_2\text{O}_3$  films*

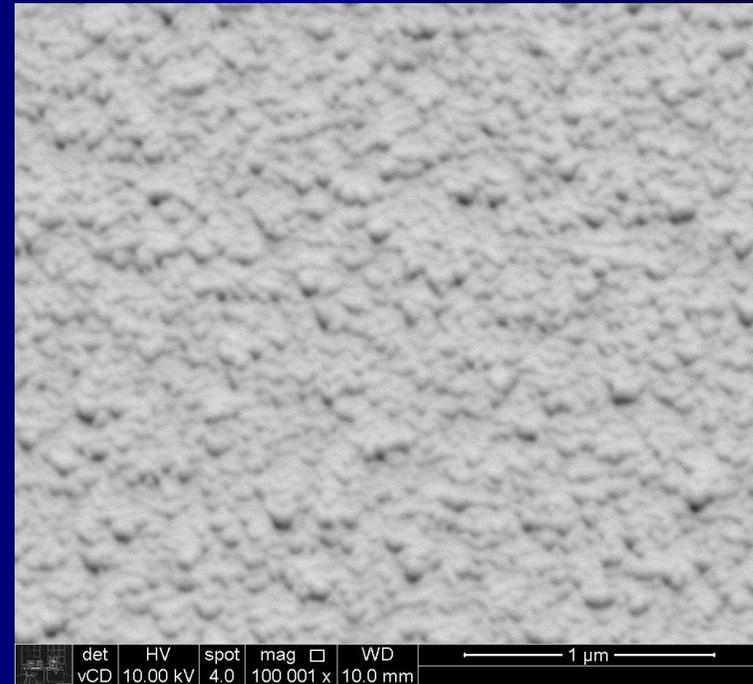


# Film morphology: SEM

Sample	T (°C)
GaO-01	20
GaO-02 blue	20
GaO-02 red	20
GaO-04	400
GaO-05	800



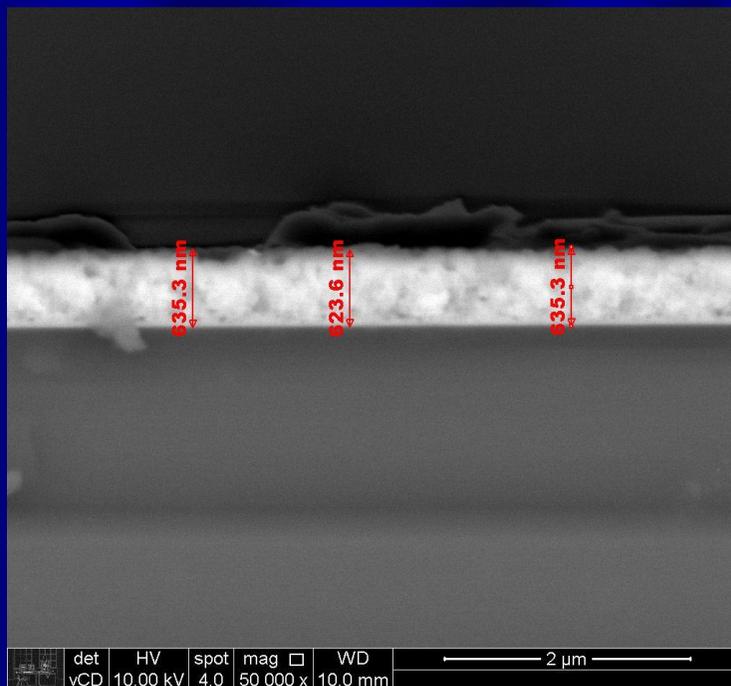
GaO1



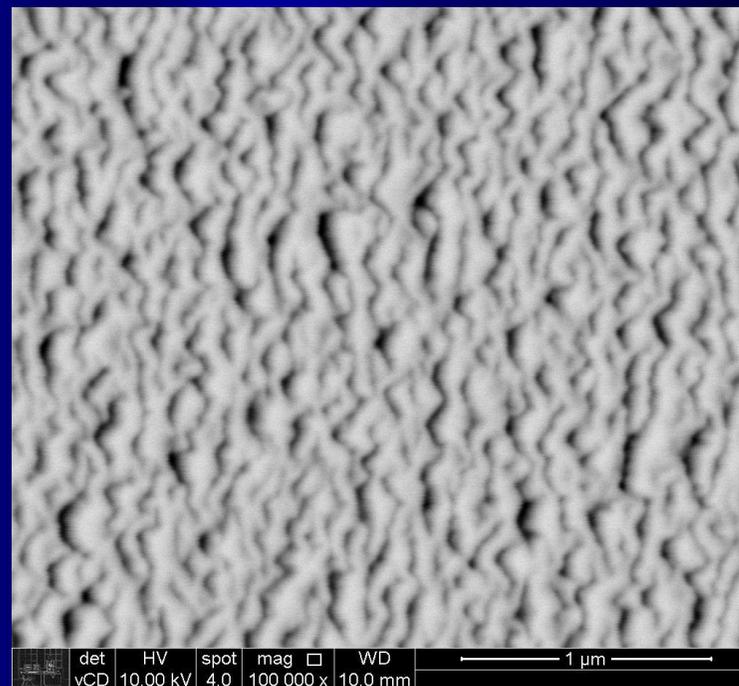
GaO2 blue

# Film morphology: SEM

Sample	T (°C)
GaO-01	20
GaO-02 blue	20
GaO-02 red	20
GaO-04	400
GaO-05	800



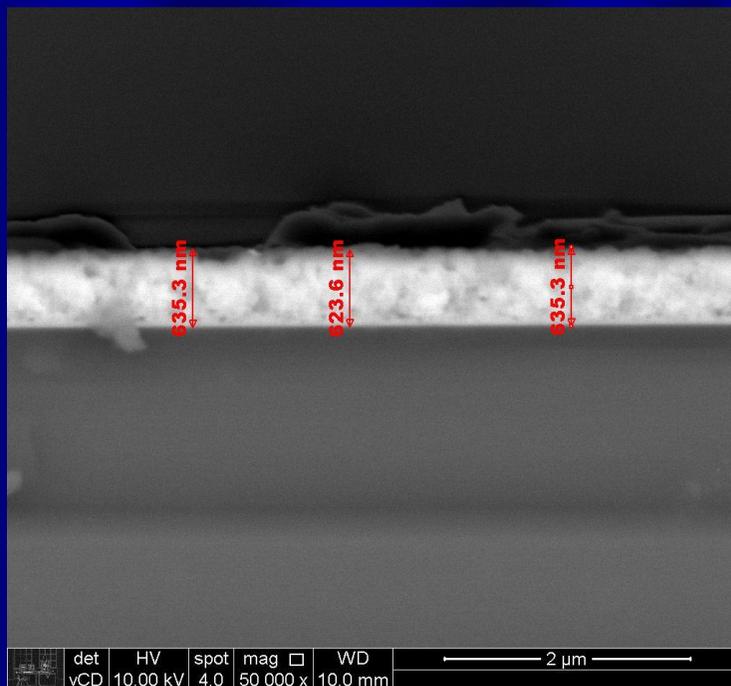
GaO4



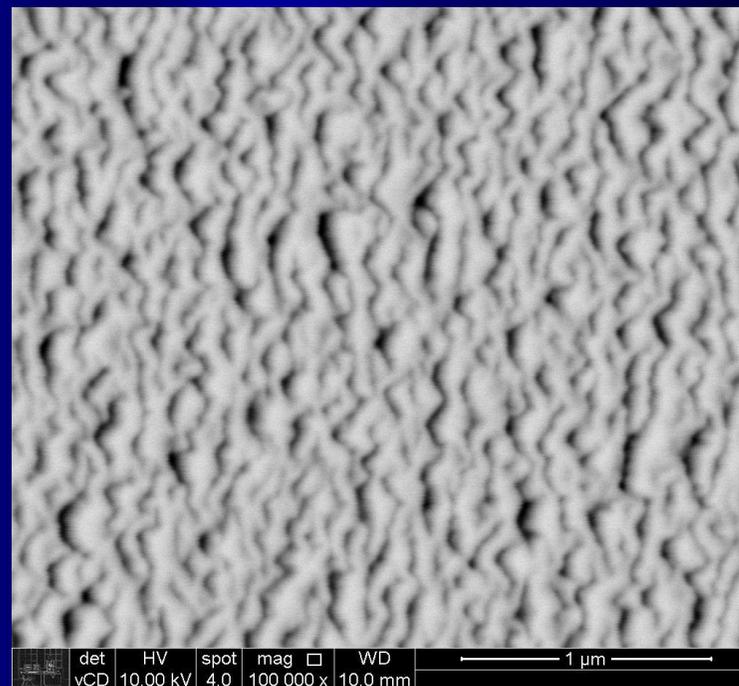
GaO4

# Film morphology: SEM

Sample	T (°C)
GaO-01	20
GaO-02 blue	20
GaO-02 red	20
GaO-04	400
GaO-05	800



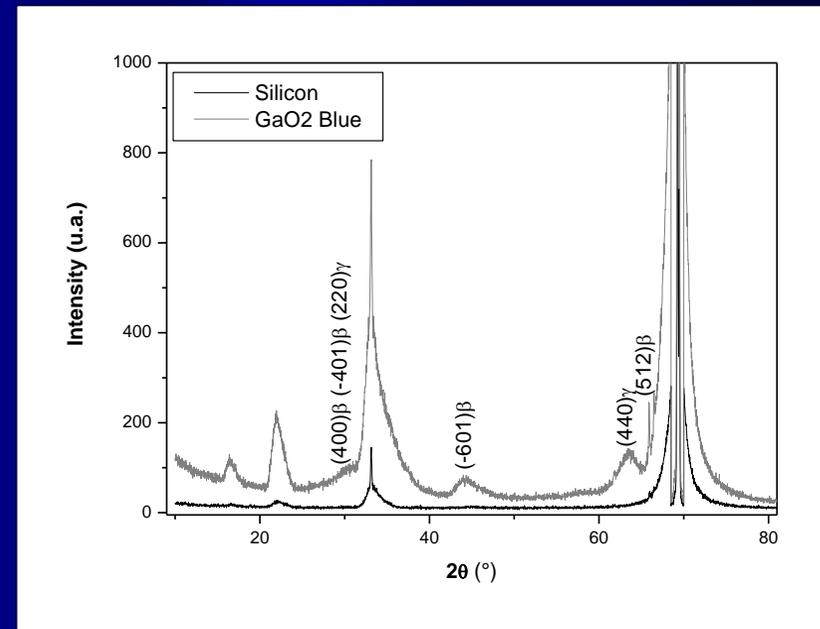
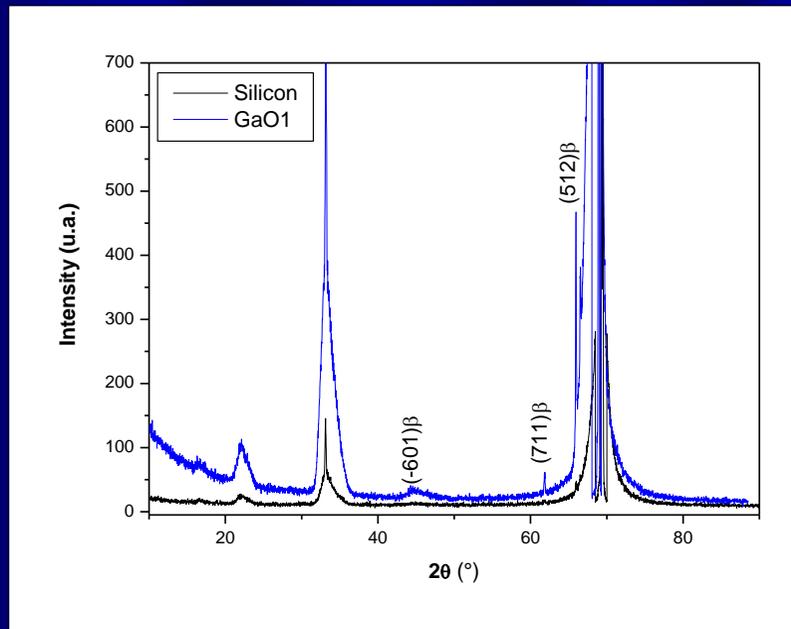
GaO4



GaO4

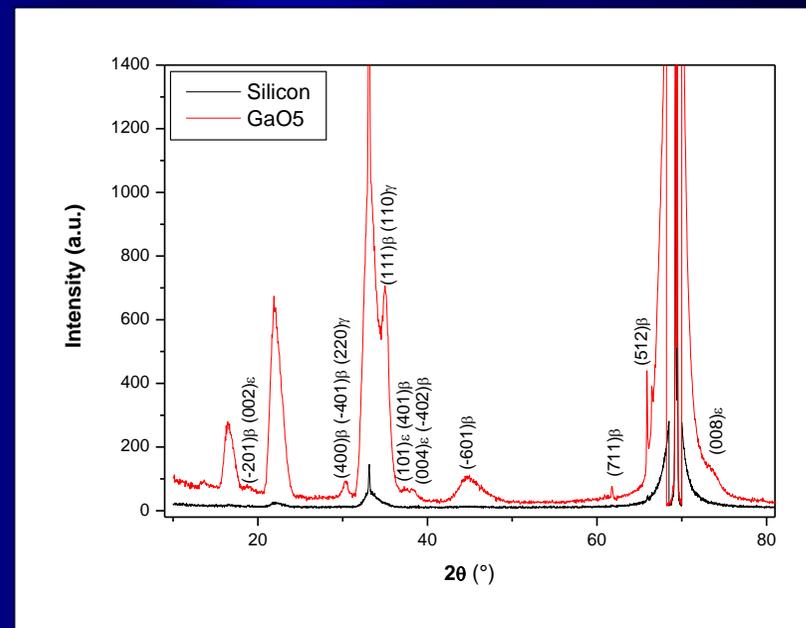
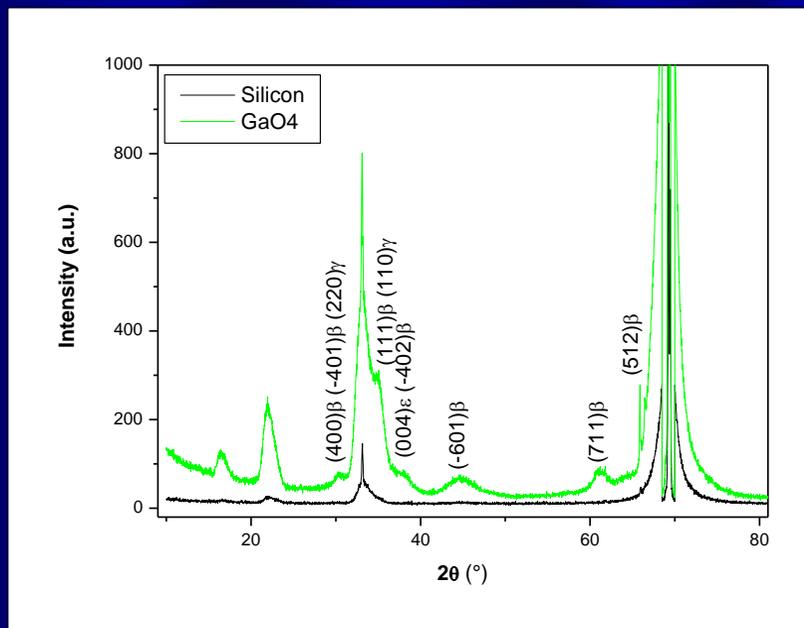
# Structural property: XRD

Sample	T (°C)
GaO-01	20
GaO-02 blue	20
GaO-02 red	20
GaO-04	400
GaO-05	800

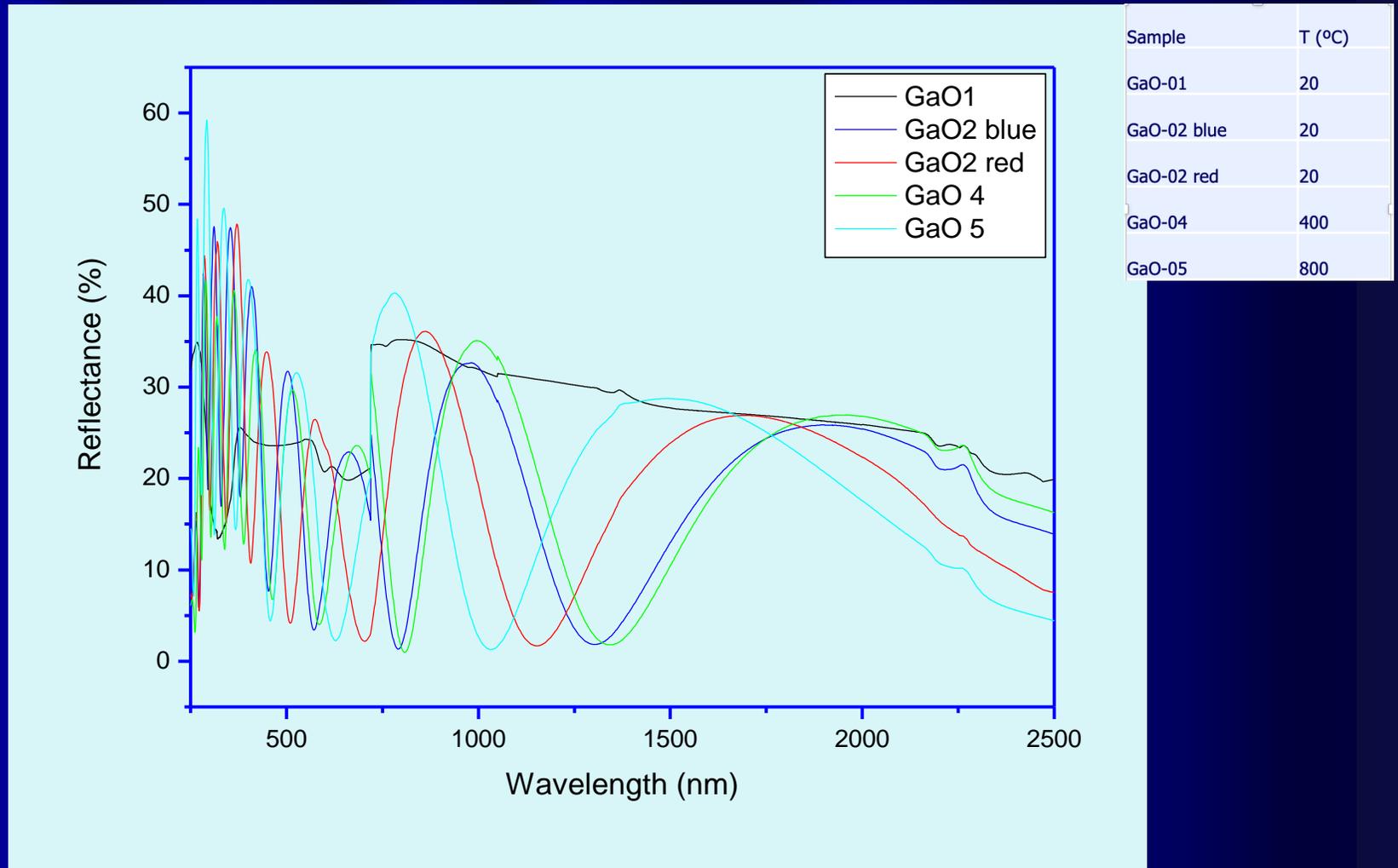


# Structural property: XRD

Sample	T (°C)
GaO-01	20
GaO-02 blue	20
GaO-02 red	20
GaO-04	400
GaO-05	800



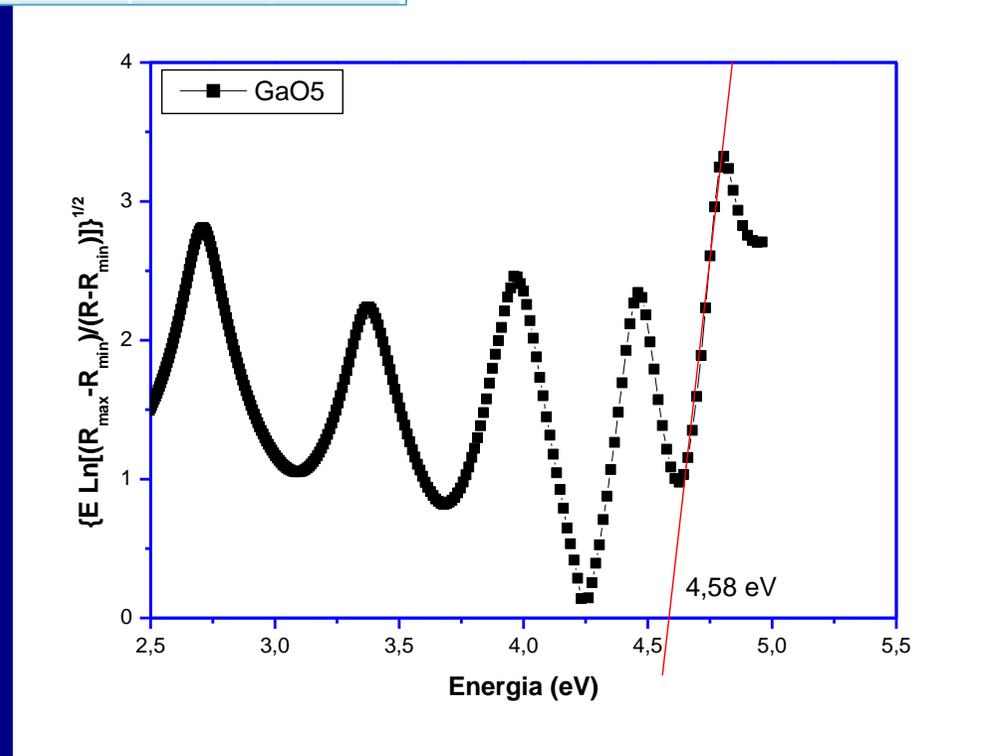
# Optical Absorption %R



**%R spectra from the gallium oxide films prepared on silicon substrate at different temperatures.**

# Band Gap values

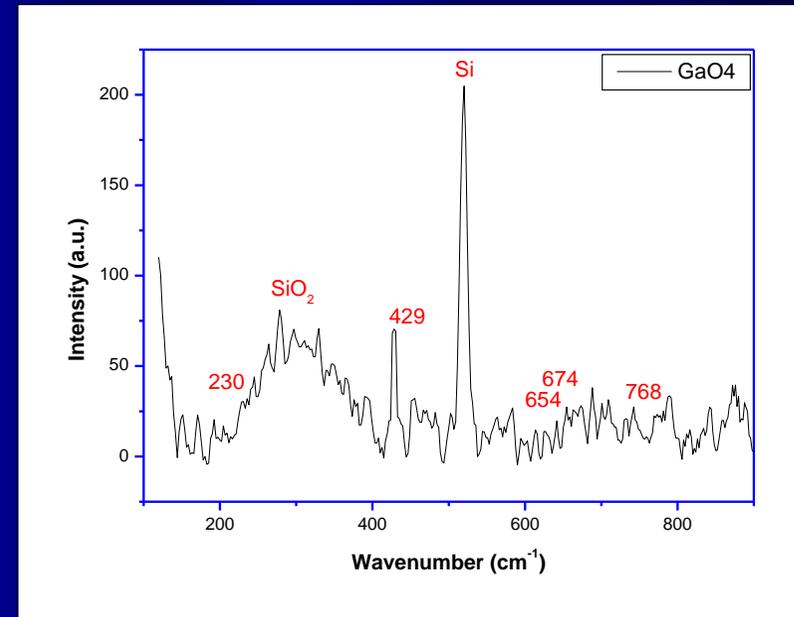
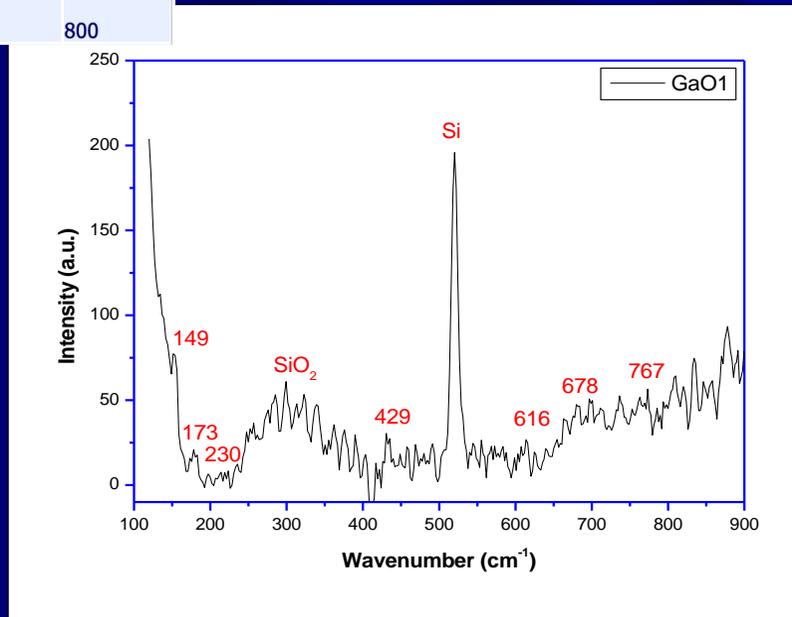
Amostra	Espessura (nm)	Band Gap (eV)
GaO1	150	4,54
GaO2 <b>Blue</b>	639	4,35
GaO2 <b>Red</b>	552	4,33
GaO4	524	4,50
GaO5	358	4,58



Sample	T (°C)
GaO-01	20
GaO-02 blue	20
GaO-02 red	20
GaO-04	400
GaO-05	800

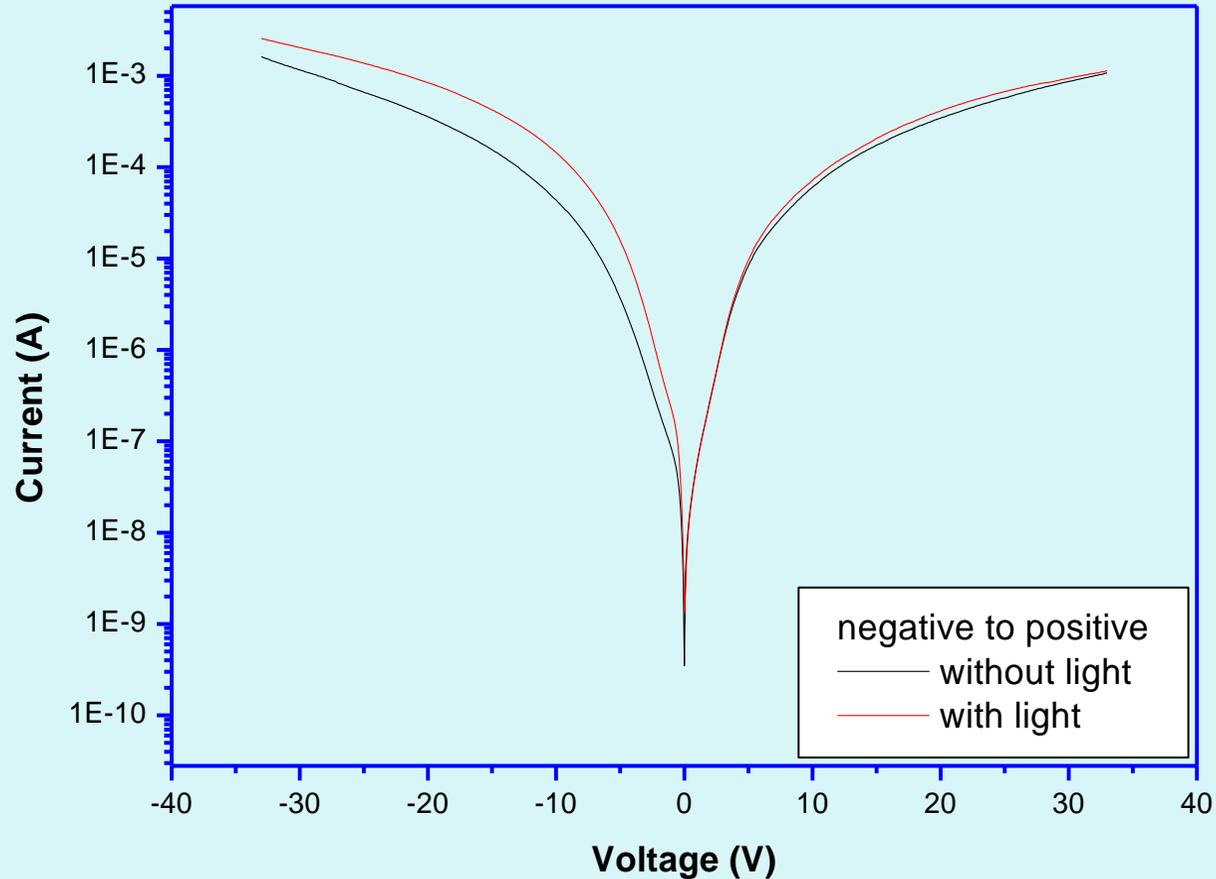
# Raman spectrum of Ga<sub>2</sub>O<sub>3</sub>/Si(100)

Sample	T (°C)
GaO-01	20
GaO-02 blue	20
GaO-02 red	20
GaO-04	400
GaO-05	800



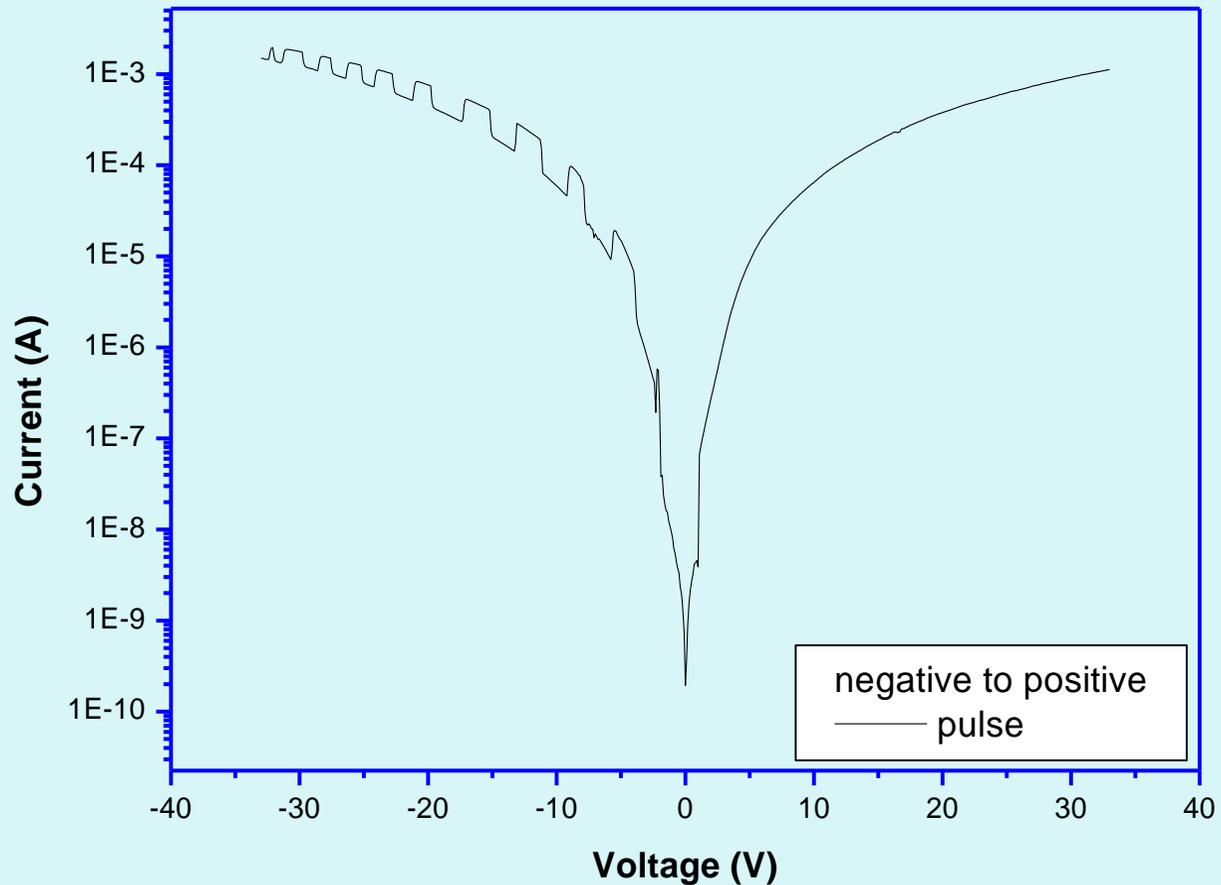
# IxV from Ga<sub>2</sub>O<sub>3</sub>/Si(100)

Sample	T (°C)
GaO-01	20
GaO-02 blue	20
GaO-02 red	20
GaO-04	400
GaO-05	800

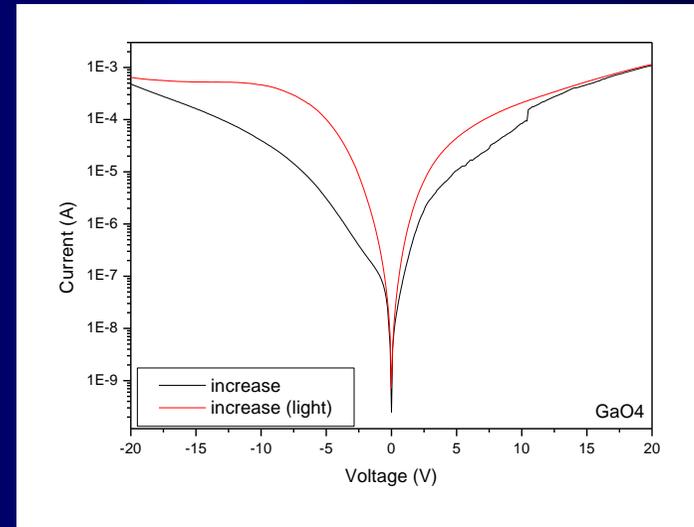
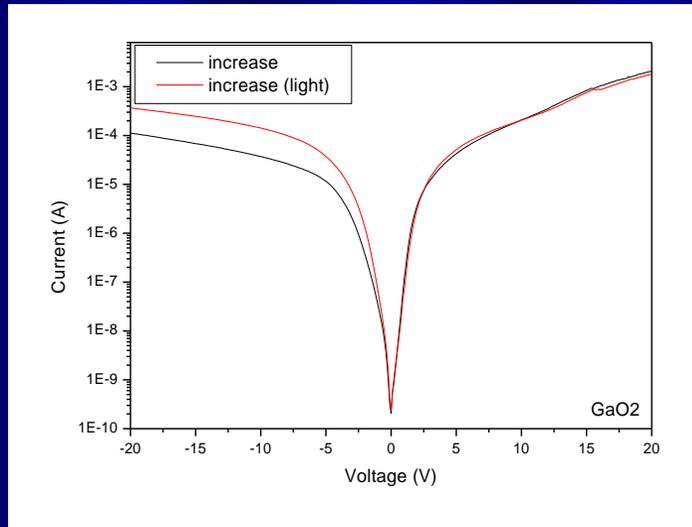


# IxV from Ga<sub>2</sub>O<sub>3</sub>/Si(100)

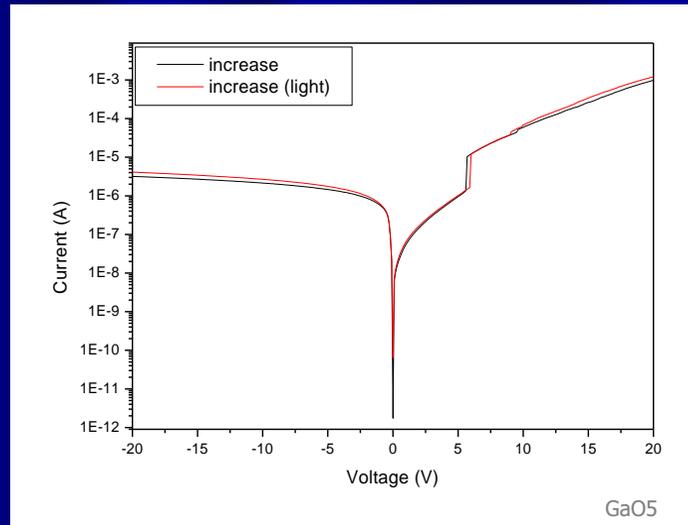
Sample	T (°C)
GaO-01	20
GaO-02 blue	20
GaO-02 red	20
GaO-04	400
GaO-05	800



# IxV from Ga<sub>2</sub>O<sub>3</sub>/Si(100)

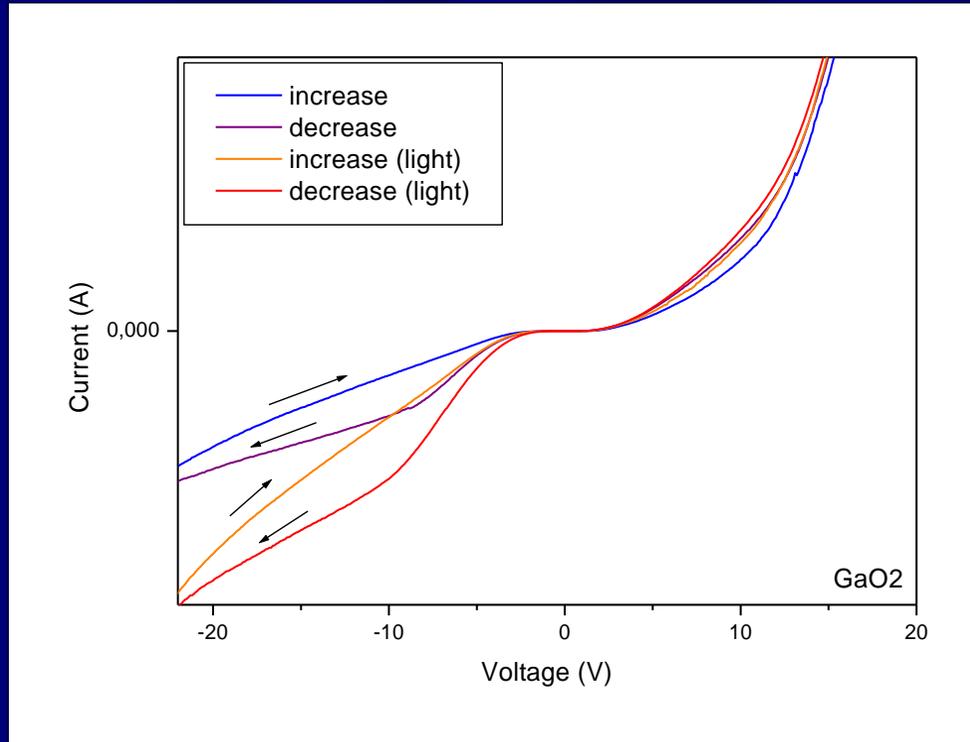


Sample	T (°C)
GaO-01	20
GaO-02 blue	20
GaO-02 red	20
GaO-04	400
GaO-05	800



# IxV from Ga<sub>2</sub>O<sub>3</sub>/Si(100)

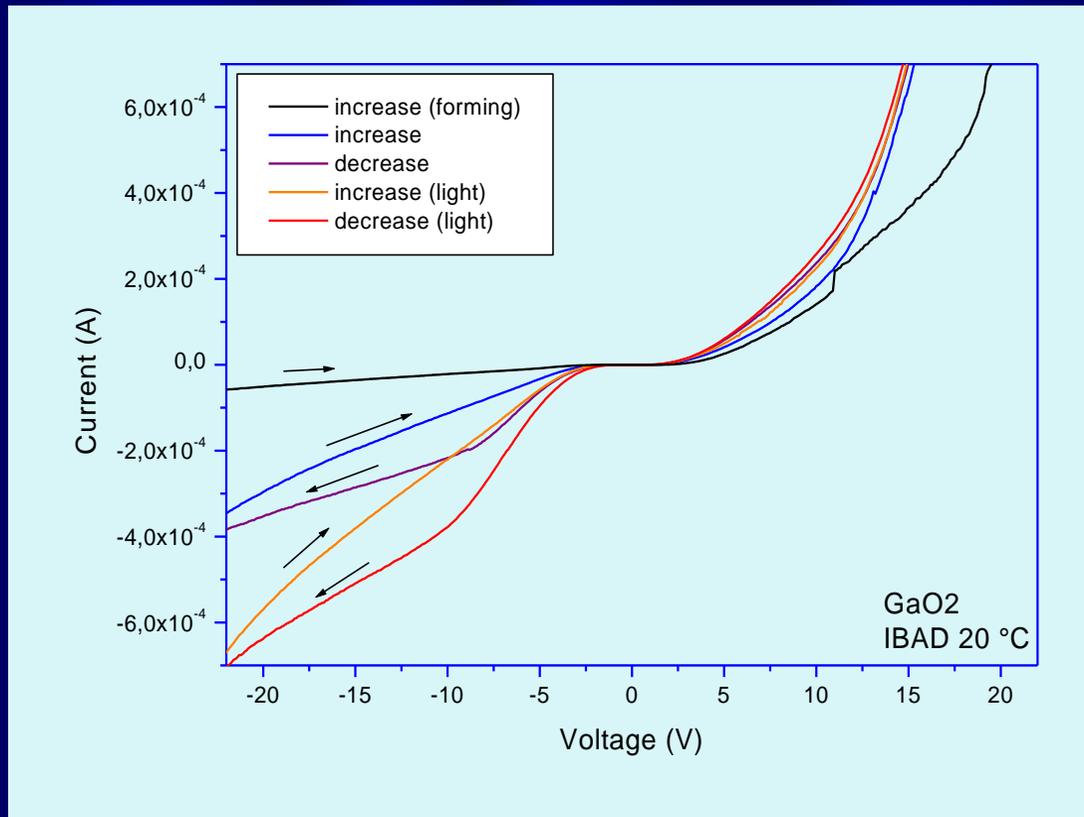
Sample	T (°C)
GaO-01	20
GaO-02 blue	20
GaO-02 red	20
GaO-04	400
GaO-05	800



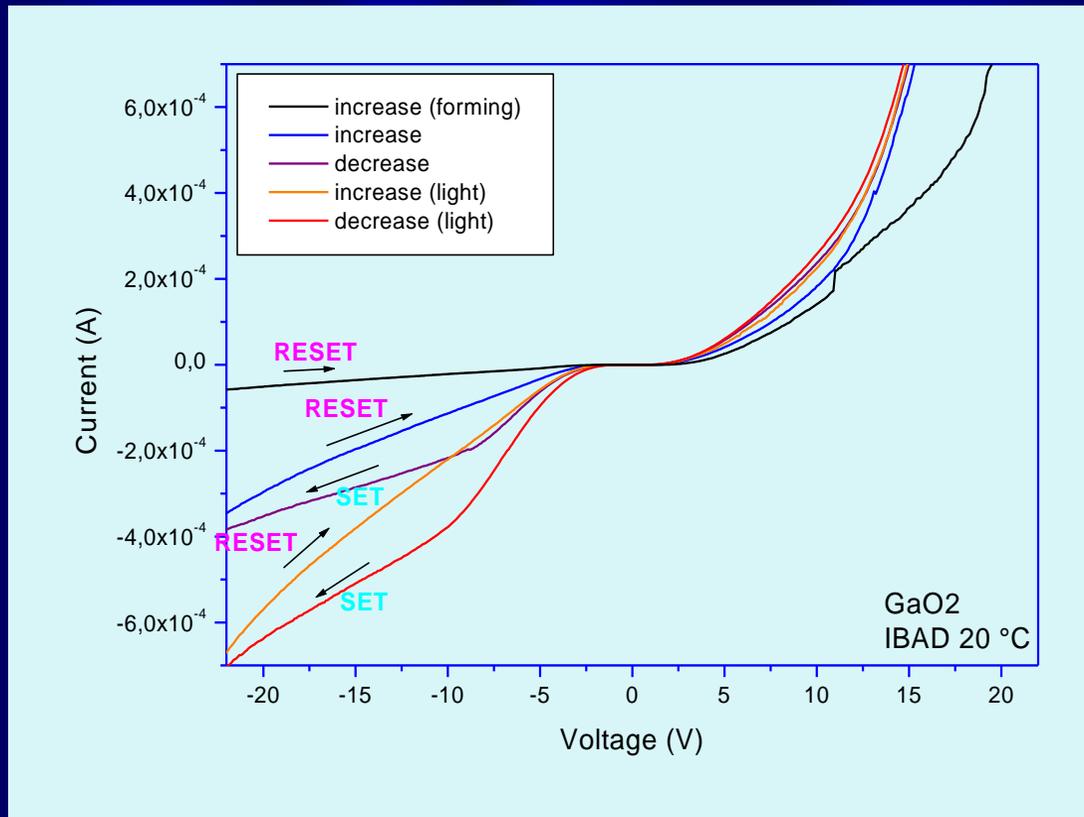
**This figure shows the photoresistive memory behavior.  
The curves shows different response for illuminated situation.  
In the region around 0 volts there are the indication of resistance state change.**

# ReRAM de GaO

# GaO2



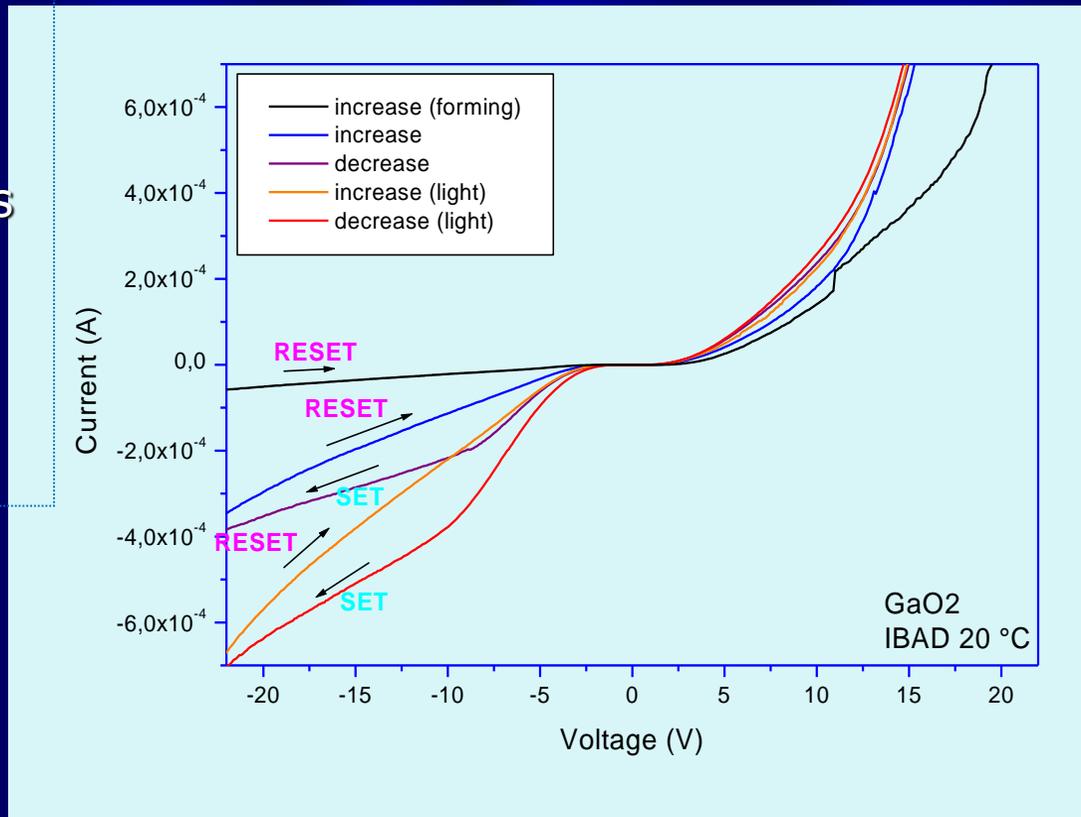
# GaO2



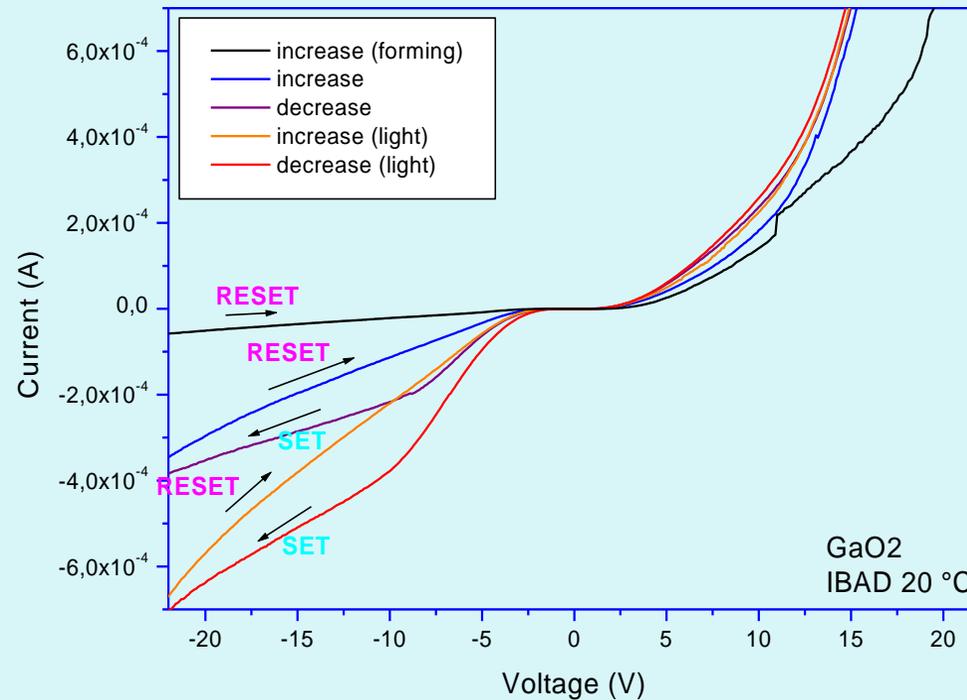
# GaO2

## RESET

All devices present same behavior when the resistance makes the transition from the low resistance state to the high resistance state.



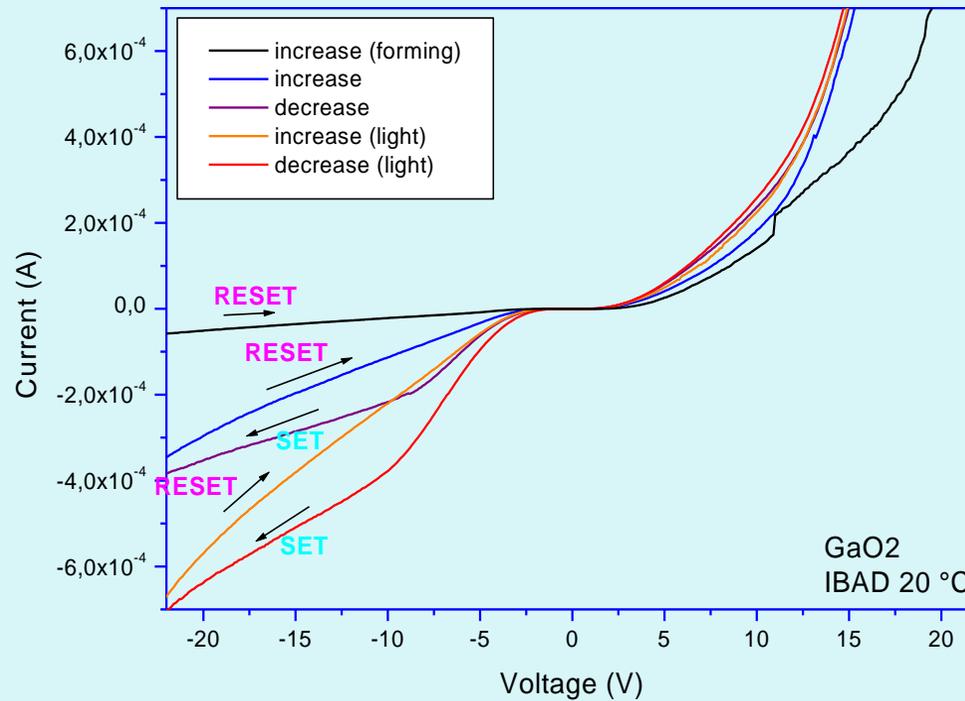
# GaO2



## SET

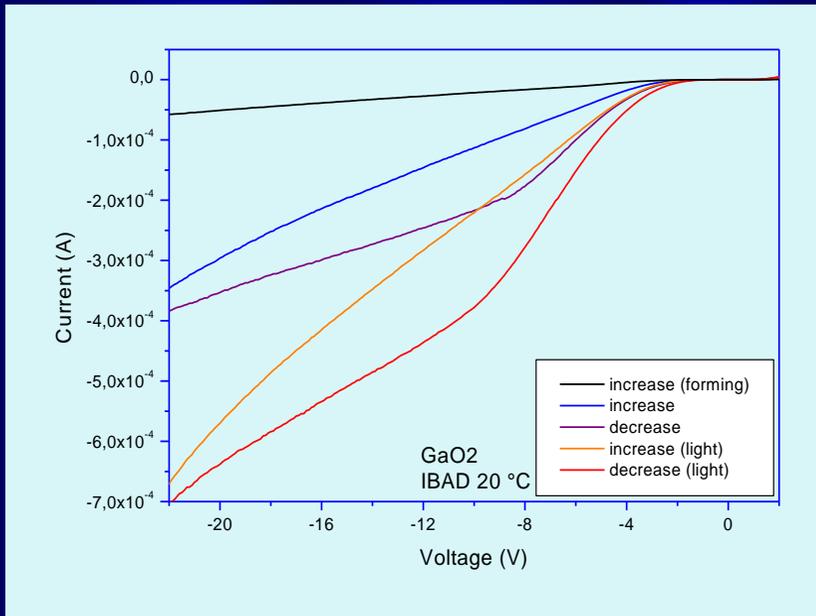
The same can be said when resistance goes the other way.

# GaO2



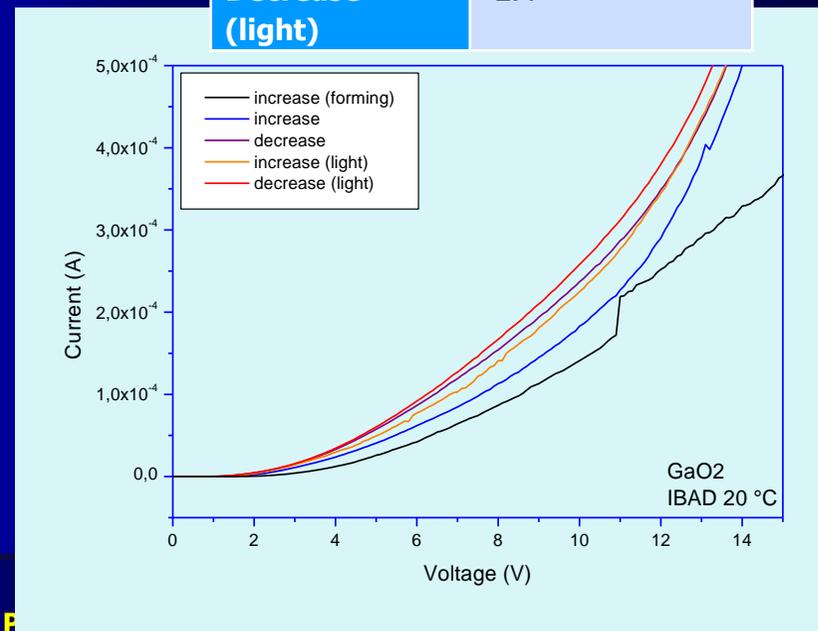
It is possible to say that all resistive memories are unipolar

# GaO2

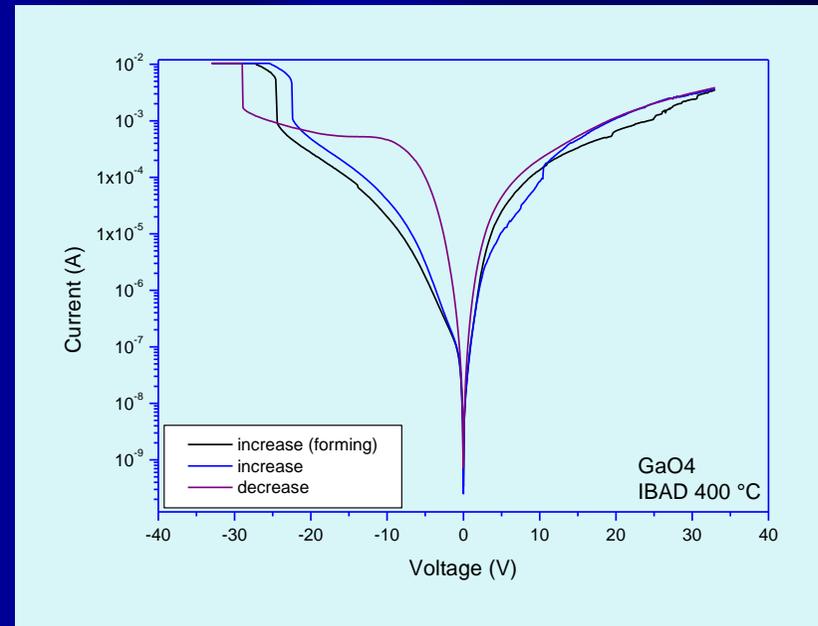
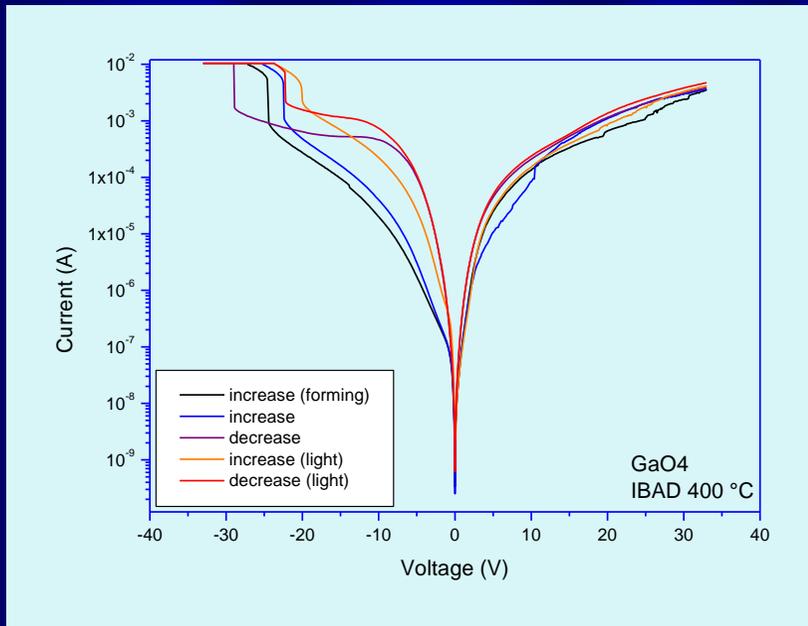


	Voltage (V) Reference: $1 \times 10^{-5}$ A
Increase (forming)	3.8
Increase	3.0
Decrease	2.6
Increase (light)	2.7
Decrease (light)	2.6

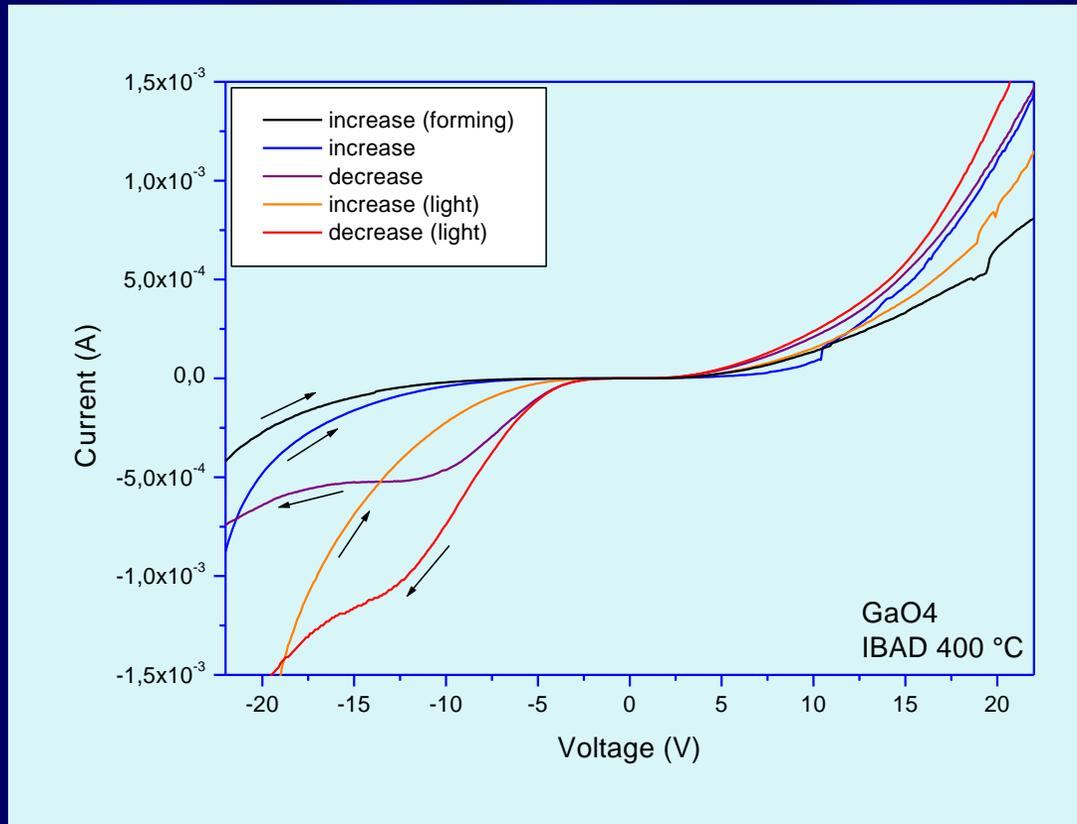
	Voltage (V) Reference: $-1 \times 10^{-5}$ A
Increase (forming)	-5.6
Increase	-3.4
Decrease	-2.8
Increase (light)	-3.0
Decrease (light)	-2.4



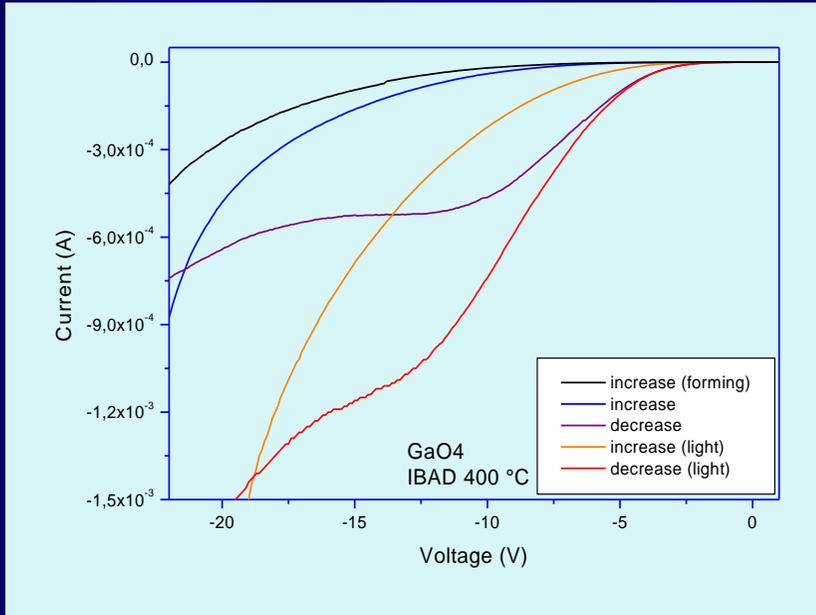
# GaO4



# GaO4

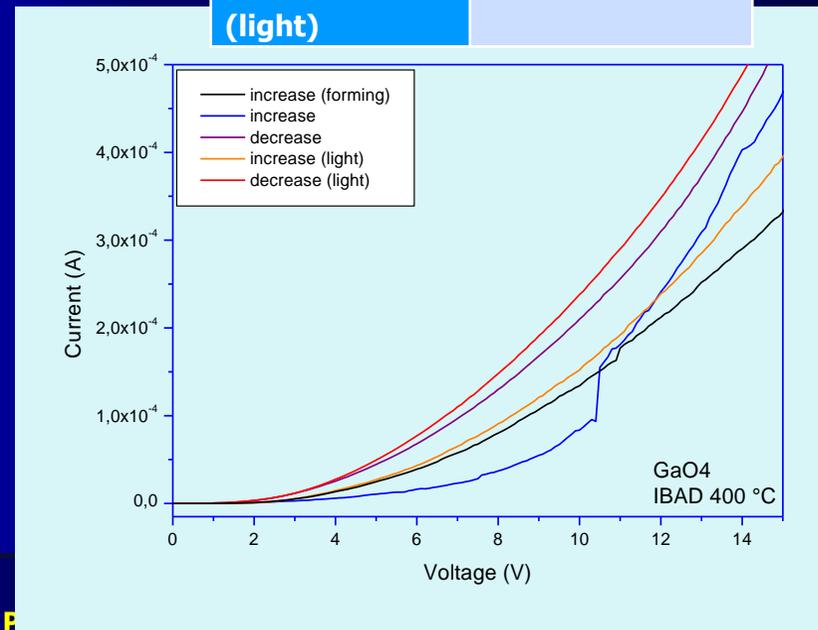


# GaO4

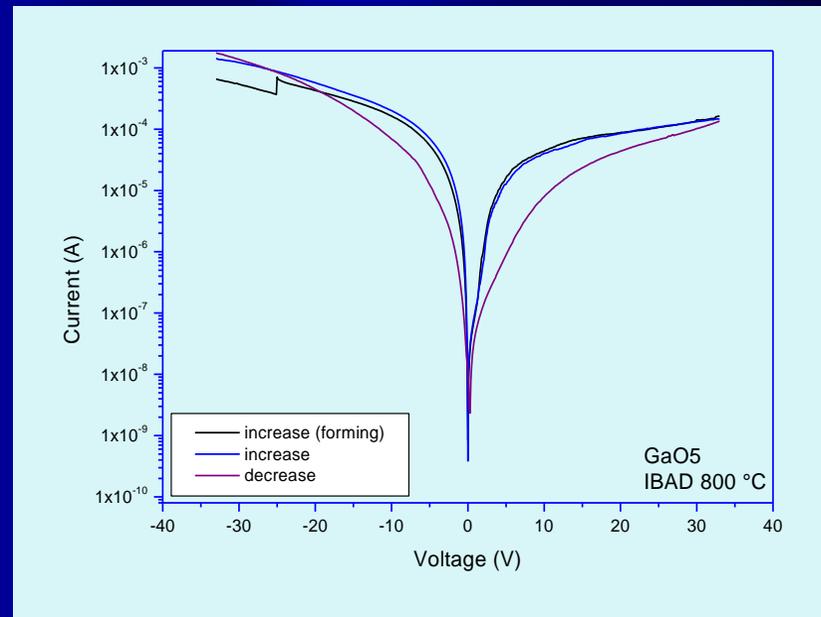
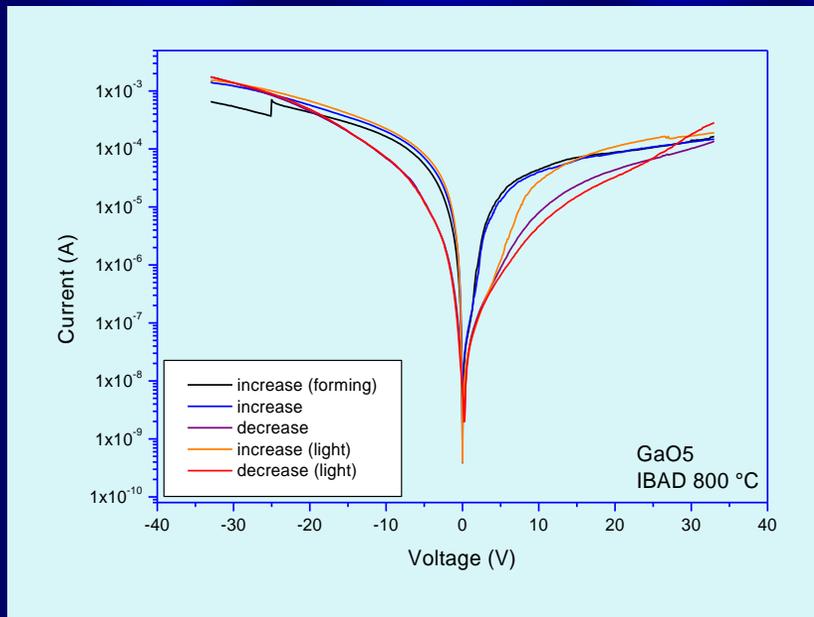


	Voltage (V) Reference: $1 \times 10^{-5}$ A
Increase (forming)	3.7
Increase	4.9
Decrease	2.9
Increase (light)	3.6
Decrease (light)	2.9

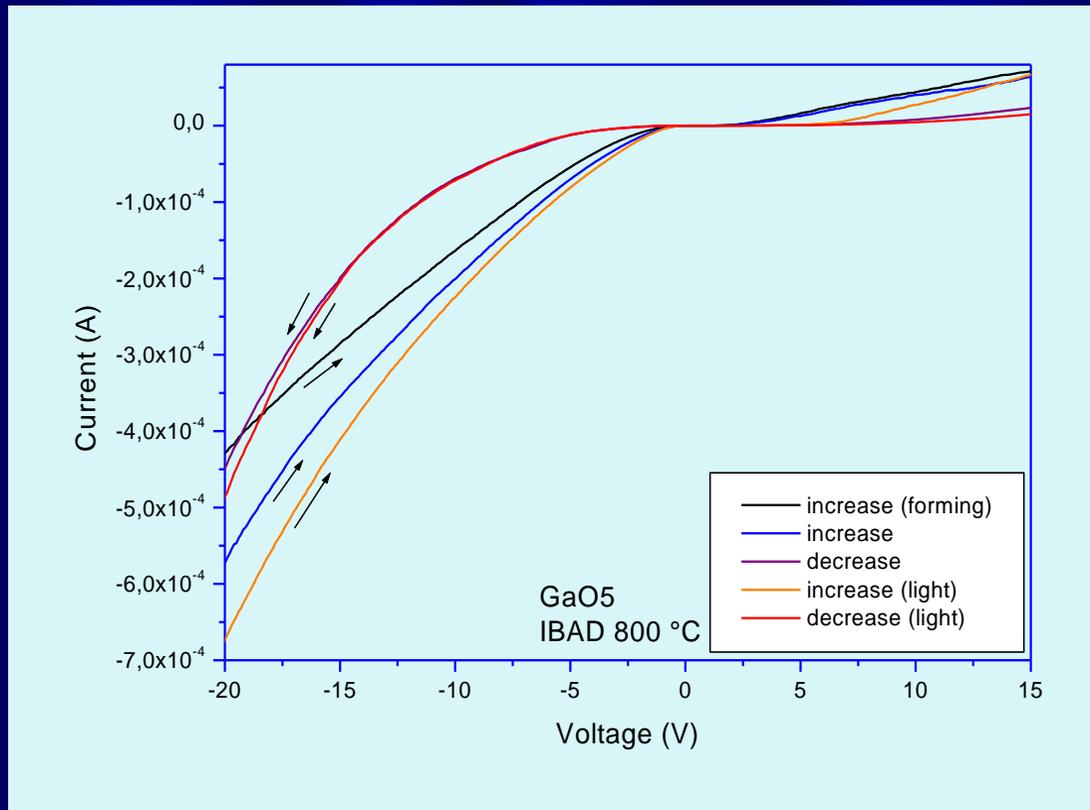
	Voltage (V) Reference: $1 \times 10^{-5}$ A
Increase (forming)	-8.3
Increase	-6.8
Decrease	-2.7
Increase (light)	-3.9
Decrease (light)	-2.6



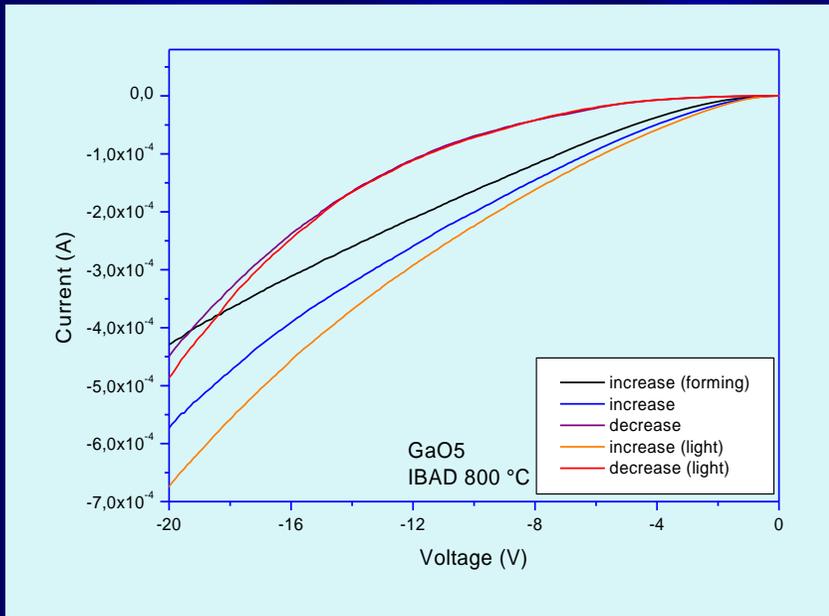
# GaO5



# GaO5

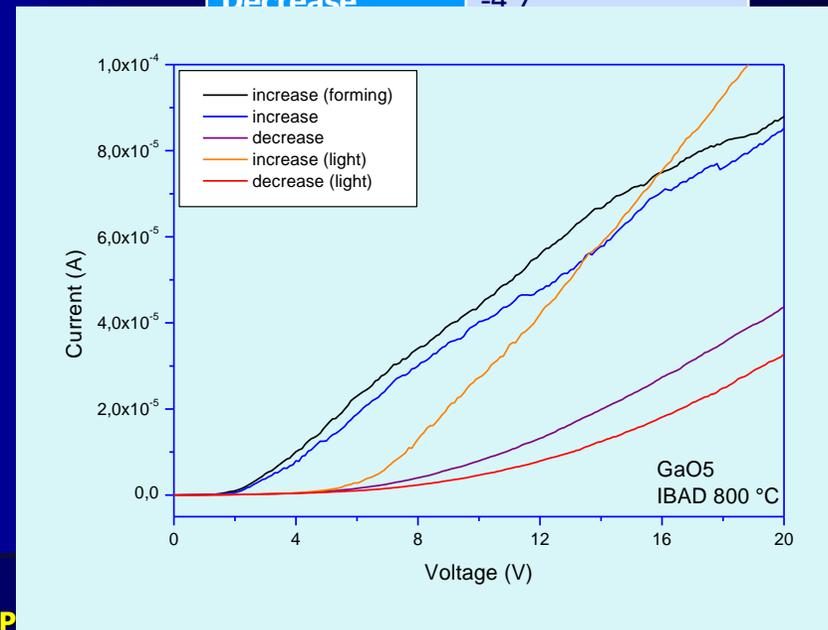


# GaO5



	Voltage (V) Reference: - $1 \times 10^{-5}$ A
Increase (forming)	-2.1
Increase	-1.6
Decrease	-4.6
Increase (light)	-1.5
Decrease	-4.7

	Voltage (V) Reference: $1 \times 10^{-5}$ A
Increase (forming)	4.0
Increase	4.4
Decrease	10.9
Increase (light)	7.7
Decrease (light)	13.1



## Com grafeno, brasileira da USP cria memória que pode revolucionar a tecnologia

Patente foi desenvolvida utilizando óxido condutor transparente com grafeno como memória resistiva

Rodrigo Mozelli | 07/02/2023 02h30, atualizada em 07/02/2023 20h37



Com grafeno, brasileira da USP cria memória que pode revolucionar a tecnologia

Novos chips são 100 vezes mais rápidos que as memórias Flash atuais: eles são do tipo resistivo...

olhardigital.com.br

<https://olhardigital.com.br/2023/02/07/ciencia-e-espaco/pesquisadora-cria-memoria-para-pcs-que-pode-substituir-as-de-silicio/>

12:22

### Inovação na USP

Uma nova maneira de fabricar memórias como essas foi criada por cientistas da **USP** e teve o pedido de registro aceito pelo **INPI** (Instituto Nacional da Propriedade Industrial) em novembro de 2022. Marina Sparvoli, pós-doutoranda do Instituto de Física (IF) da USP, em colaboração com outros pesquisadores, desenvolveu um mecanismo de memória baseado nos **memristores**, a partir de materiais jamais combinados.

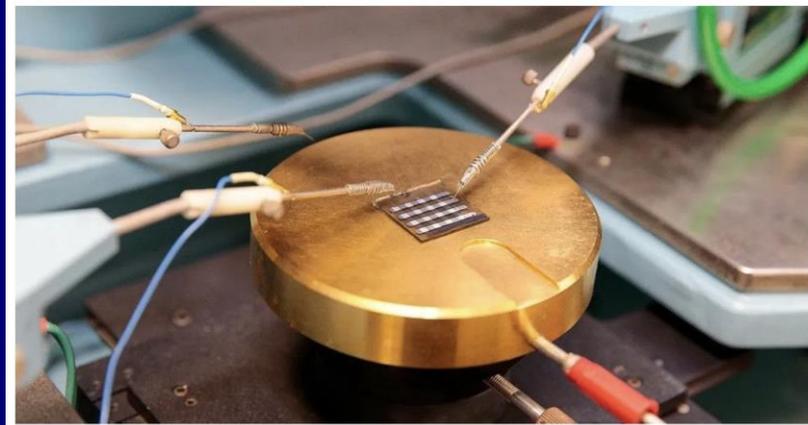


### Material inédito

O professor José Fernando Chubaci, que supervisiona Marina no pós-doutorado, conta que a aplicação de Iton a memórias resistivas nunca foi feita antes.

“Esse é um projeto na fronteira do conhecimento mundial, que terá aplicação no mercado internacional daqui a dez, 15 ou 20 anos. A doutora Marina conseguiu criar memórias resistivas usando grafeno com o Iton, ampliando espaço de pesquisa na área e trazendo conhecimento ao nosso laboratório.”

*José Fernando Chubaci, professor do Instituto de Física da USP*



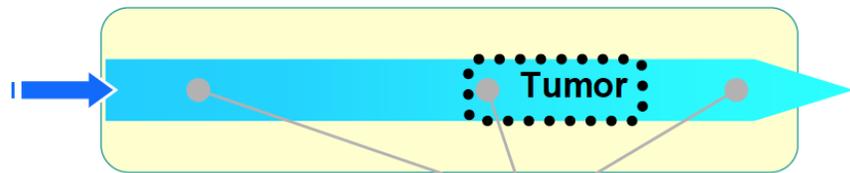
# Perspectivas



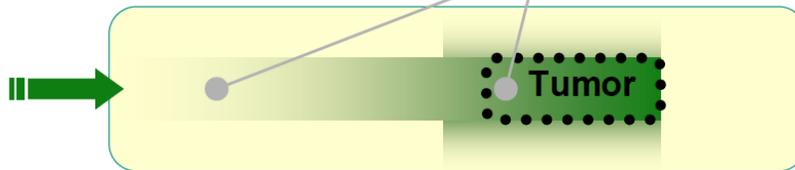
## 1. History of the heavy ion radiotherapy

### Comparison between heavy ion and other radiotherapy

#### Radiotherapy with X-ray and proton



RBEs are equal at any place



#### Heavy ion radiotherapy



RBE is higher at tumor

- Good concentrate physical dose distribution due to Bragg curve
- Lower multiple scattering in the lateral direction
- Large relative biological effectiveness (RBE) at the stopping point

# Perspectivas



## 1. History of the heavy ion radiotherapy

### HIMAC project

#### Requirements

Ion species: He to Si (to Ar for research)

Range: 25cm

Irradiation size: 15 x 15 cm (22cm dia.)

Dose rate: 5Gy/min.

- 1984 Project was funded
- 1988 Start of construction
- 1993 Complete
- 1994 Start of clinical trial

#### Size:

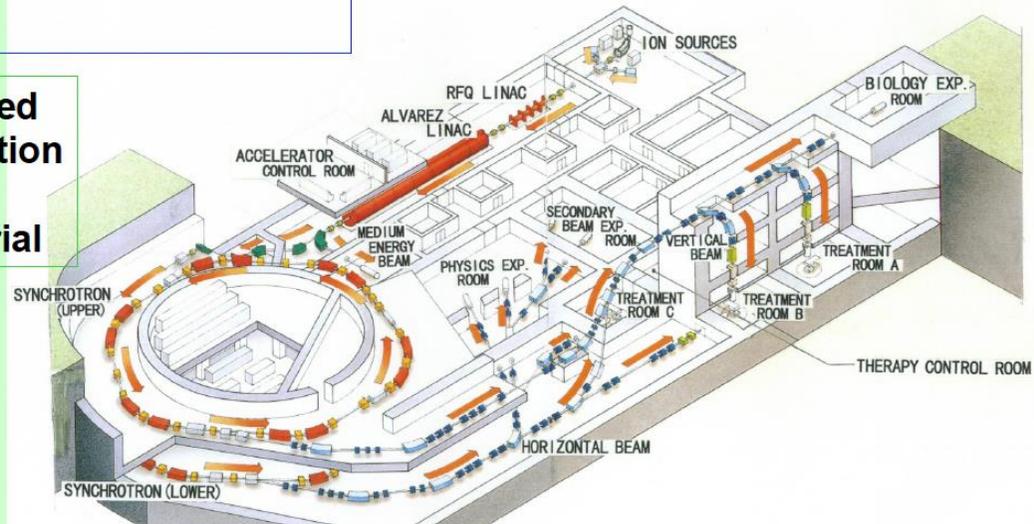
60 x 120 m

Construction cost:

32.6GJPY

(Building 14.6GJPY)

(machine 18.0GJPY)



### HIMAC (Heavy Ion Medical Accelerator in Chiba)

Y. Hirao *et al.*, Nucl. Phys. A 538, 541c (1992).

# Perspectivas

- ✓ Produção de materiais especiais como óxidos e nitretos;
- ✓ Aplicação dos sistemas de caracterização por feixes iônicos nos materiais produzidos;
- ✓ Geral:
  - ✓ aplicação de caracterização por feixes de íons em outras áreas;
  - ✓ Aplicação de feixe de íons em radioterapia.



# IBAD system: Details

Base Pressure =  $7.0 \times 10^{-5}$  Pa  
Working Pressure =  $2.0 \times 10^{-2}$  Pa

